

Introduction to Extensional Rheology

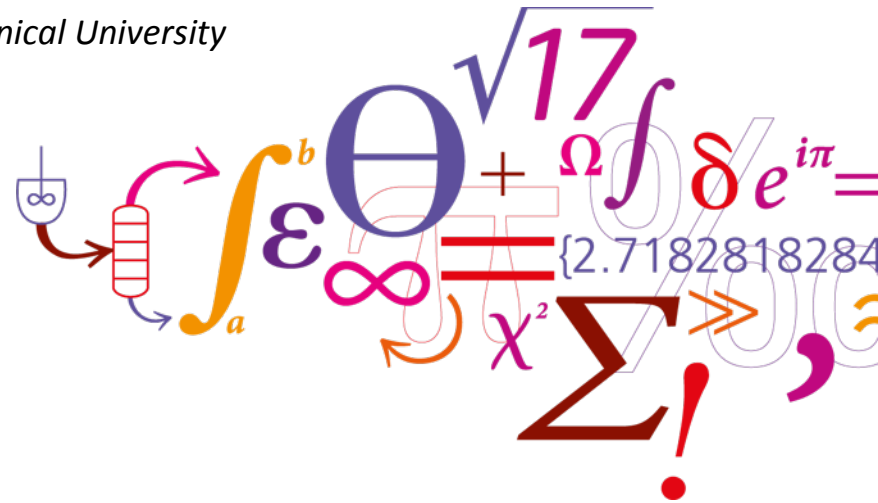
Qian Huang

Department of Chemical and Biochemical Engineering, Technical University of Denmark, 2800 Lyngby, Denmark



DoDyNet - Elementary Modules I

Sep. 25th, 2018, FORTH, Heraklion



Where do we meet extensional flow?

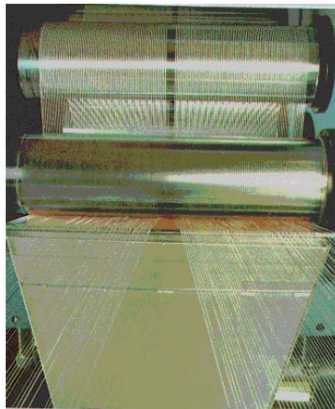
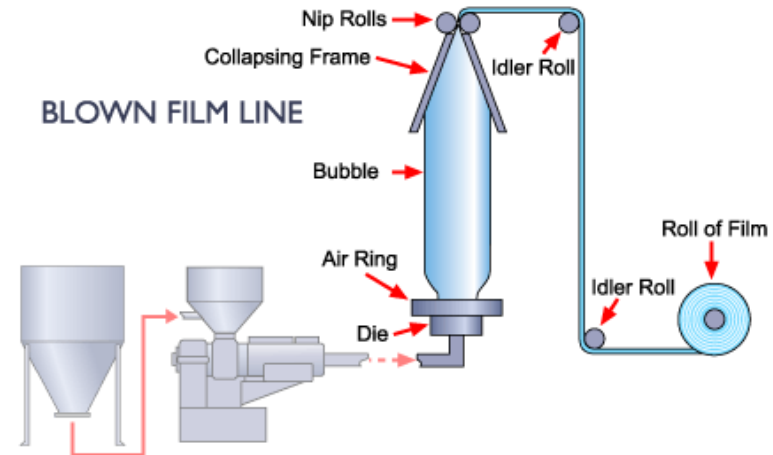


Do you know this feeling?

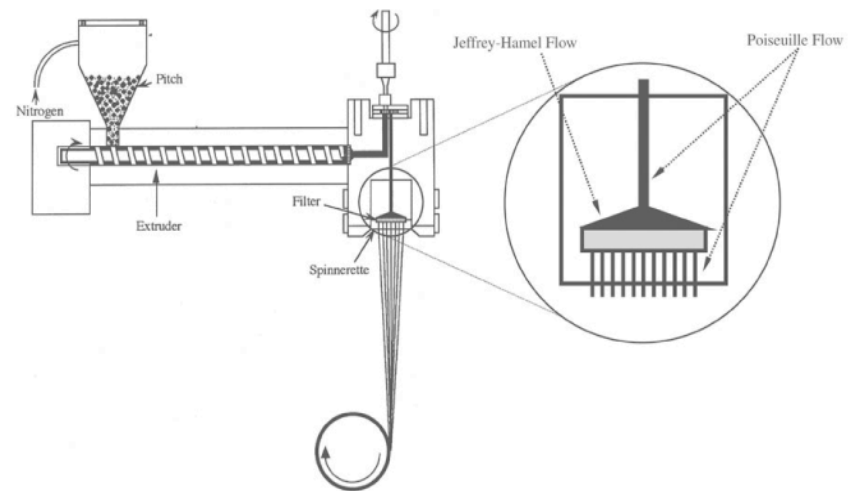
Where do we meet extensional flow?



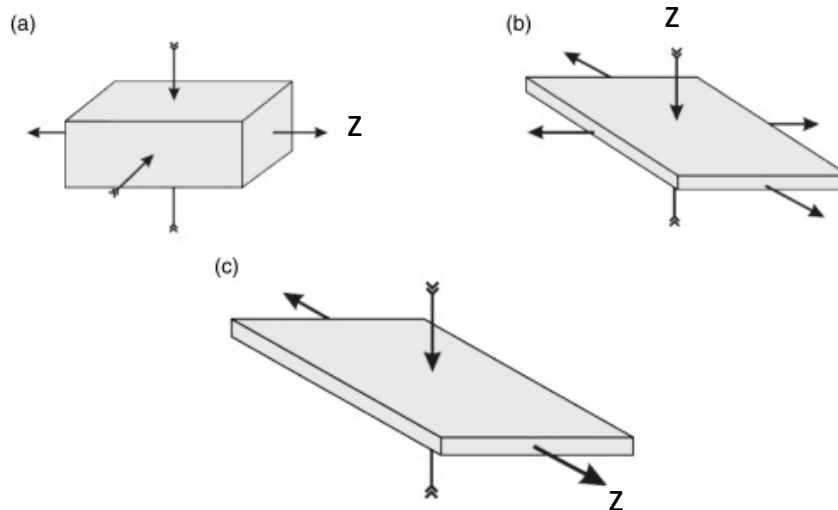
Film blowing



Fiber spinning



Typical extensional flows



Velocity fields: $(\nabla \cdot \mathbf{v}) = 0$

$$v_x = -\frac{1}{2}\dot{\epsilon}(1+b)x$$

$$v_y = -\frac{1}{2}\dot{\epsilon}(1-b)y$$

$$v_z = +\dot{\epsilon}z$$

$\dot{\epsilon}$ Strain rate, 1/s

(a) Uniaxial extensional flow

$b = 0, \dot{\epsilon} > 0$ Elongational flow

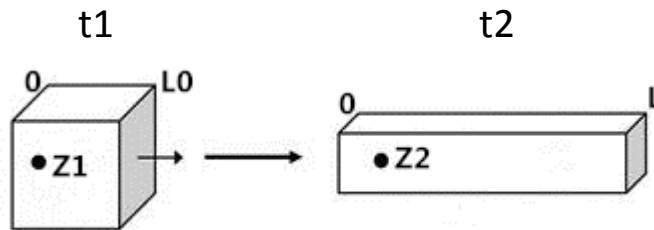
(b) Biaxial extensional flow

$b = 0, \dot{\epsilon} < 0$

(c) Planar extensional flow

$b = 1$

Uniaxial extensional flow



Velocity fields:

$$v_x = -\frac{1}{2}\dot{\epsilon}x$$

$$v_y = -\frac{1}{2}\dot{\epsilon}y$$

$$v_z = +\dot{\epsilon}z$$

Consider constant strain rate:

- $L(t)$ increases exponentially with time
- The distance between two particles along the stretching direction increases exponentially with time

$$v_z = \frac{dz}{dt} \quad \frac{dz}{dt} = \dot{\epsilon}z$$

$$\frac{1}{z}dz = \dot{\epsilon}dt$$

$$\int_{z1}^{z2} \frac{1}{z}dz = \int_{t1}^{t2} \dot{\epsilon}dt$$

$$\ln\left(\frac{z2}{z1}\right) = \dot{\epsilon}(t2 - t1)$$

$$\frac{z2}{z1} = e^{\dot{\epsilon}(t2-t1)}$$

Partical A: $z_1 \rightarrow z_2$ $z'_2 - z_2 = (z'_1 - z_1)e^{\dot{\epsilon}(t2-t1)}$

Partical B: $z'_1 \rightarrow z'_2$

$$\frac{L}{L_0} = e^{\dot{\epsilon}t} \quad \ln\left(\frac{L}{L_0}\right) = \dot{\epsilon}t \quad \dot{\epsilon}t = \epsilon$$

Stretch ratio

Hencky strain

Hencky strain rate (Stretch rate, Elongation rate...)

What happens if we stretch it with constant velocity $\frac{dL}{dt}$

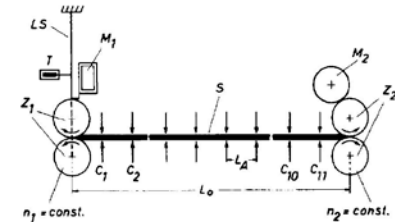
Not convenient to define the velocity field...(homework)

How to produce uniaxial extensional flow experimentally?

A VERY difficult task...

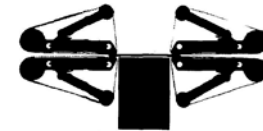
Historical - Extensional Rheometer Designs:

- The Rheometrics Melt Extensiometer (RME)
(Improved from Meissner type rheometer)



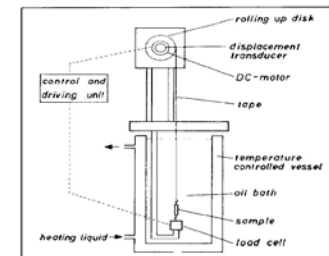
Meissner (1969), (1971), rotary clamp

- The Münstedt Tensile Rheometer (MTR)



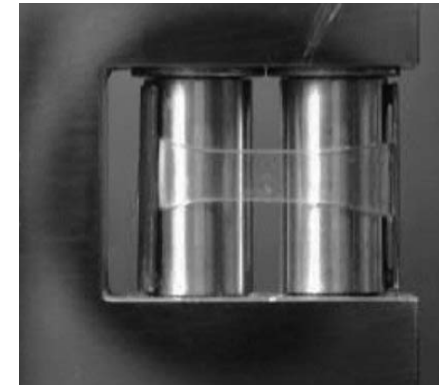
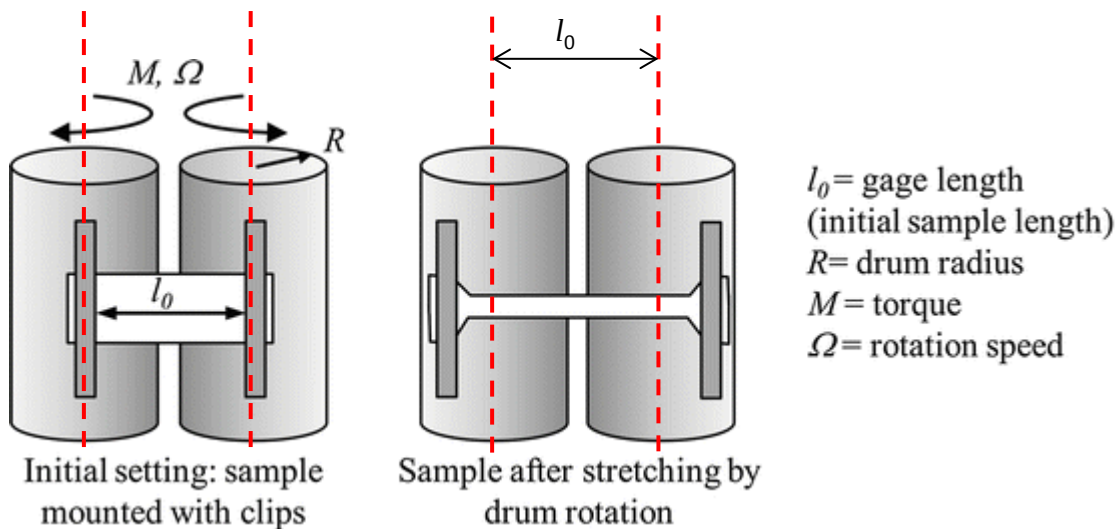
Meissner and Hostettler (1994), conveyor belt

- Sentmanat Extensional Rheometer (SER)
- Filament Stretching Rheometer (FSR)



Münstedt and Laun (1979)

SER & FSR



Sentmanat Extensional Rheometer (SER)

Sentmanat (2004)

$$\dot{\epsilon} = \frac{\Omega R}{l_0/2} = \frac{2\Omega R}{l_0}$$

$$v_z = +\dot{\epsilon}z$$

Extensional Viscosity Fixture (EVF)

TA Instruments

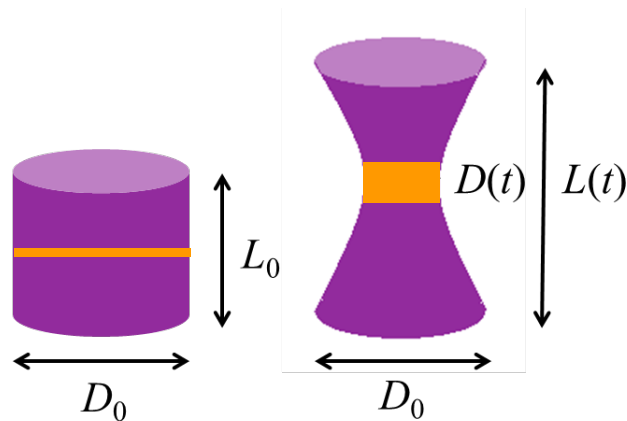
$$F = M/R$$

$$\sigma_E = F/A$$

A is calculated, not measured

(But can be measured, e.g. by camera)

SER & FSR



$$\varepsilon_z = \ln \frac{L(t)}{L_0}$$

Hencky Strain $\epsilon(t) = -2 \ln \frac{D(t)}{D_0}$

Stress $\sigma_E = \langle \sigma_{zz} - \sigma_{rr} \rangle = \frac{F(t) - m_f g / 2}{\pi (D(t) / 2)^2}$

Filament Stretching Rheometer (**FSR**)

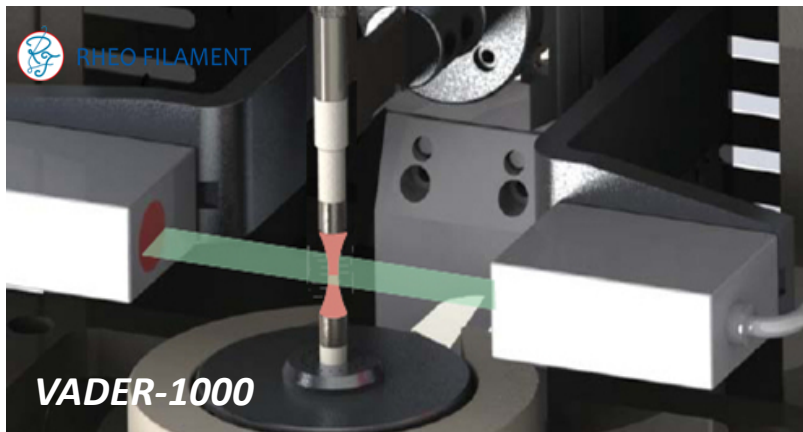
$$\begin{aligned} \sigma_{zz} &= F_z / A = F / A - p_{atm} \\ \sigma_{xx} &= F_x / A = -p_{atm} \end{aligned}$$

Something ignored:

Surface tension, inertia

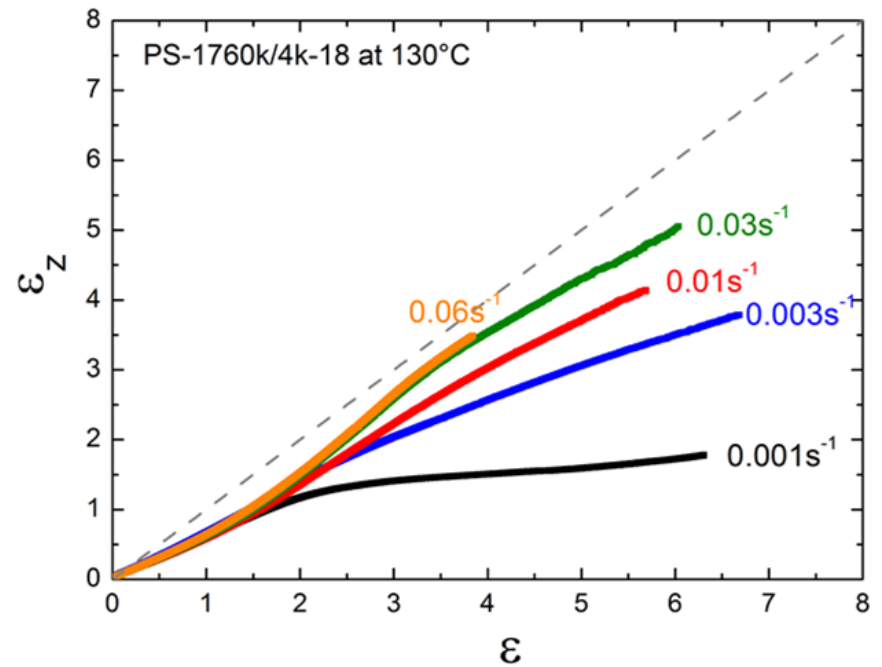
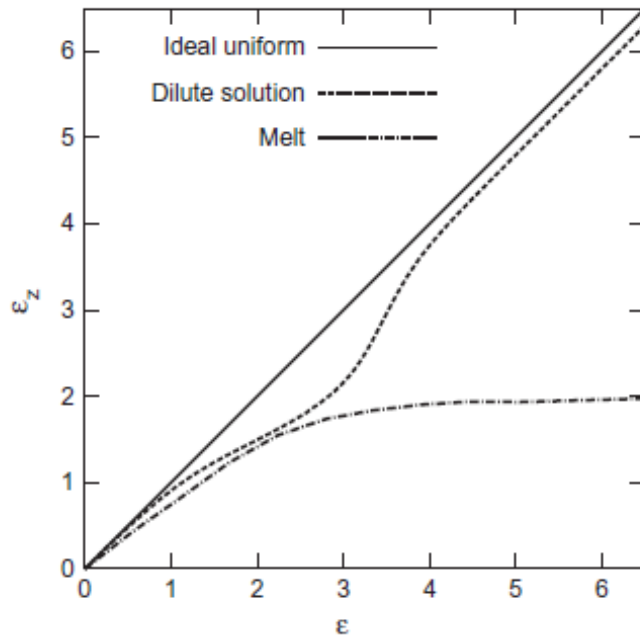
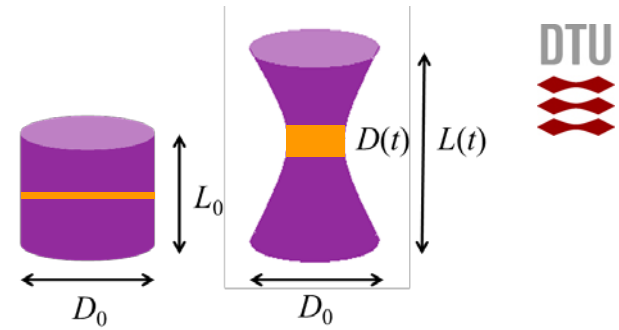
How to make stretch rate constant ???

Open loop, close loop control...

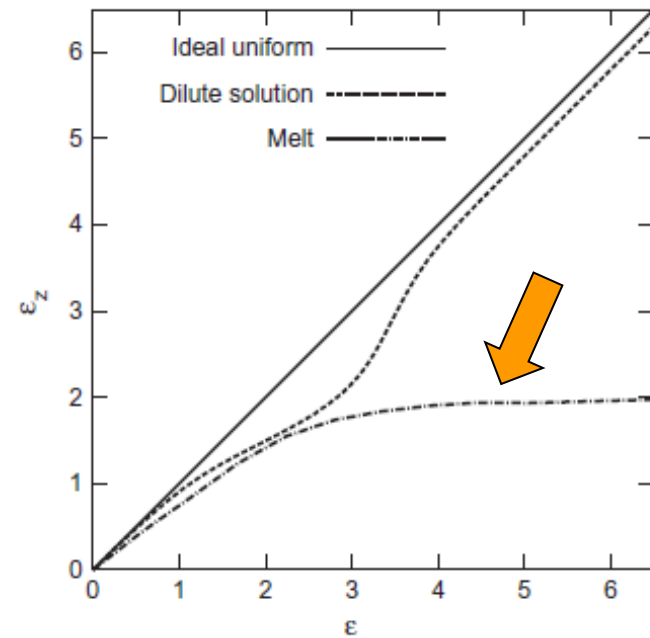
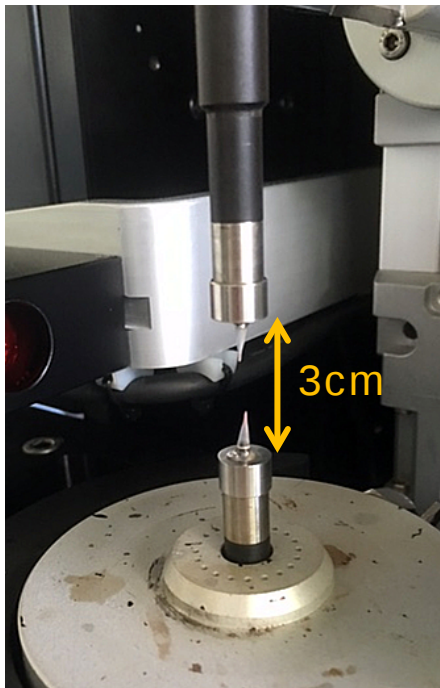


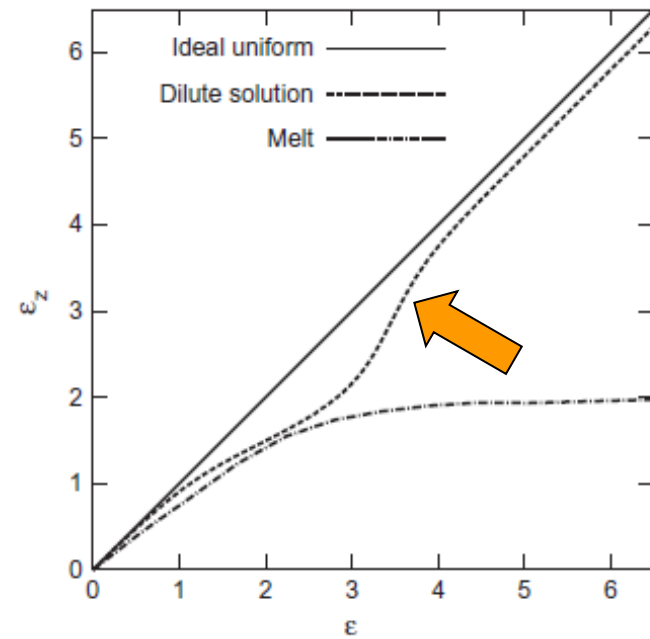
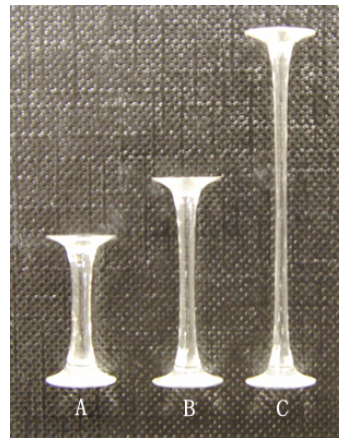
DTU (2015)

SER & FSR



SER & FSR





LVE envelope & strain hardening

In general no connection between rheometric functions in shear and extension. **Except in small deformations.**

Recall the general linear viscoelastic (LVE) model:

$$\sigma_{ij}(t) = \int_{t'=-\infty}^t G(t-t') \dot{\gamma}_{ij}(t') dt'$$

$$\dot{\gamma} = (\nabla \mathbf{v}) + (\nabla \mathbf{v})^\dagger$$

$G(t-t')$: Linear Viscoelastic Relaxation modulus

In uniaxial extension:

$$\begin{aligned} v_x &= -\frac{1}{2} \dot{\epsilon} x \\ v_y &= -\frac{1}{2} \dot{\epsilon} y \\ v_z &= +\dot{\epsilon} z \end{aligned} \quad \dot{\gamma} = \begin{pmatrix} -0.5\dot{\epsilon} & 0 & 0 \\ 0 & -0.5\dot{\epsilon} & 0 \\ 0 & 0 & \dot{\epsilon} \end{pmatrix} + \begin{pmatrix} -0.5\dot{\epsilon} & 0 & 0 \\ 0 & -0.5\dot{\epsilon} & 0 \\ 0 & 0 & \dot{\epsilon} \end{pmatrix}$$

$$= \begin{pmatrix} -\dot{\epsilon} & 0 & 0 \\ 0 & -\dot{\epsilon} & 0 \\ 0 & 0 & 2\dot{\epsilon} \end{pmatrix}$$

In simple shear:

$$\dot{\gamma} = \begin{pmatrix} 0 & \dot{\gamma}_0 & 0 \\ \dot{\gamma}_0 & 0 & 0 \\ 0 & 0 & 0 \end{pmatrix}$$

LVE envelope & strain hardening

In general no connection between rheometric functions in shear and extension. **Except in small deformations.**

Recall the general linear viscoelastic (LVE) model:

$$\sigma_{ij}(t) = \int_{t'=-\infty}^t G(t-t') \dot{\gamma}_{ij}(t') dt'$$

$$\dot{\gamma} = (\nabla \mathbf{v}) + (\nabla \mathbf{v})^\dagger$$

$G(t-t')$: Linear Viscoelastic Relaxation modulus

In simple shear: $\sigma_{yx}(t > 0) = \int_{t'=0}^t G(t-t') \dot{\gamma}_0 dt'$

$$\dot{\gamma} = \begin{pmatrix} 0 & \dot{\gamma}_0 & 0 \\ \dot{\gamma}_0 & 0 & 0 \\ 0 & 0 & 0 \end{pmatrix}$$

$$\eta^+(t > 0) = \int_{t'=0}^t G(t-t') dt'$$

LVE envelope: η^+ vs. t



Transient shear viscosity, or **Shear stress growth coefficient** ('official name')

Multi-mode Maxwell
model (with n modes)

$$G(t-t') = \sum_{k=1}^n \frac{\eta_k}{\lambda_k} e^{-(t-t')/\lambda_k}$$

$$\eta^+ = \sum_{k=1}^n \eta_k (1 - e^{-t/\lambda_k})$$

LVE envelope & strain hardening

In general no connection between rheometric functions in shear and extension. **Except in small deformations.**

Recall the general linear viscoelastic (LVE) model:

$$\sigma_{ij}(t) = \int_{t'=-\infty}^t G(t-t') \dot{\gamma}_{ij}(t') dt' \quad \dot{\gamma} = (\nabla \mathbf{v}) + (\nabla \mathbf{v})^\dagger$$

$G(t-t')$: Linear Viscoelastic Relaxation modulus

In uniaxial extension: $\sigma_{zz} - \sigma_{xx} = \int_{t'=0}^t G(t-t') (\dot{\gamma}_{zz} - \dot{\gamma}_{xx}) dt' = 3 \int_{t'=0}^t G(t-t') \dot{\epsilon} dt'$

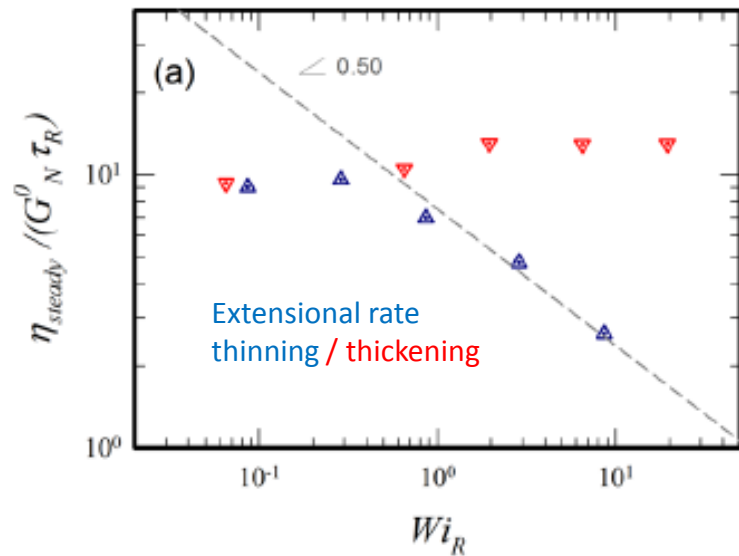
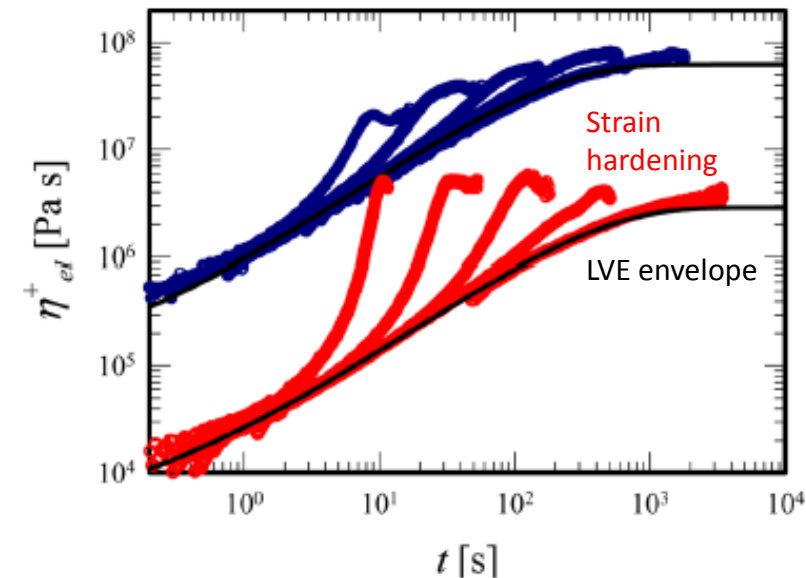
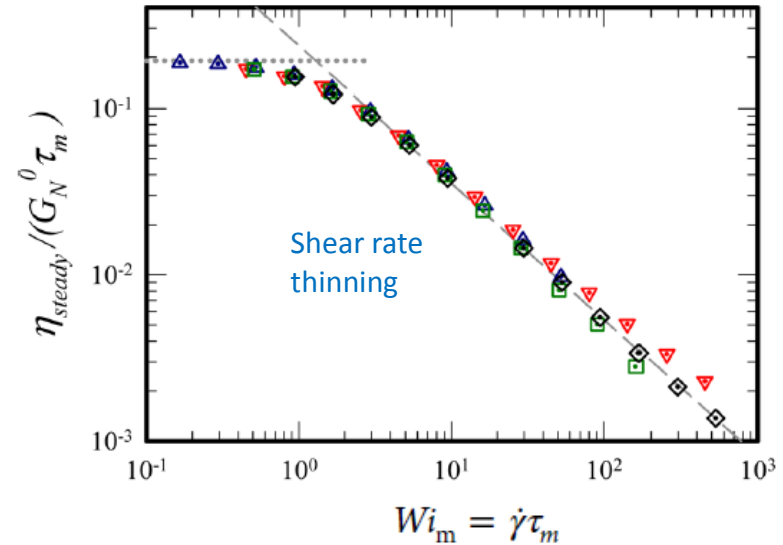
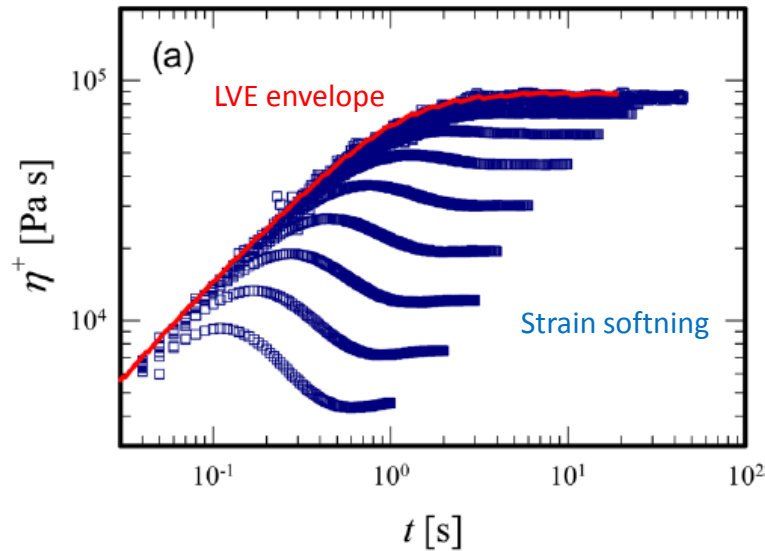
$$\dot{\gamma} = \begin{pmatrix} -\dot{\epsilon} & 0 & 0 \\ 0 & -\dot{\epsilon} & 0 \\ 0 & 0 & 2\dot{\epsilon} \end{pmatrix} \quad \eta_E^+ = \frac{\sigma_{zz} - \sigma_{xx}}{\dot{\epsilon}} = 3 \int_{t'=0}^t G(t-t') dt' \quad \text{LVE envelope: } \eta_E^+ \text{ vs. } t$$



Transient extensional viscosity, or **Extensional stress growth coefficient** ('official name')

$$\eta^+(t > 0) = \int_{t'=0}^t G(t-t') dt'$$

LVE envelope & strain hardening



LVE envelope & strain hardening

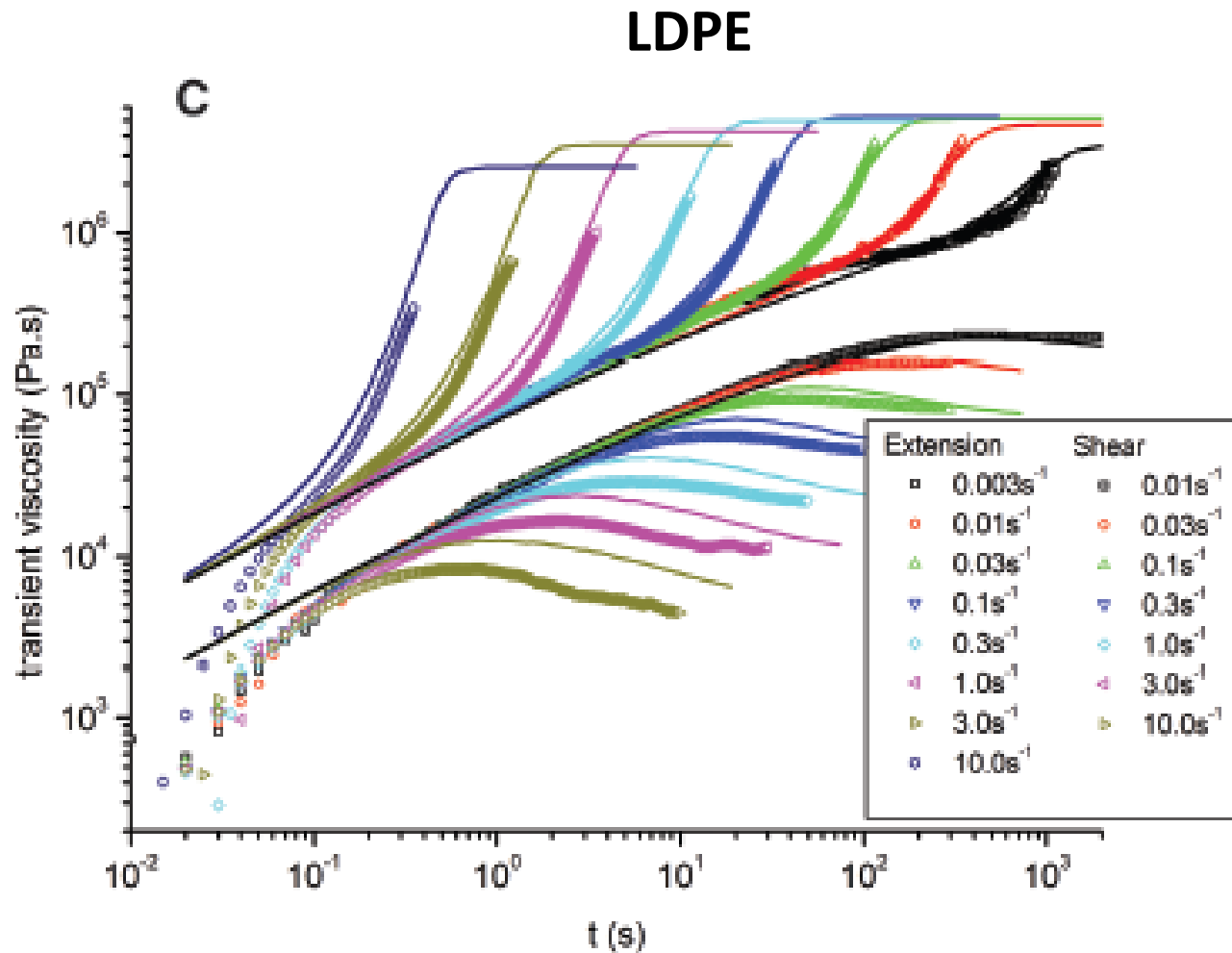


Figure from: Read et al., Science 2011, 333, 1871-1874

Extensional rheology

Linear molecules (melts & solutions)

Branched molecules

Associating polymers with hydrogen bonding

Creep (constant stress) measurements

Combining with SANS

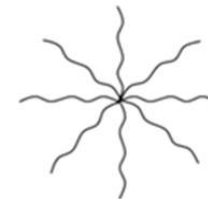
Influence on mechanical properties



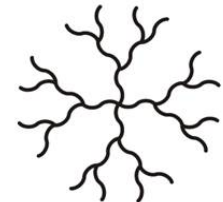
Linear



Comb



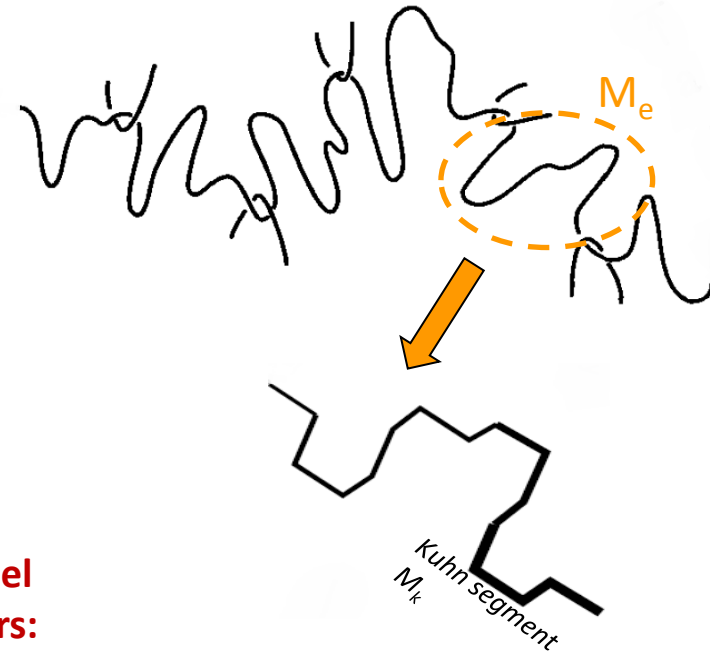
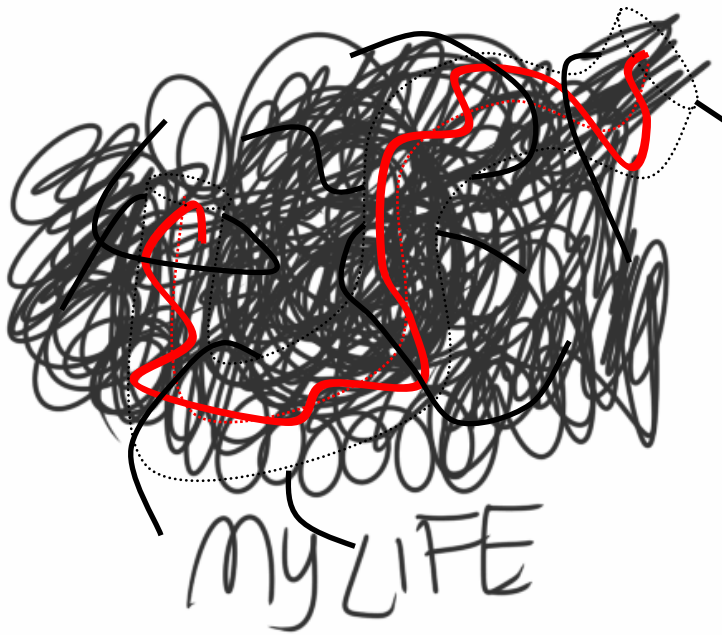
Star



Dendritic

Linear molecules (melts & solutions)

Simplest case: Monodisperse linear polymers



Tube model parameters:

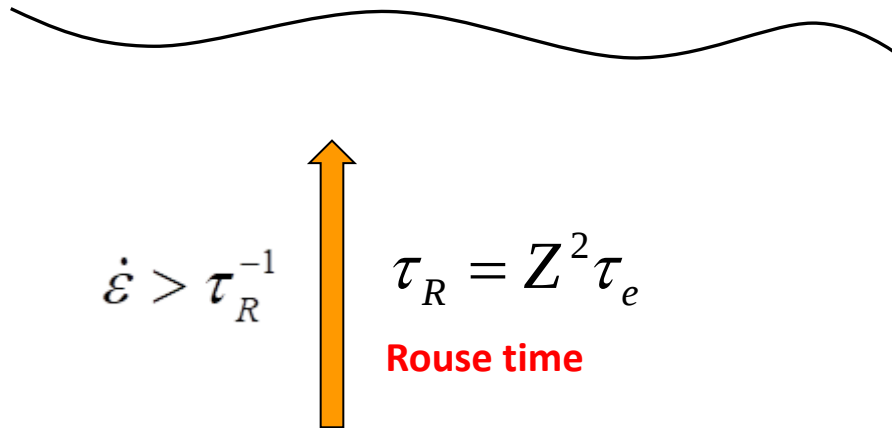
Number of entanglements per chain: $Z = M/M_e$

Number of Kuhn steps per entanglement: $N_e = M_e/M_k$

Entanglement Relaxation time: τ_e

Linear molecules (melts & solutions)

Chain stretching



$$\dot{\epsilon} > \tau_R^{-1}$$

$$\tau_R = Z^2 \tau_e$$

Rouse time

Maximum Stretch Ratio:

$$\lambda_{\max} = \frac{Nb}{Za} = \sqrt{\frac{N}{Z}} = \sqrt{N_e}$$

Chain orientation



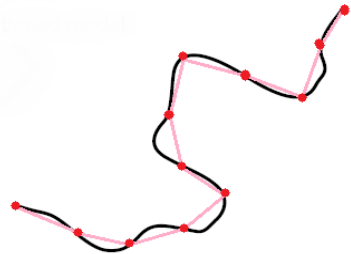
$$\dot{\epsilon} > \tau_d^{-1}$$

$$\tau_d = cZ^{3.4} \tau_e$$

Reptation time

$$R^2 = Za^2 = Nb^2$$

Linear molecules (melts & solutions)



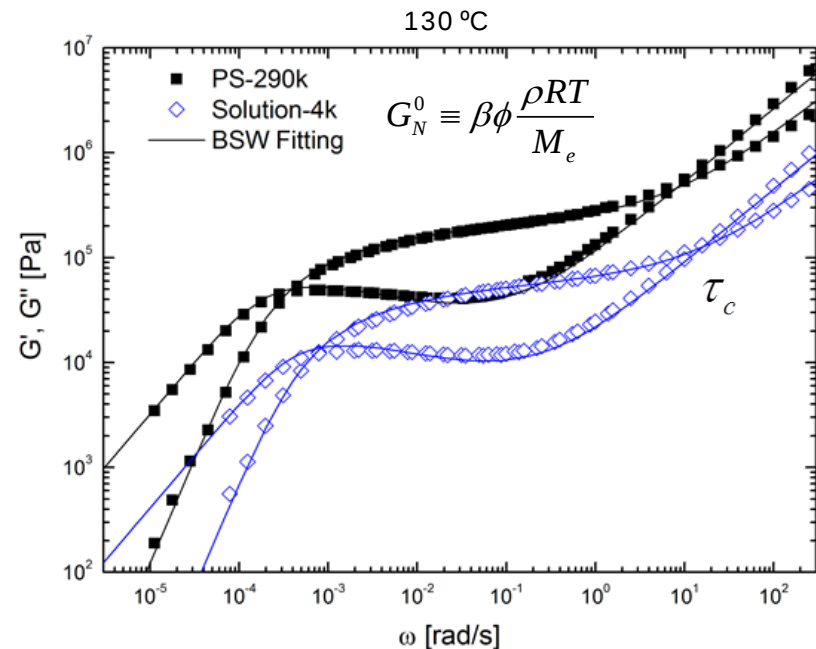
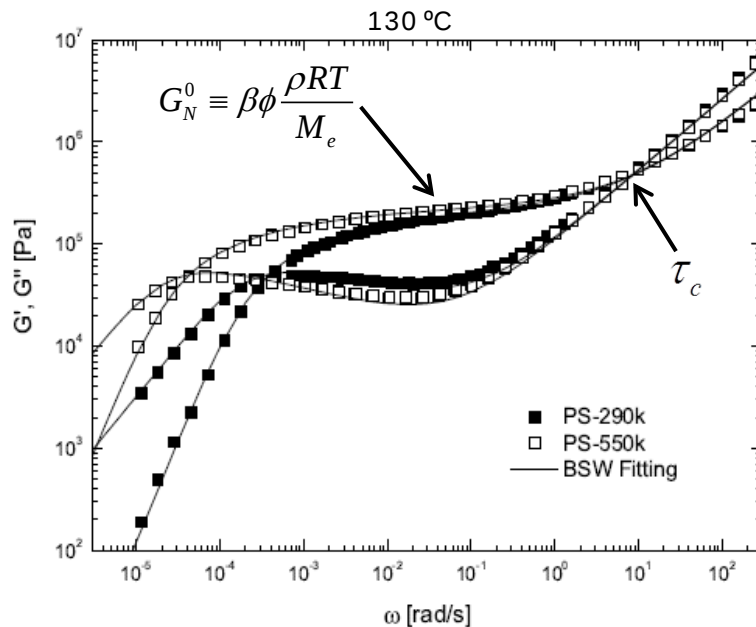
PS-290k (Z=22)



PS-550k (Z=41)

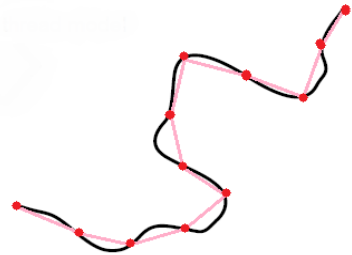


52% PS-550k (Z=22)



Huang et al., ACS Macro Letters 2, 741–744 (2013)

Linear molecules (melts & solutions)



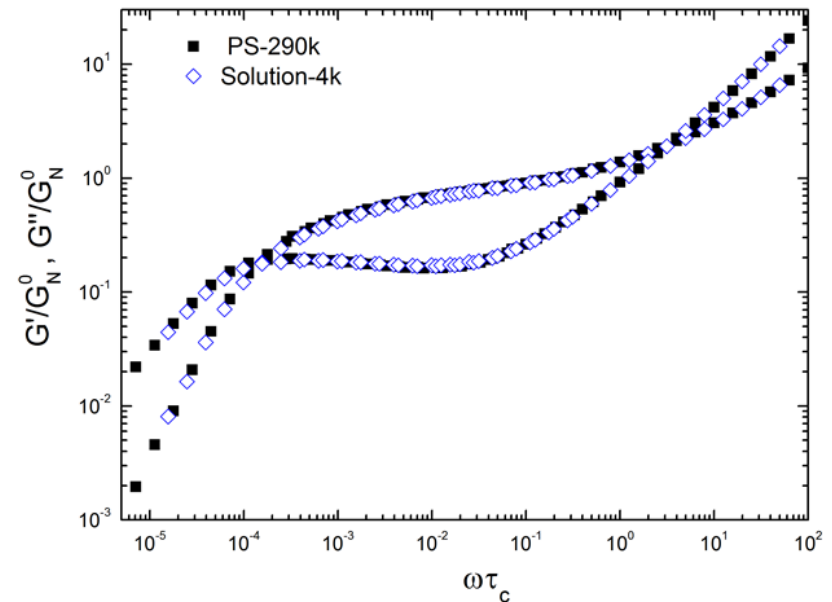
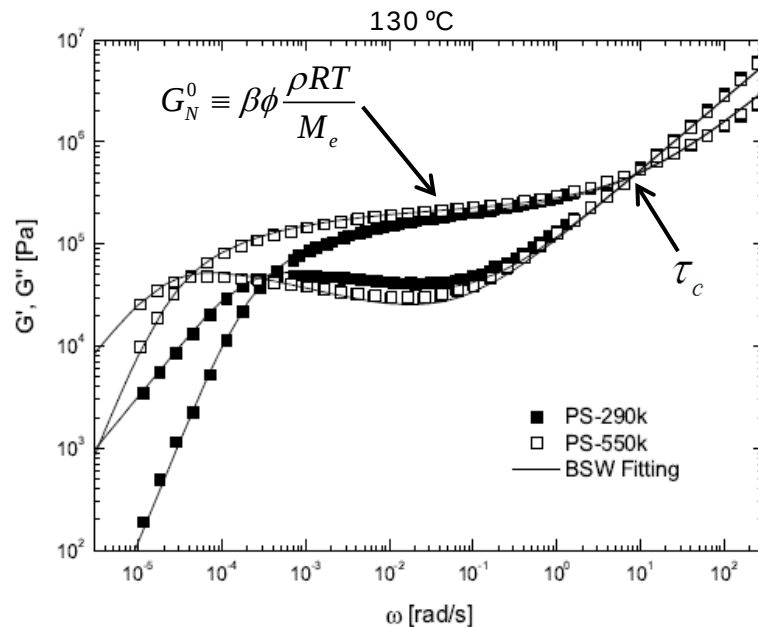
PS-290k (Z=22)



PS-550k (Z=41)

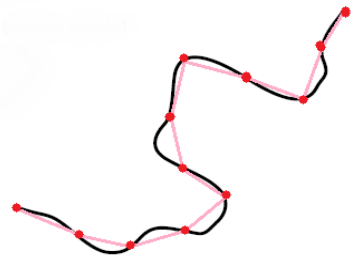


52% PS-550k (Z=22)



Huang et al., ACS Macro Letters 2, 741–744 (2013)

Linear molecules (melts & solutions)



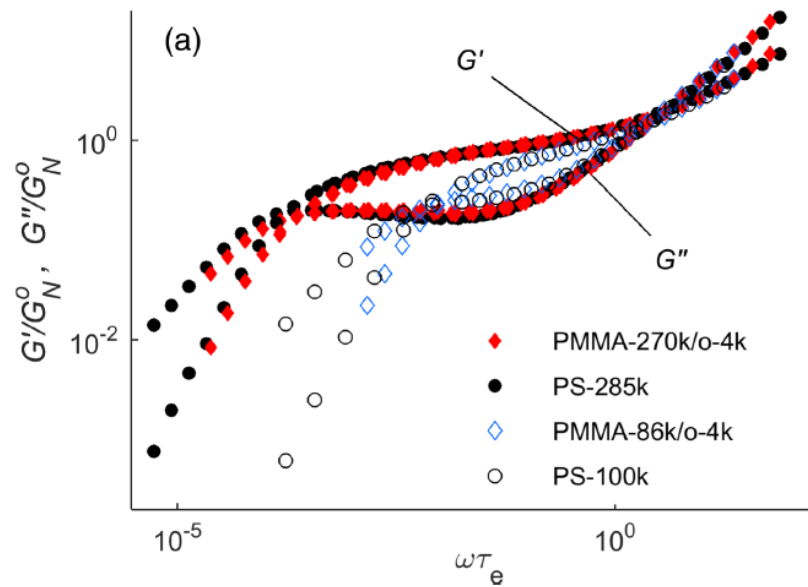
PS-290k ($Z=22$)



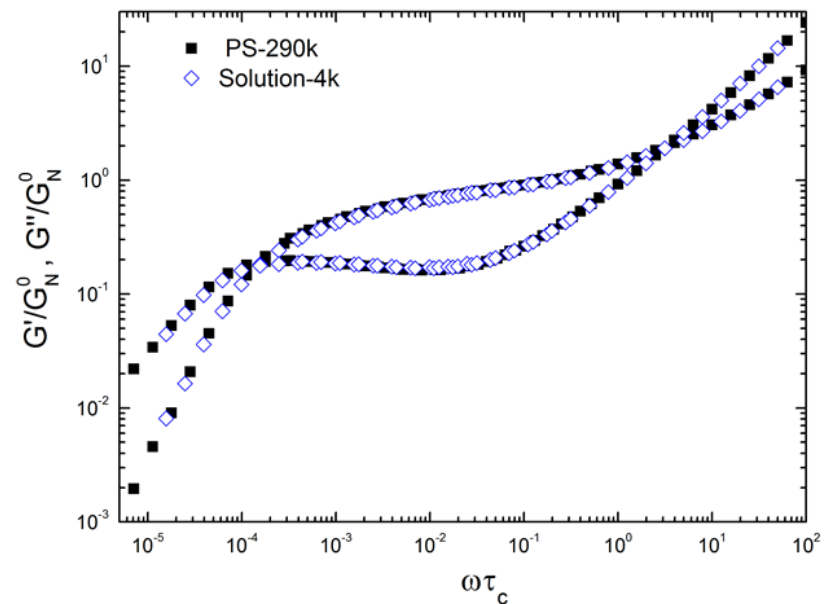
PS-550k ($Z=41$)



52% PS-550k ($Z=22$)

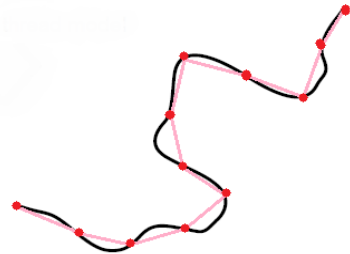


Wingstrand et al., PRL 115, 078302 (2015)



Huang et al., ACS Macro Letters 2, 741–744 (2013)

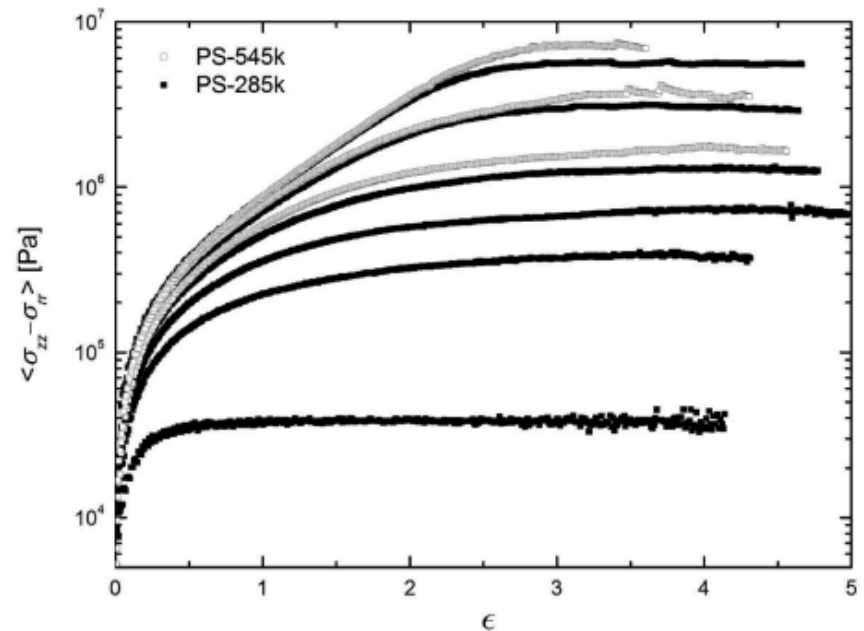
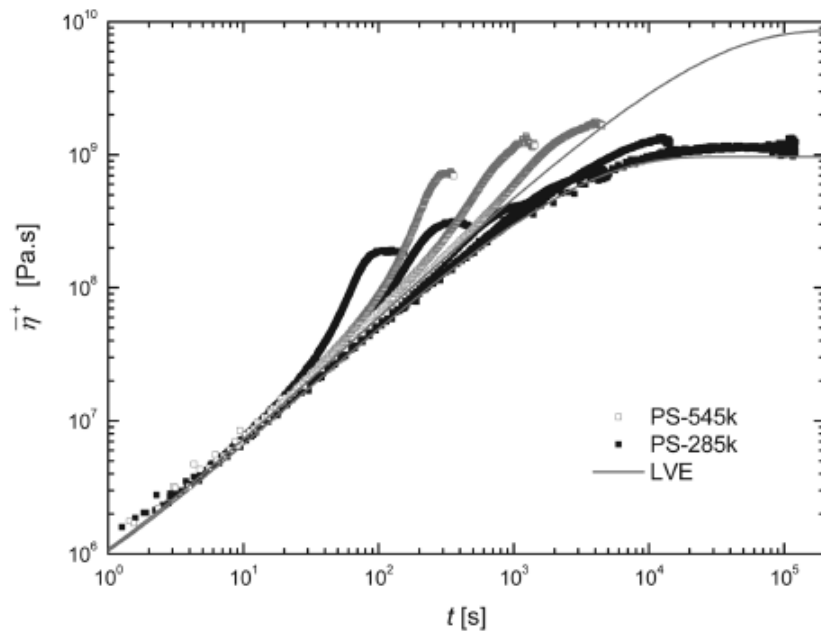
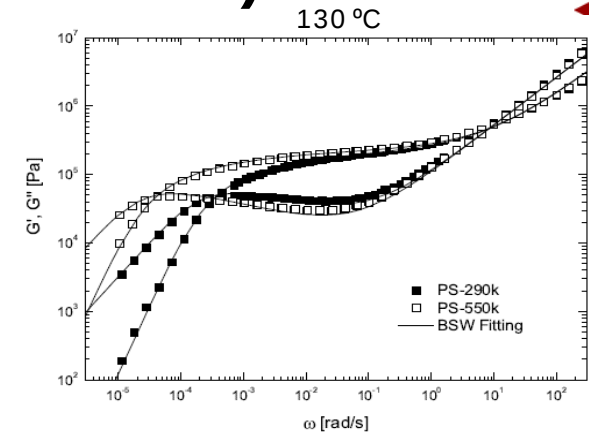
Linear molecules (melts & solutions)



PS-290k (Z=22)

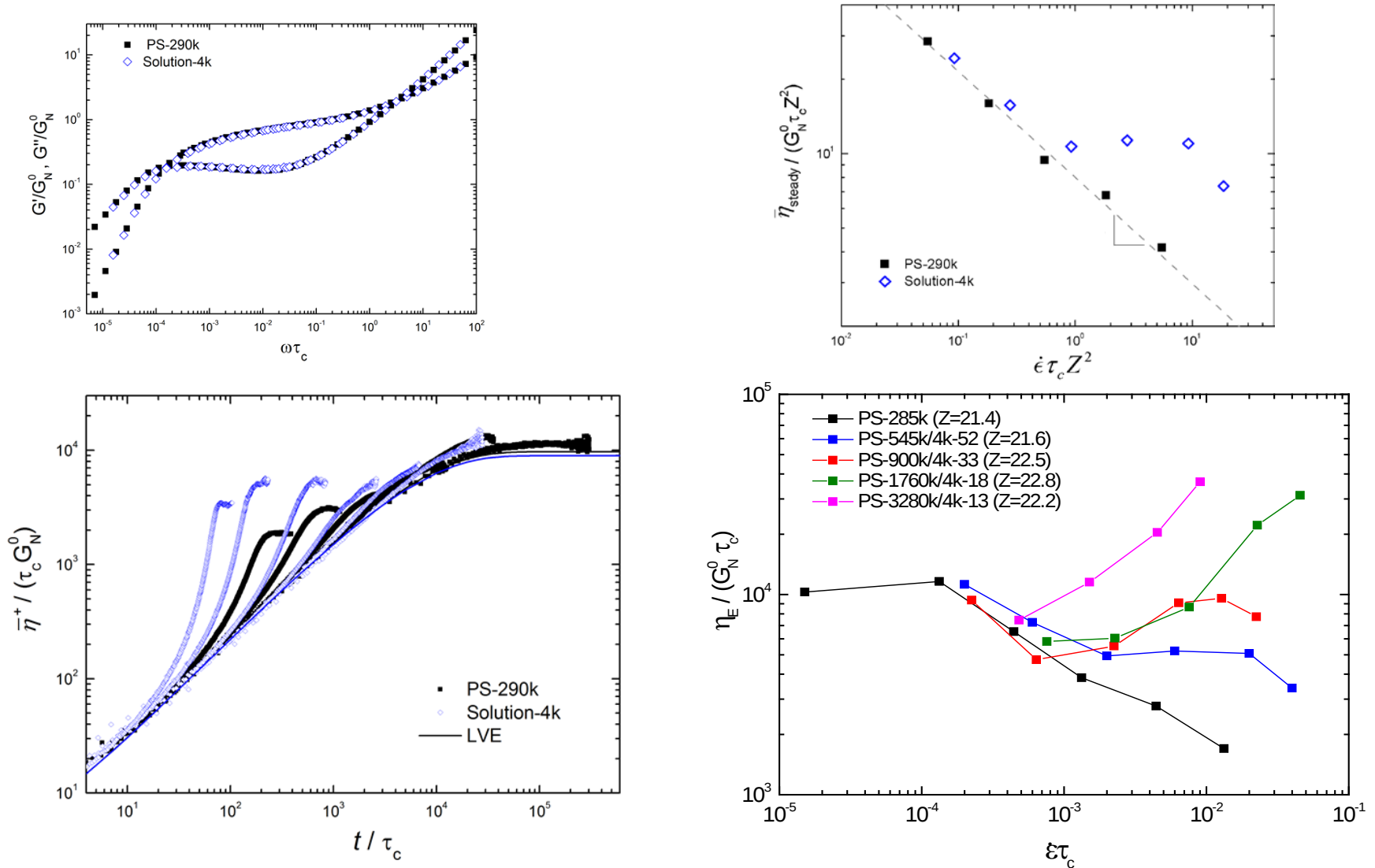


PS-550k (Z=41)



Huang et al., *Macromolecules* 2013, 46, 5026–5035

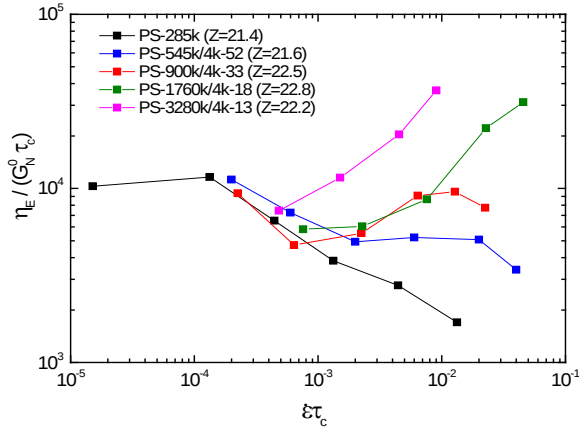
Linear molecules (melts & solutions)



Huang et al., ACS Macro Letters 2, 741–744 (2013)

Huang et al., Macromolecules 48, 4158–4163 (2015)

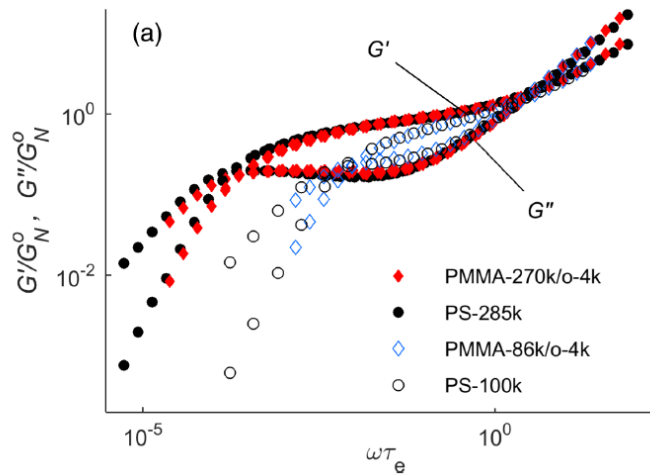
Linear molecules (melts & solutions)



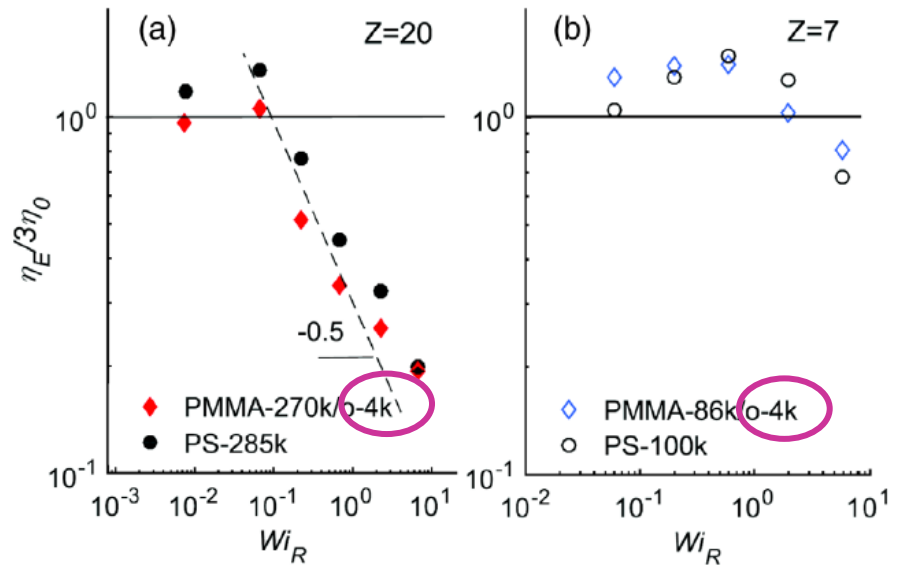
Maximum chain stretch ratio:

$$\lambda_{\max} = \sqrt{N_e}$$

Need to choose the oligomer as long as possible



Same Z, same N_e



Wingstrand et al., PRL 115, 078302 (2015)

Linear molecules (melts & solutions)

Simplest case: Not fully understood yet...

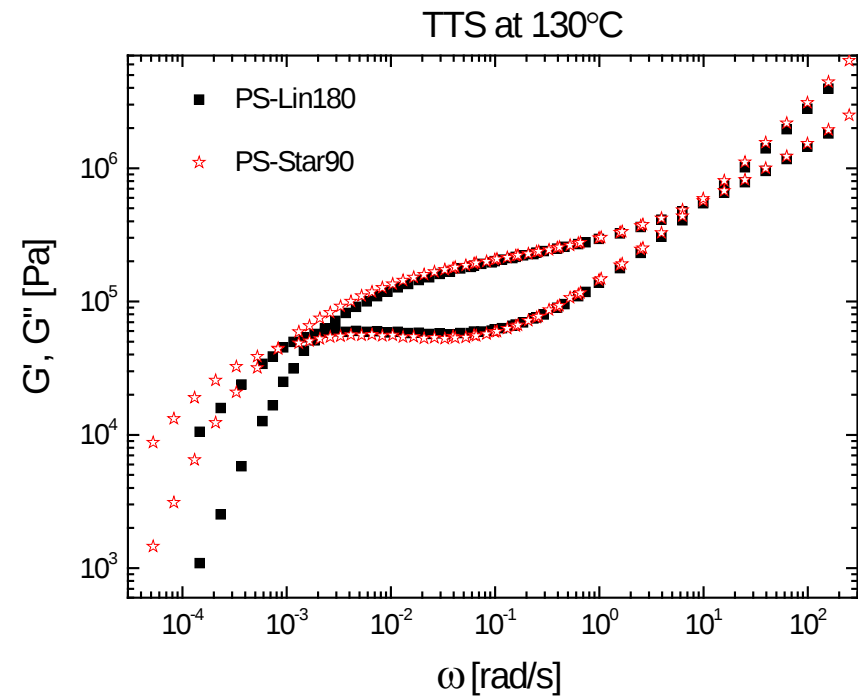
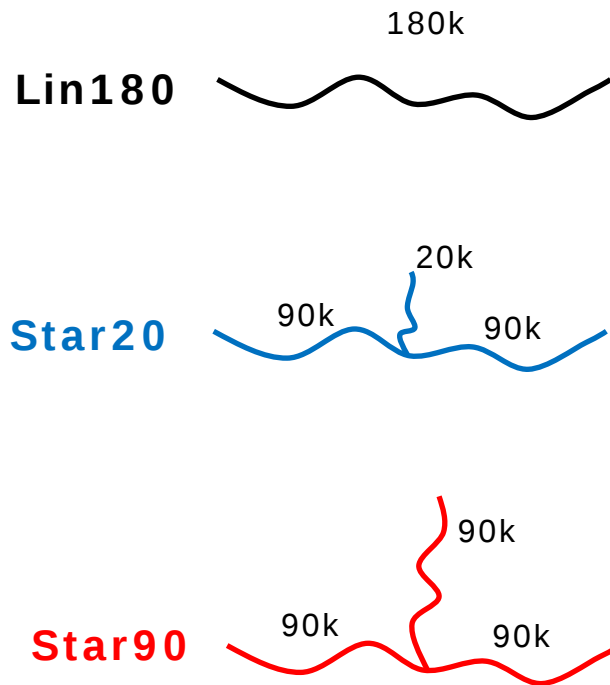


Entanglements = Mess...

Entanglements = Fun!

Branched molecules - Star

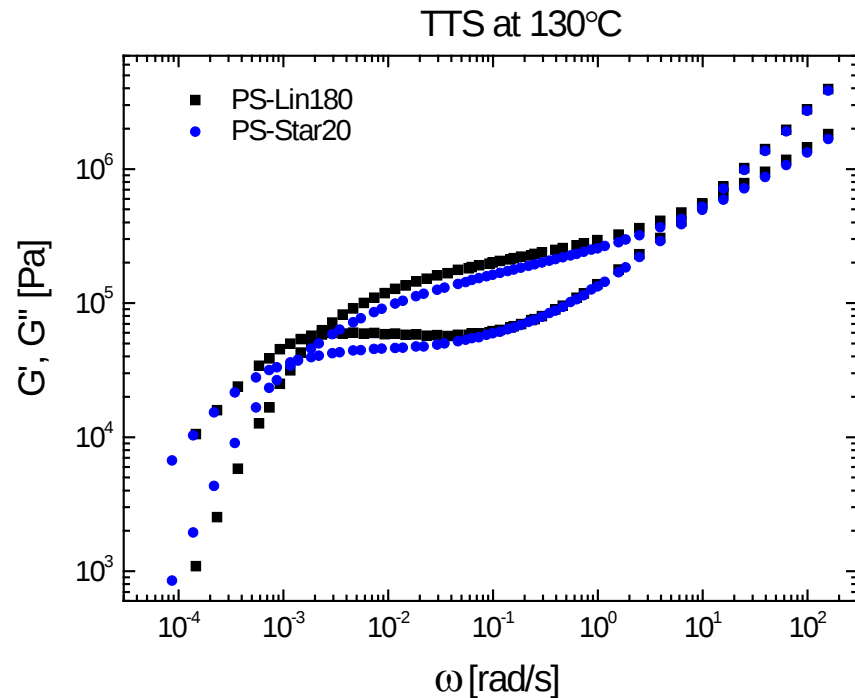
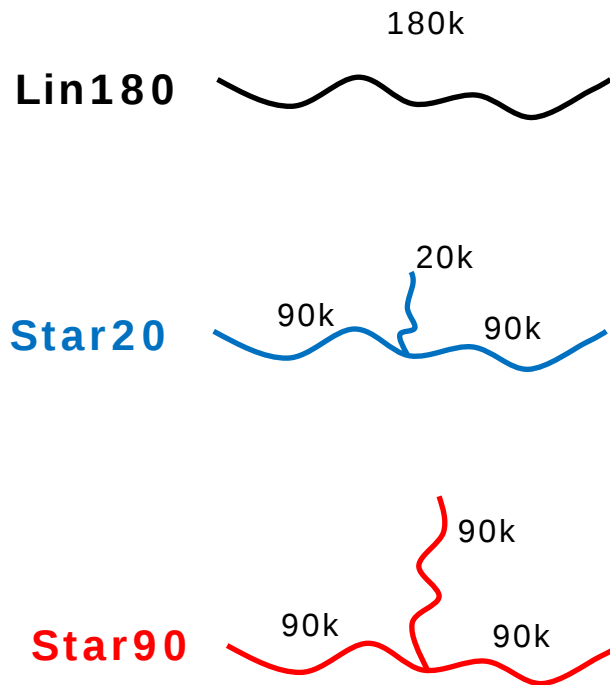
Simplest case: 3-arm star



Huang et al., Macromolecules 49, 6694–6699 (2016)

Branched molecules - Star

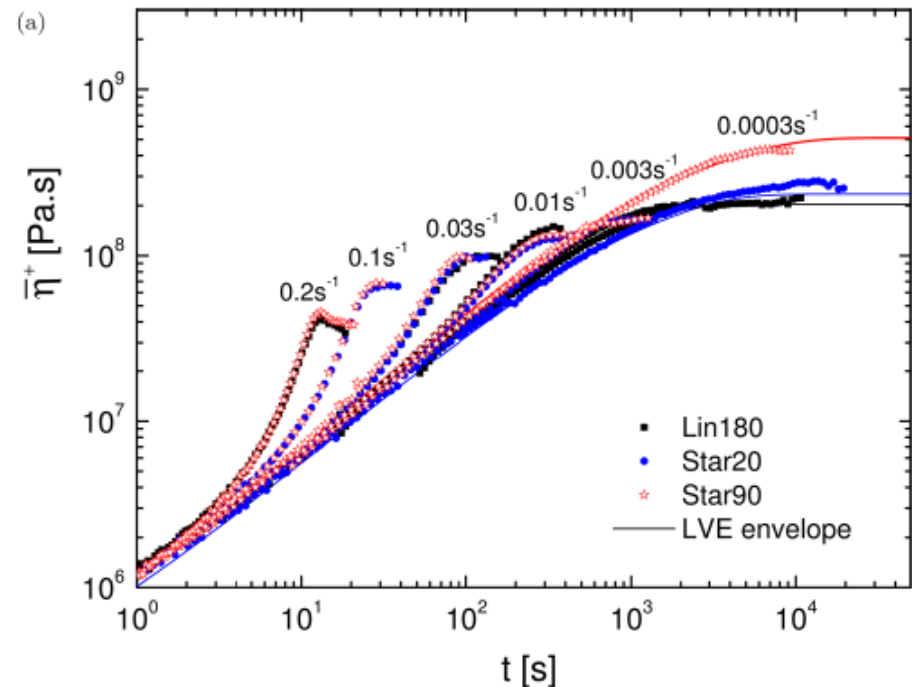
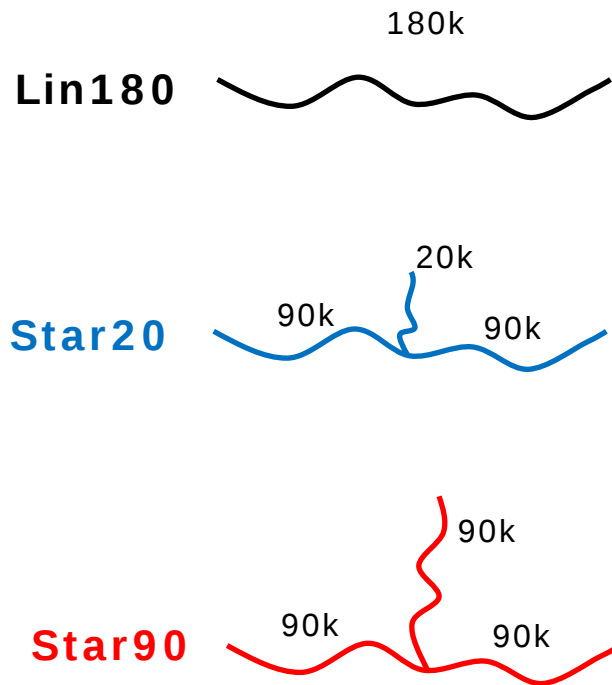
Simplest case: 3-arm star



Huang et al., *Macromolecules* 49, 6694–6699 (2016)

Branched molecules - Star

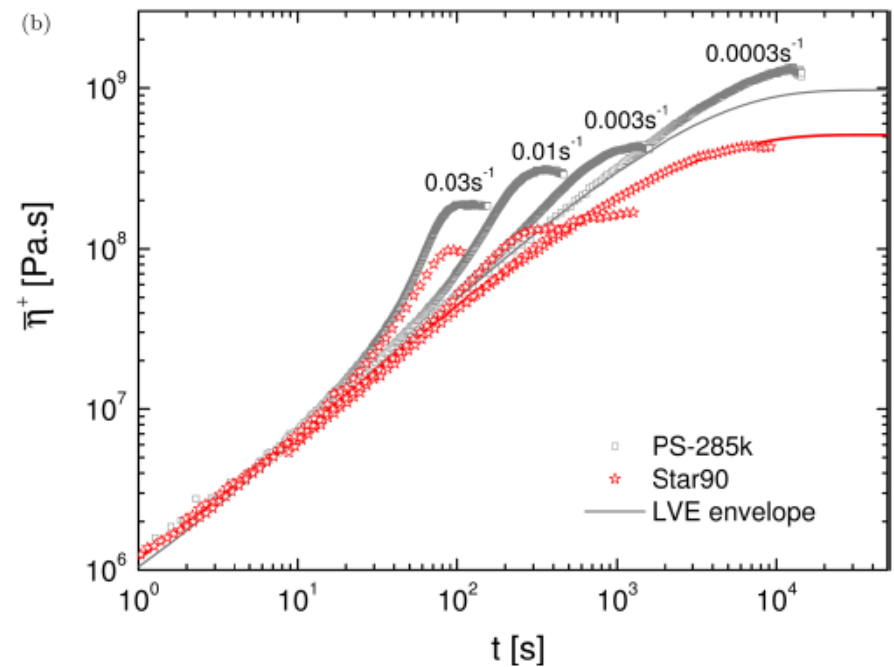
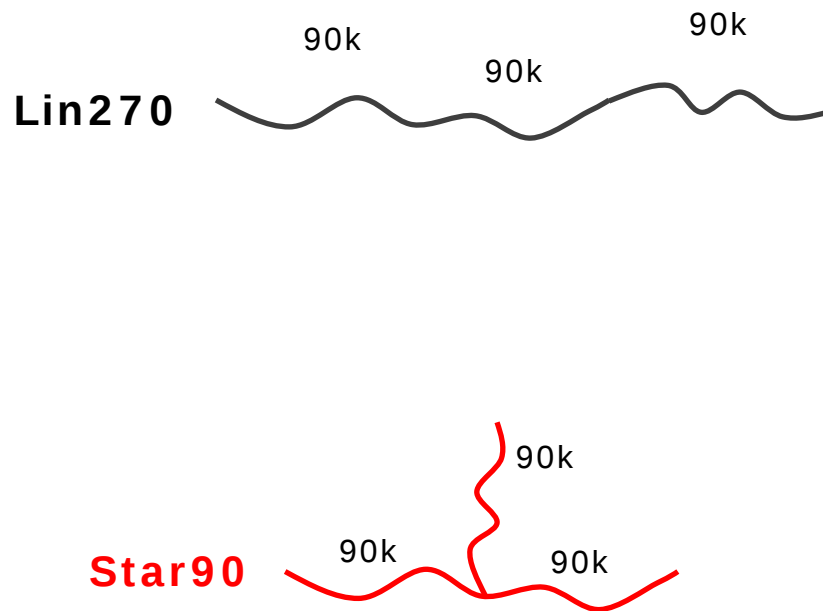
Simplest case: 3-arm star



Huang et al., *Macromolecules* 49, 6694–6699 (2016)

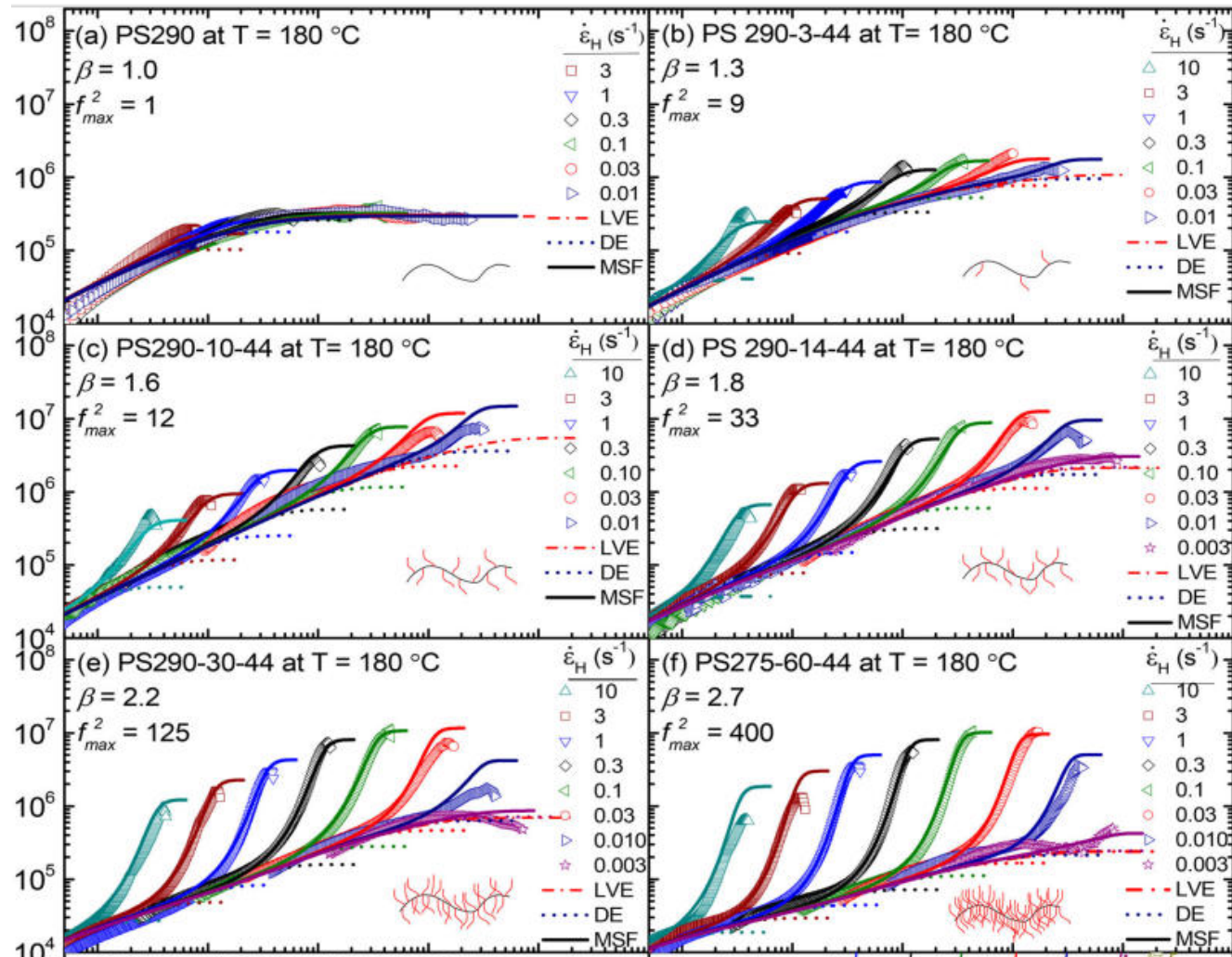
Branched molecules - Star

Simplest case: 3-arm star



Huang et al., Macromolecules 49, 6694–6699 (2016)

Branched molecules - Comb

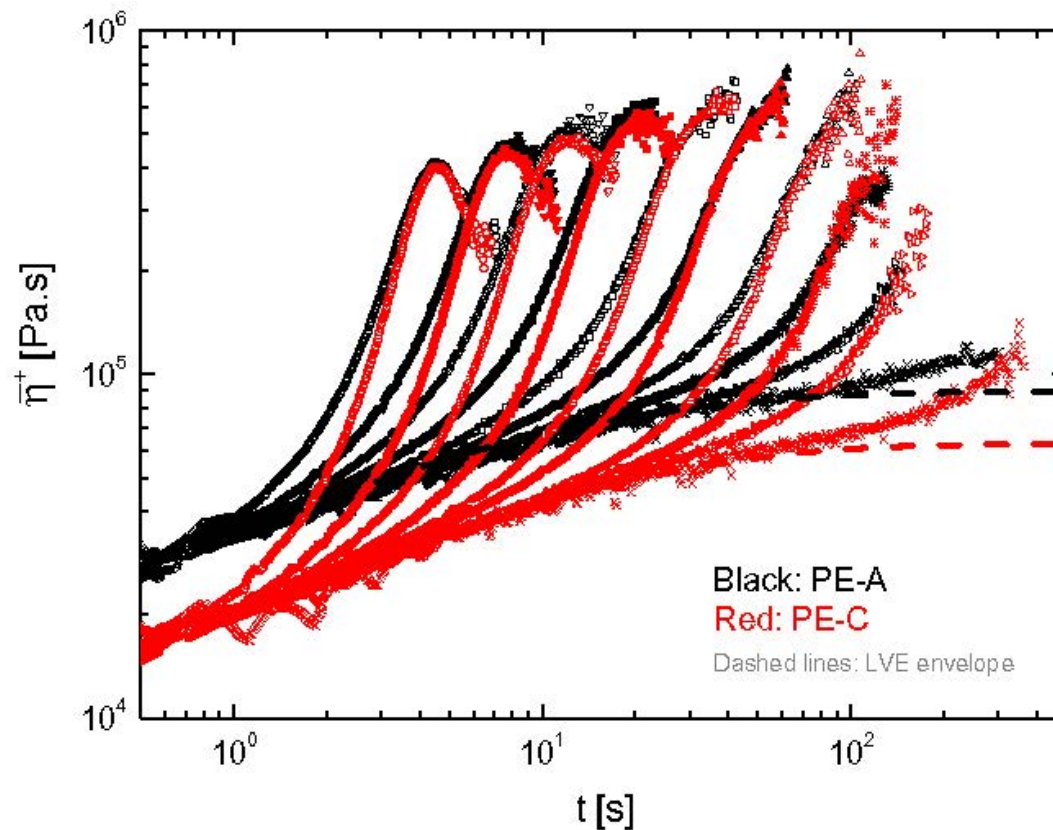


$$\text{DE: } \boldsymbol{\sigma}(t) = \int_{-\infty}^t M(t-t') \mathbf{S}_{DE}(t, t') dt'$$

$$\text{MSF: } \boldsymbol{\sigma}(t) = \int_{-\infty}^t M(t-t') f^2(t, t') \mathbf{S}_{DE}(t, t') dt'$$

Branched molecules - Commercial polymers

Most complicated case: randomly branched, polydisperse...

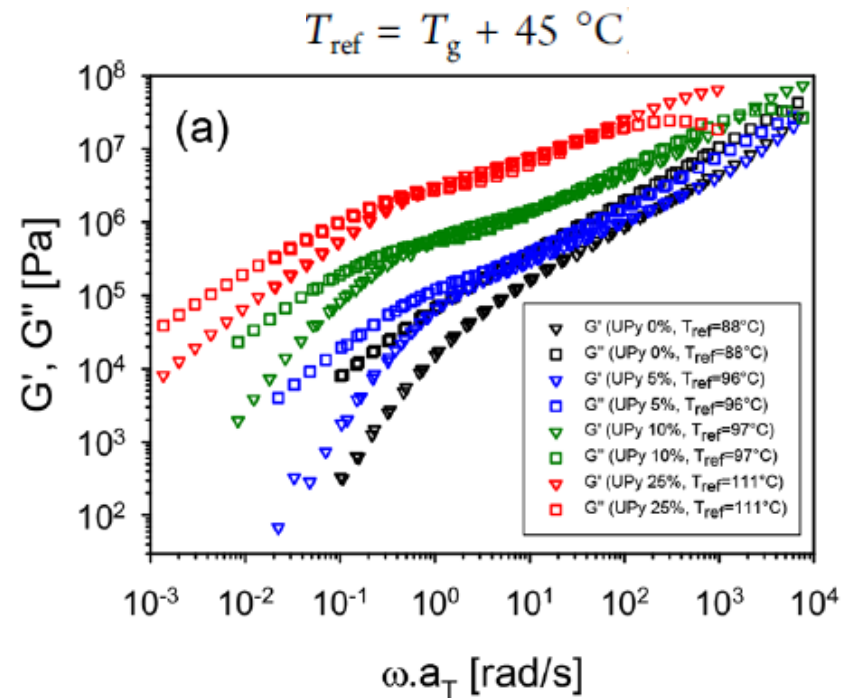
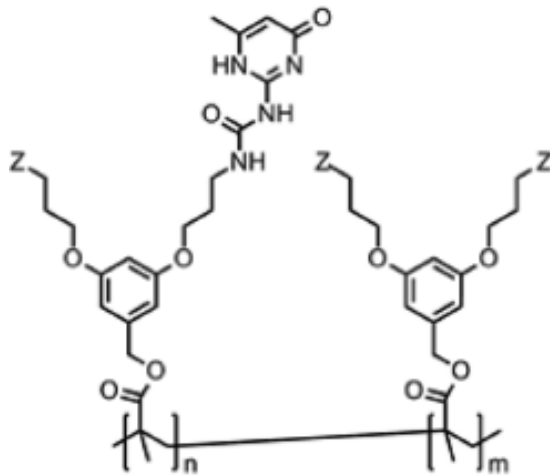


Huang et al., Rheol Acta (2016) 55:343–350

Associating polymers

Associating polymers with hydrogen bonding

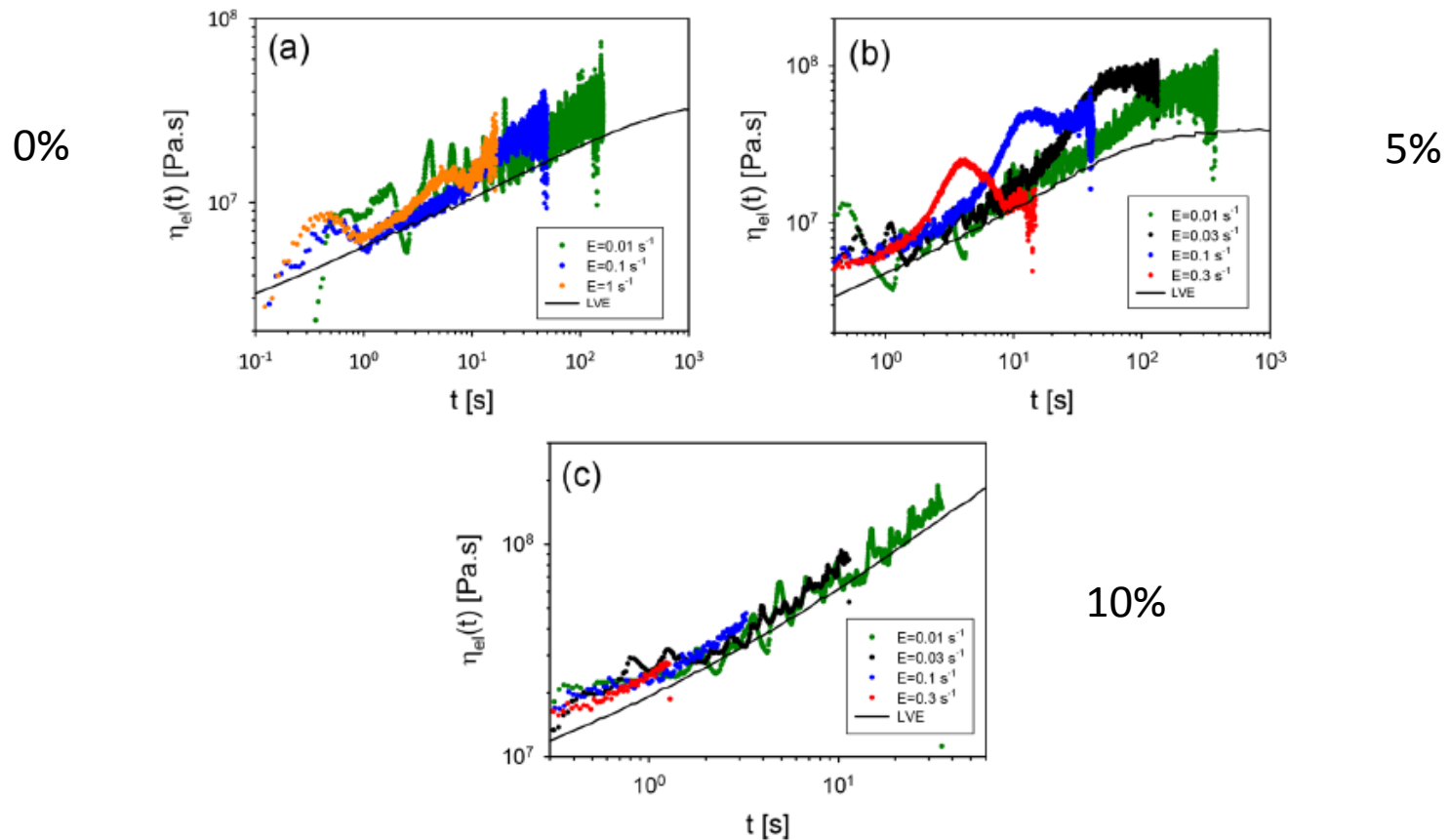
PMMA based dendronized polymers, with strongly hydrogen bonding 2-ureido-4[1H]-pyrimidinone (UPy) moieties



Scherz and Costanzo et al., *Macromolecules* 2017, 50, 5176–5187

Associating polymers

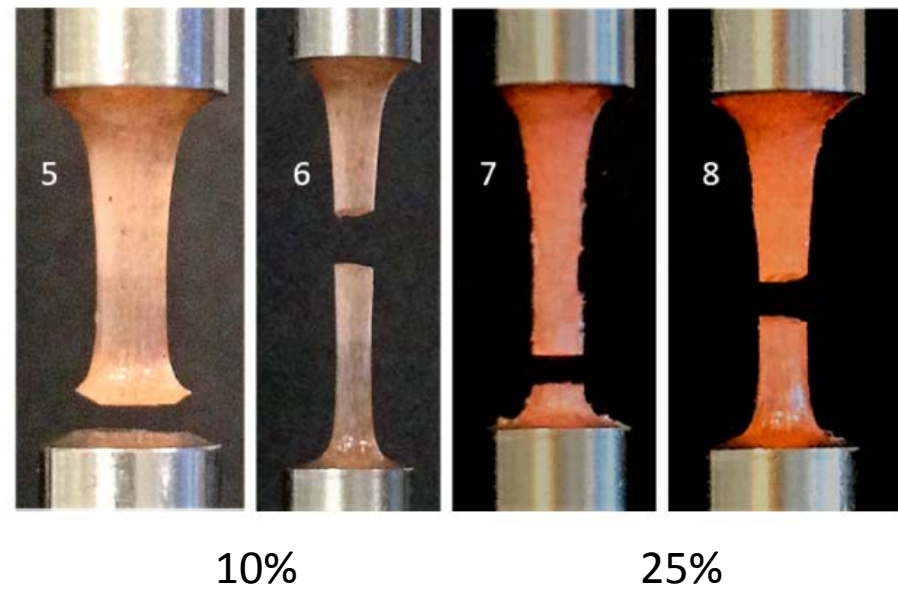
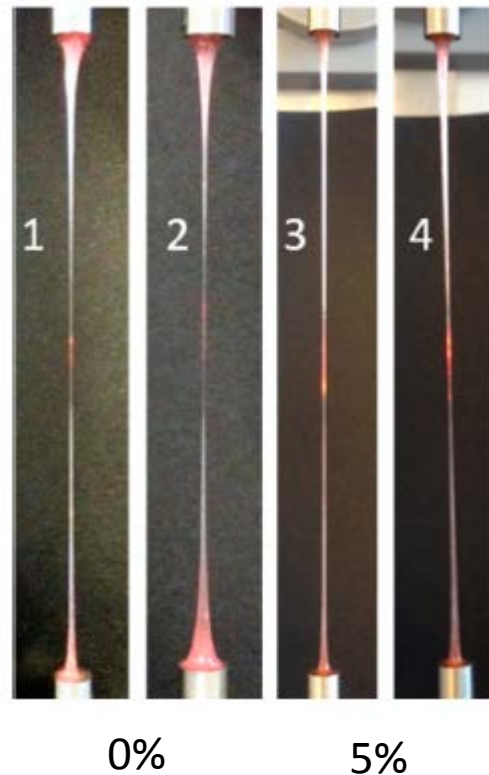
Associating polymers with hydrogen bonding



Scherz and Costanzo et al., *Macromolecules* 2017, 50, 5176–5187

Associating polymers

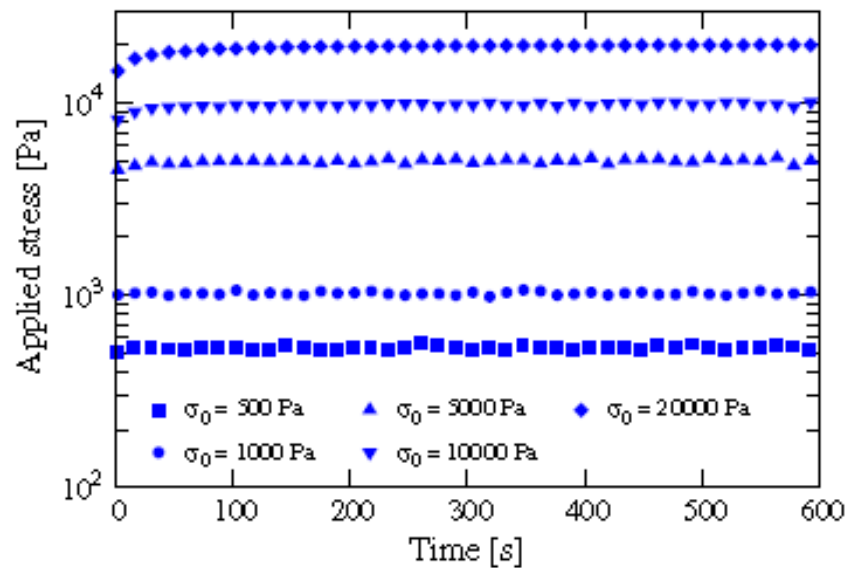
Associating polymers with hydrogen bonding



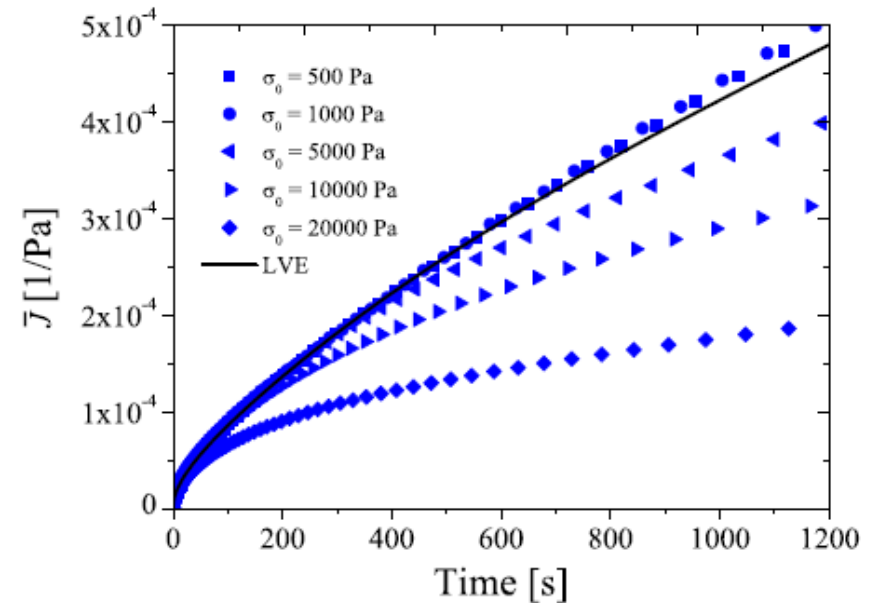
Scherz and Costanzo et al., Macromolecules 2017, 50, 5176–5187

Creep measurements

Applied



Measured

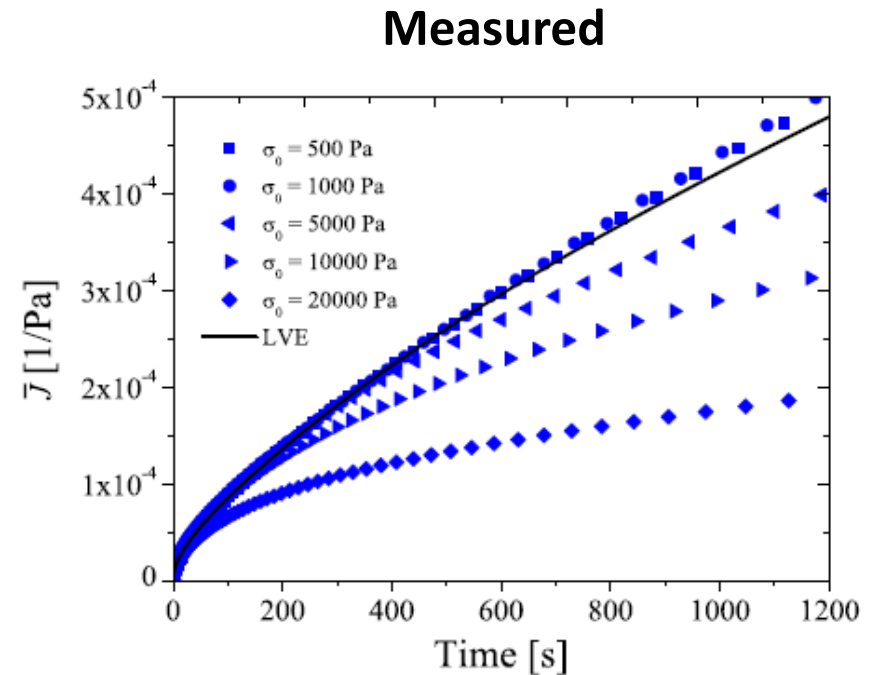
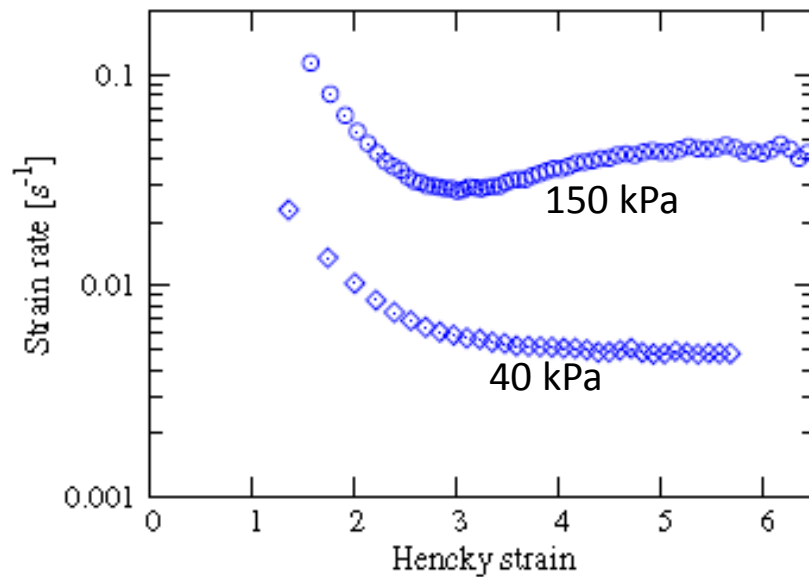


Extensional compliance

$$\bar{J}(t) = \epsilon(t)/\sigma_0$$

Figures: Alvarez *et al.*, PRL 110, 168301 (2013)

Creep measurements

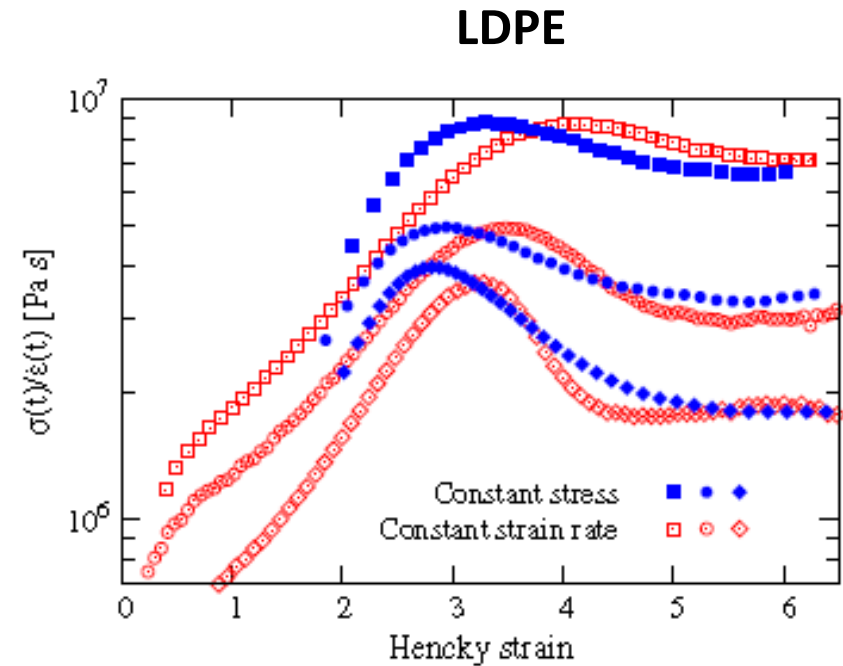
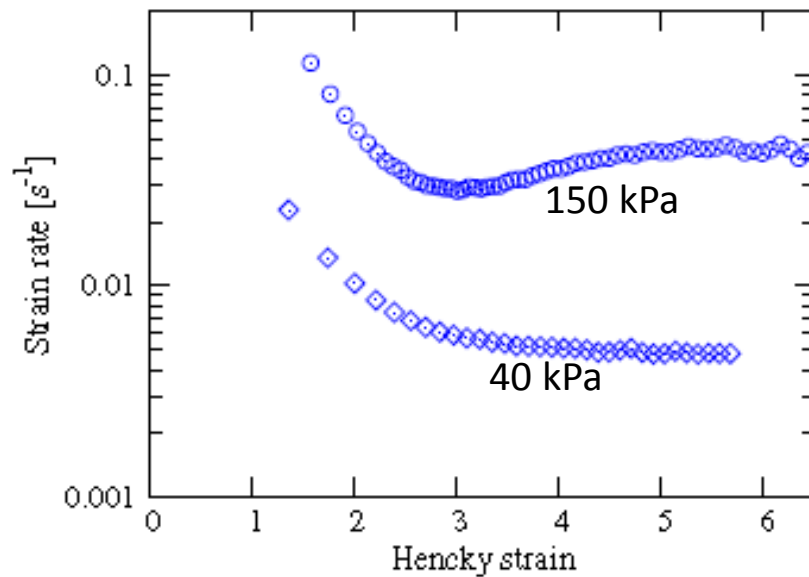


Extensinal compliance

$$\bar{J}(t) = \epsilon(t)/\sigma_0$$

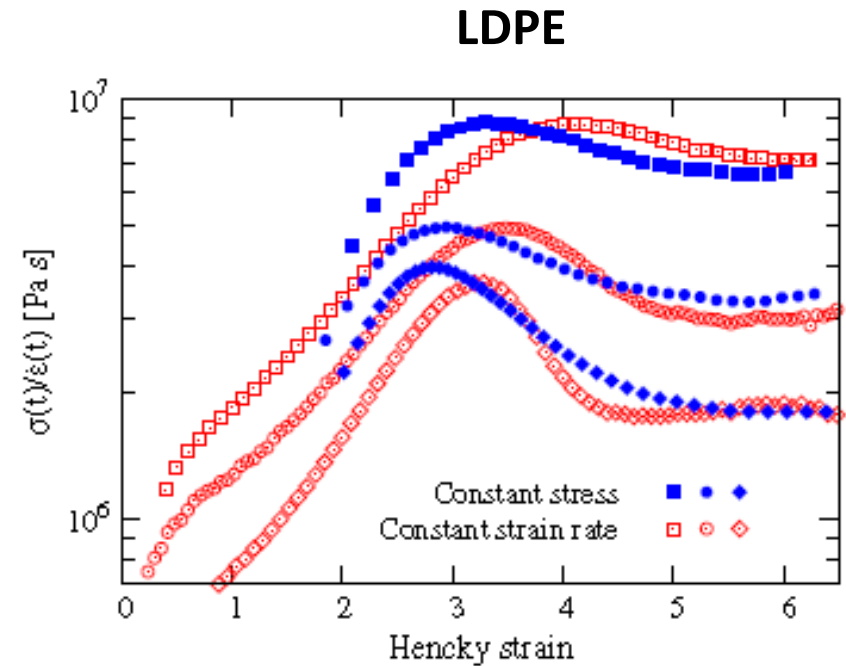
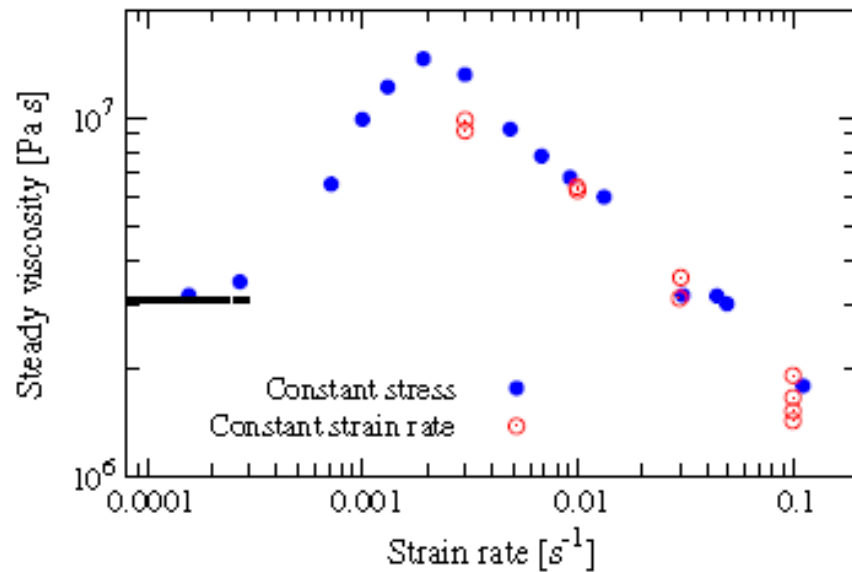
Figures: Alvarez *et al.*, *PRL* 110, 168301 (2013)

Creep measurements



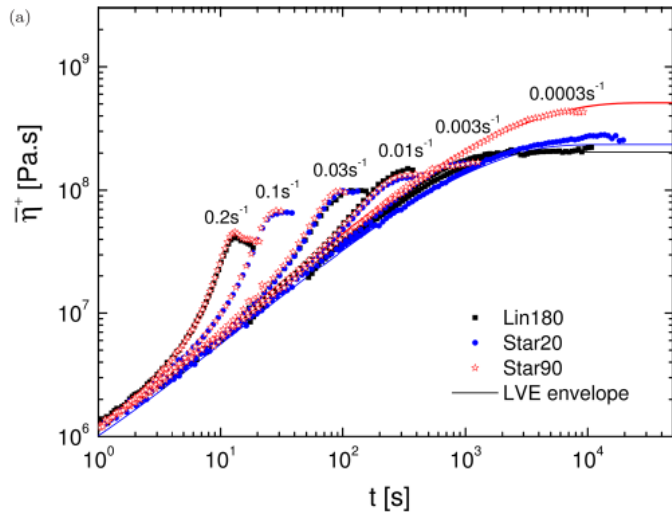
Figures: Alvarez *et al.*, *PRL* 110, 168301 (2013)

Creep measurements

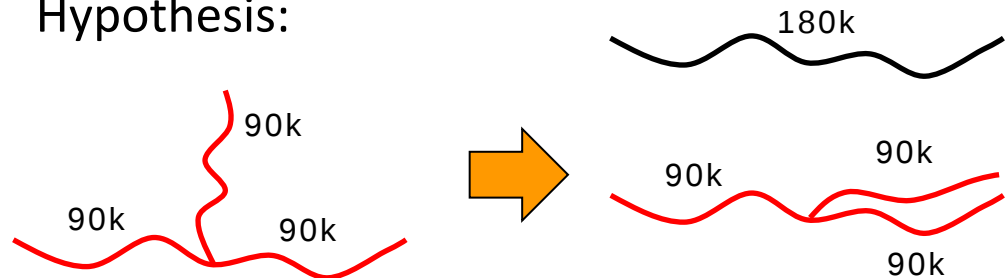


Figures: Alvarez *et al.*, *PRL* 110, 168301 (2013)

Combining with SANS

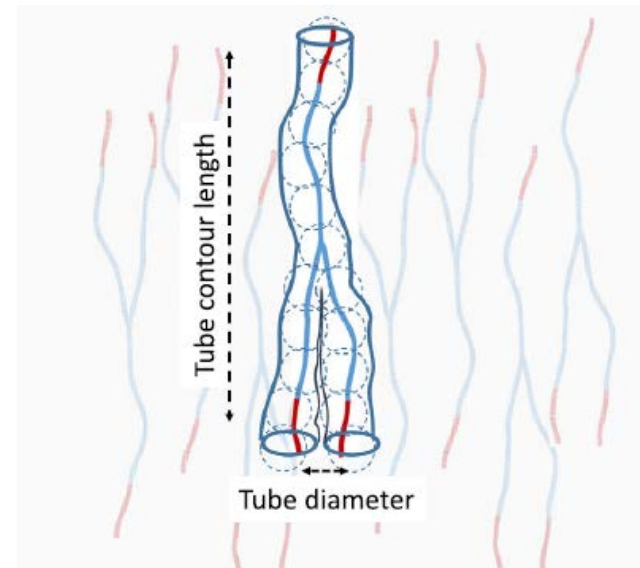
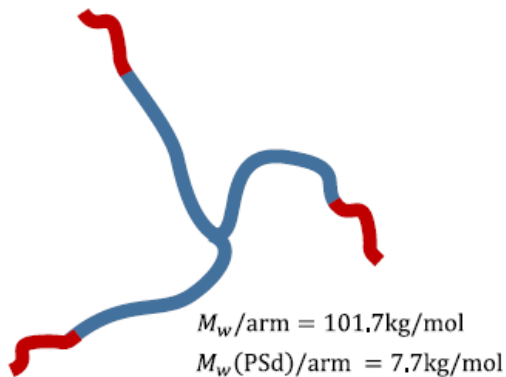


Hypothesis:

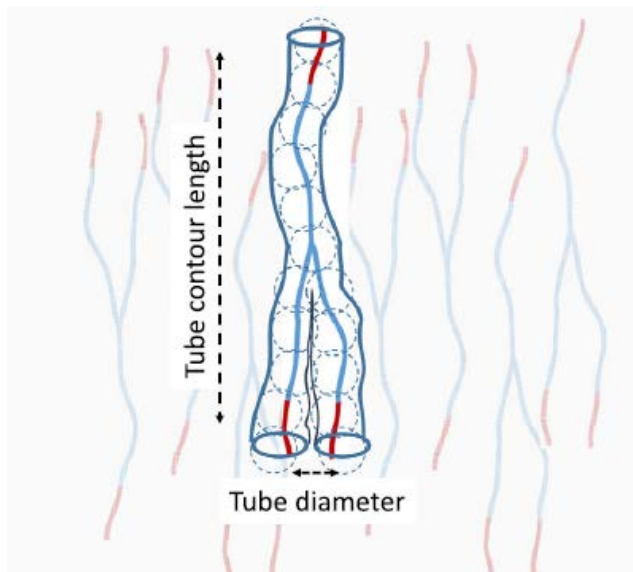


Extensional rheology cannot give direct information of molecular conformations

3-Arm Polystyrene Star-Polymer
Each arm has end-block of deuterated monomers

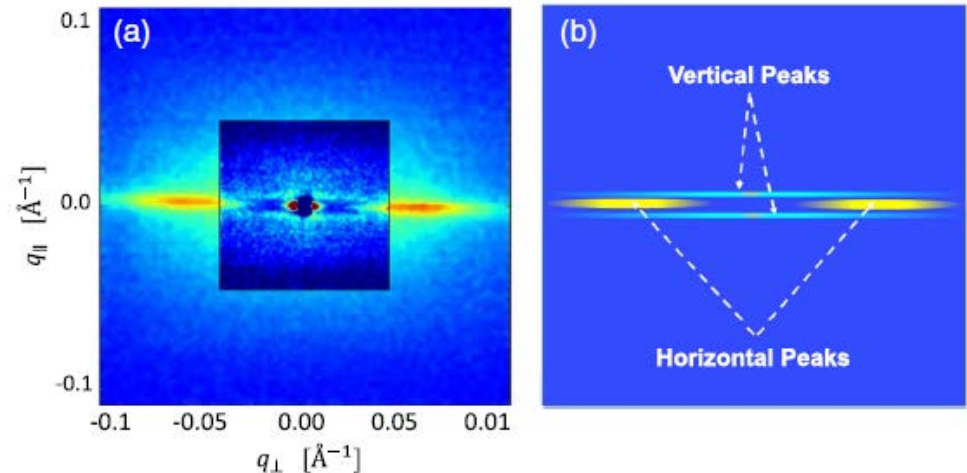
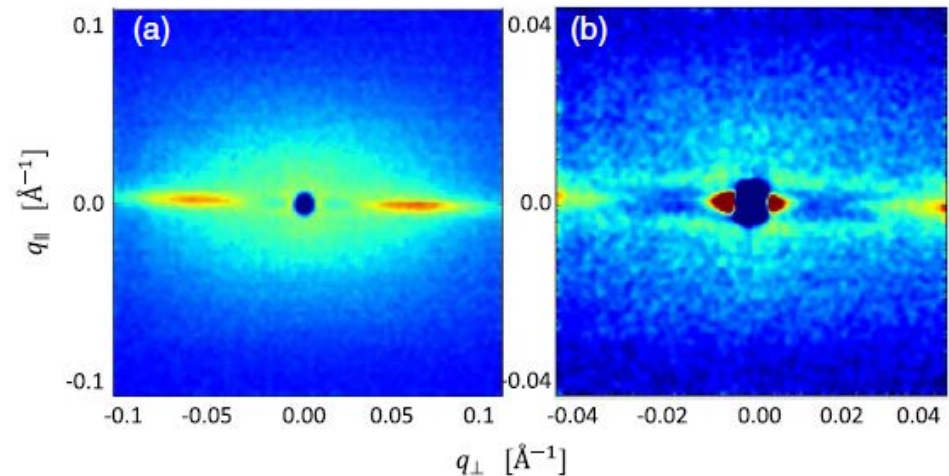


Combining with SANS



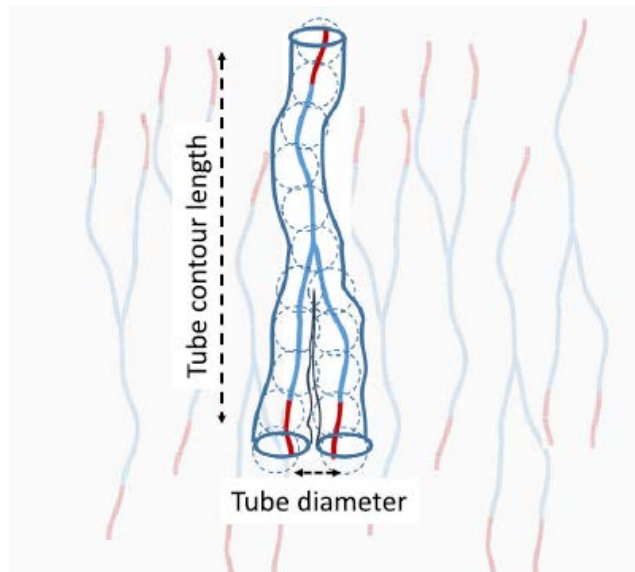
The distance resulting from the horizontal peaks is roughly **100 Å**

Tube diameter for PS melt: **85 Å**

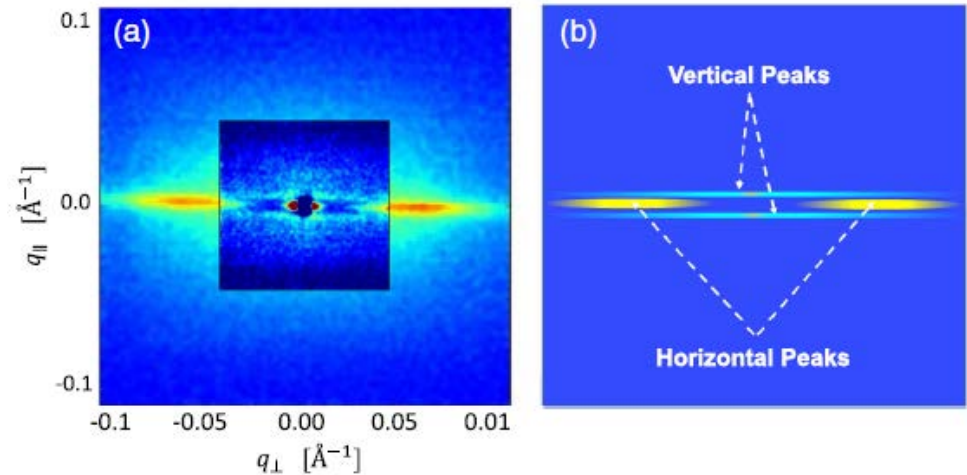
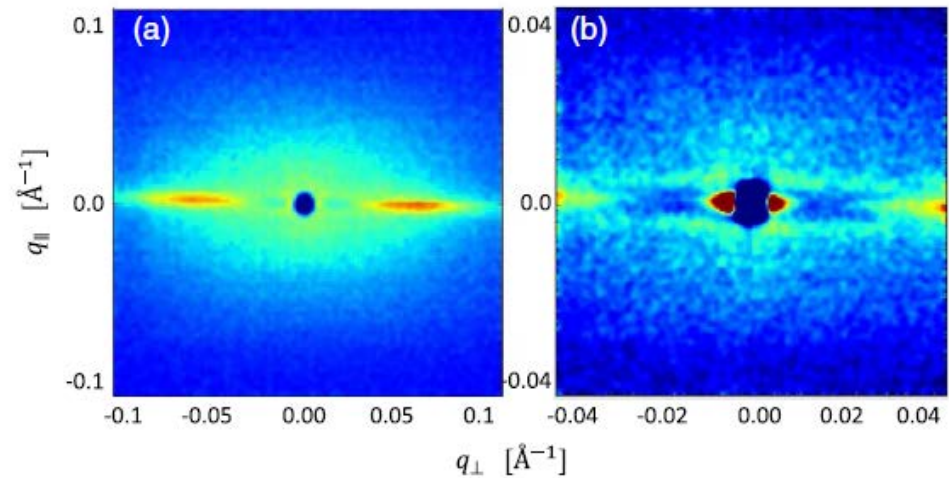


Mortensen et al., PRL 120, 207801 (2018)

Combining with SANS



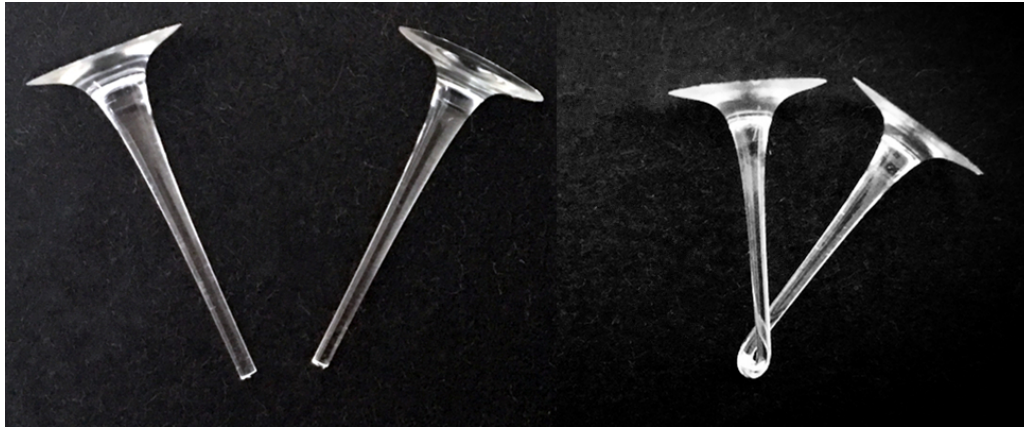
Still need to compare with the corresponding linear chain...



Mortensen et al., PRL 120, 207801 (2018)

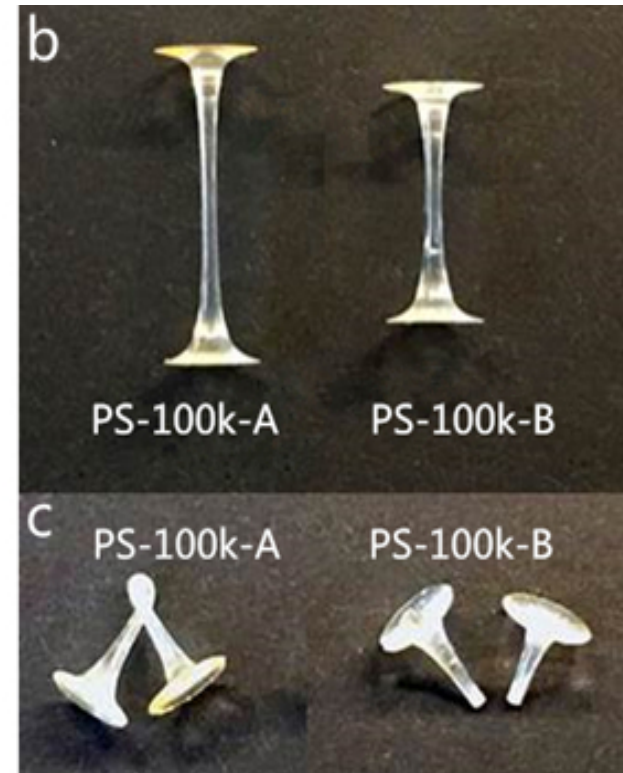
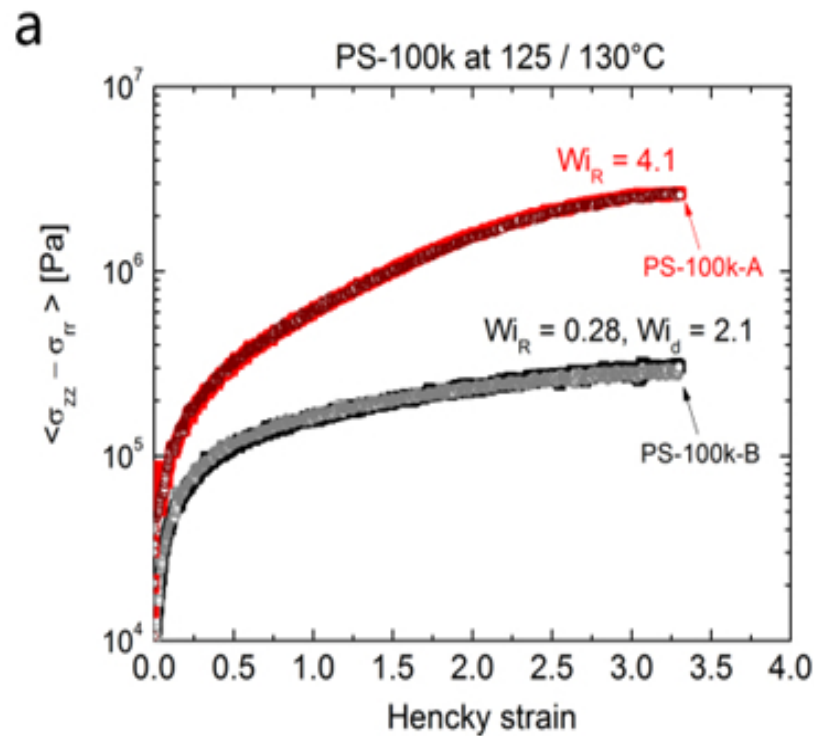
Influence on mechanical properties

Is polystyrene always brittle at room temperature?



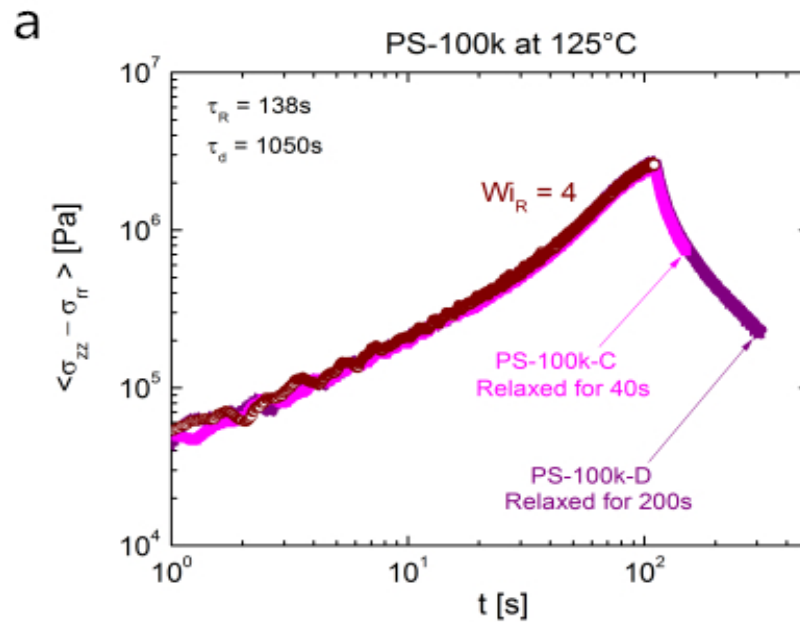
How to make flexible PS by just changing processing conditions?

Influence on mechanical properties



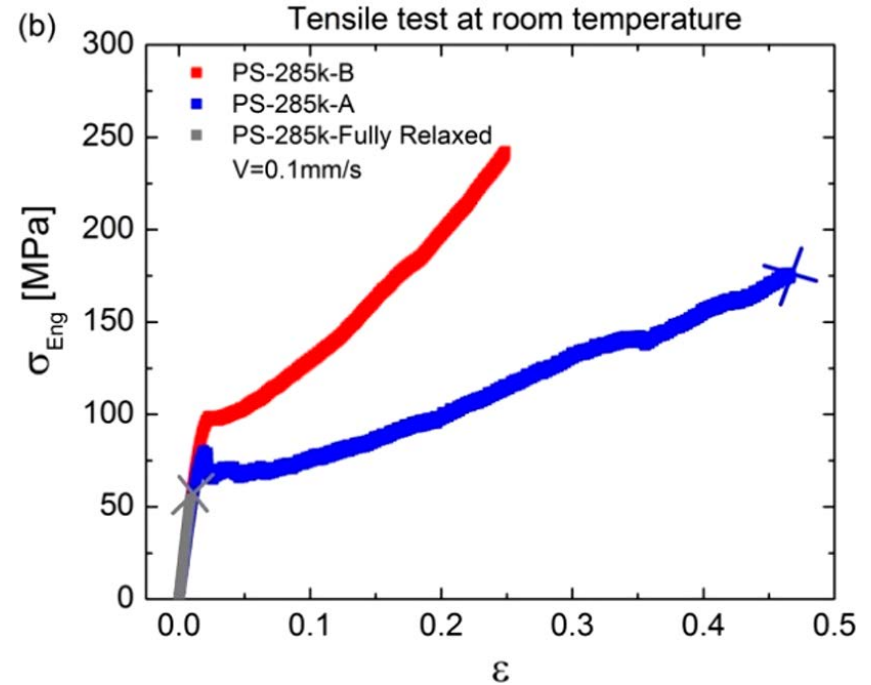
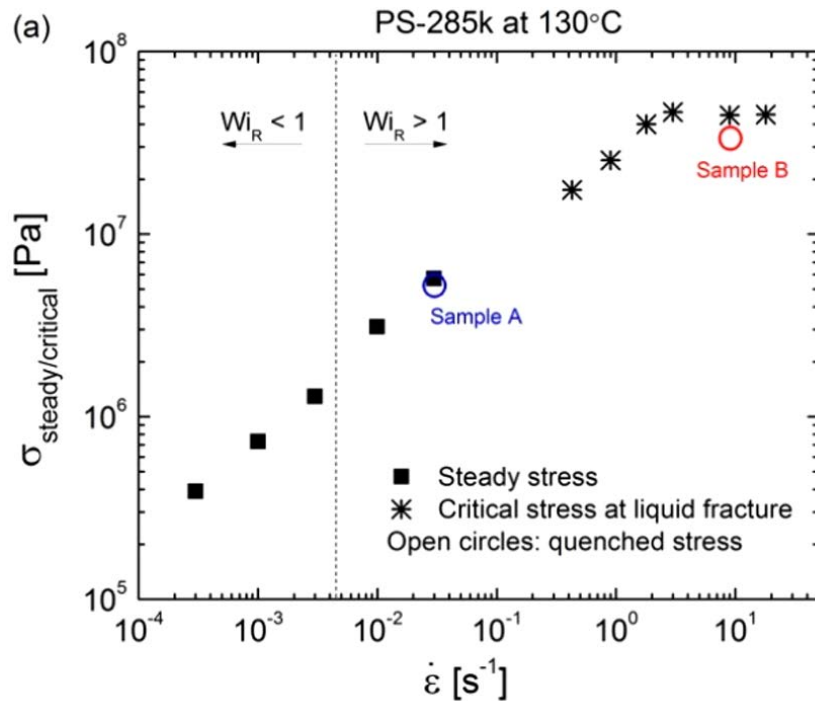
Huang et al., ACS Macro Letters 7, 1126–1130 (2018)

Influence on mechanical properties



Huang et al., ACS Macro Letters 7, 1126–1130 (2018)

Influence on mechanical properties



Huang et al., ACS Macro Letters 7, 1126–1130 (2018)

Thank you

