

Nonlinear shear rheology of metallo-supramolecular polymer networks

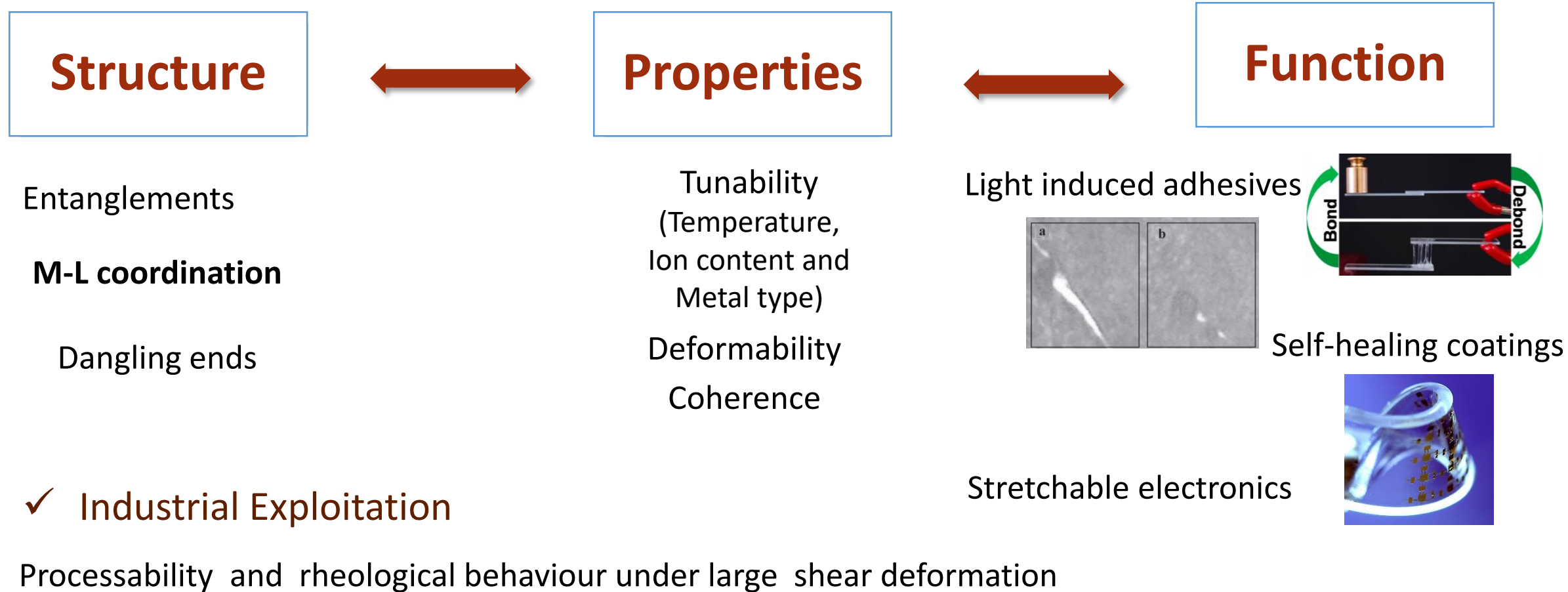
Christina Pyromali^{1,2*}, Yanzhao Li³, Flanco Zhuge³, Charles-André Fustin³, Evelyne van Ruymbeke³ and Dimitris Vlassopoulos^{1,2}

DoDyNet Montpellier Meeting

21-10-2021



Motivation



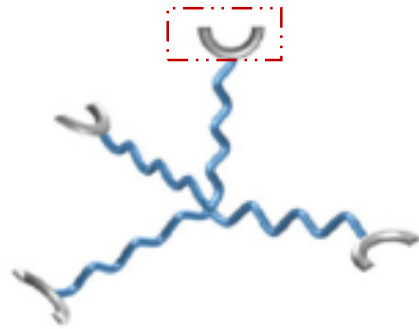
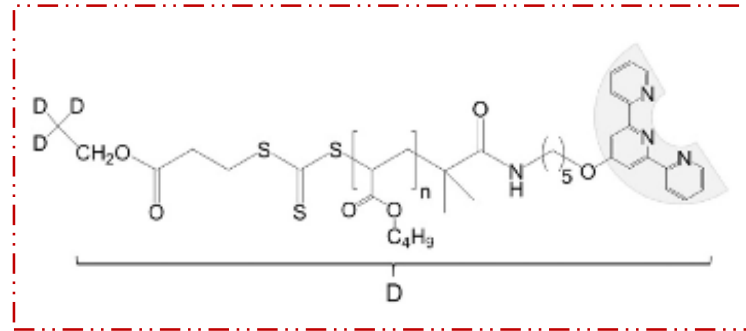
Heinzmann et al., *ACS Appl. Mater. Interfaces* **6**, 4713–4719 (2014).

Bode, S. et al. *Advanced Materials* **25**, 1634–1638 (2013).

Rao, Y.-L. et al. *J. Am. Chem. Soc.* **138**, 6020–6027 (2016).

Materials

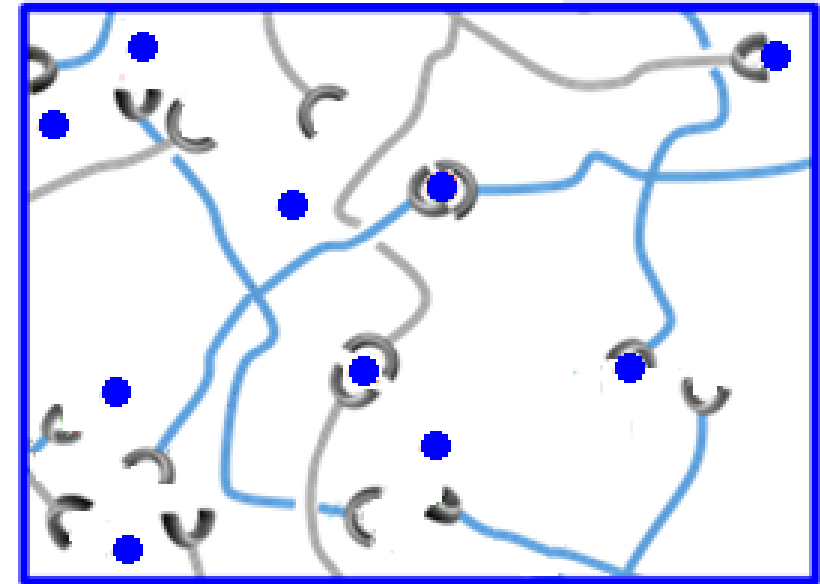
Terpyridine ligand



+



Zn^{2+}



Same Charge

Similar ionic radius

Thermodynamic stability

M-L cross-linking stability

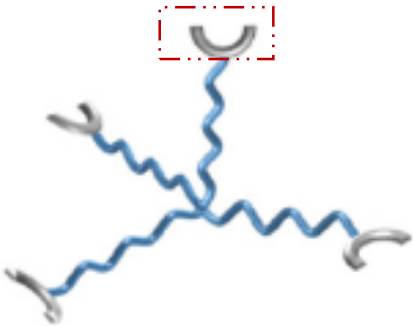
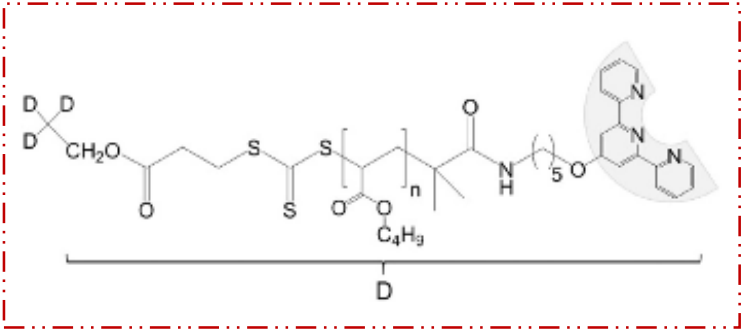
4 arm star poly(n-butyl acrylate)s

M_n [kg/mol]	M_e [kg/mol]	$Z=M_n/M_e$	Dispersity	T_g [°C]
249	18	3.4 per arm	1.2	-52

	K_1 [M ⁻¹]	$k_{a,1}$ [M ⁻¹ s ⁻¹]	$k_{d,1}$ [s ⁻¹]	K_2 [M ⁻¹]	$k_{a,2}$ [M ⁻¹ s ⁻¹]	$k_{d,2}$ [s ⁻¹]	$\beta=K_1 \cdot K_2$
Zn(II)	10^6	$10^{6.1}$	$10^{0.1}$	$10^{5.2}$	-	-	$10^{11.2}$
Cu (II)	10^{12}	$10^{7.3}$	-	10^6	-	-	10^{18}
Co (II)	$10^{8.4}$	$10^{4.4}$	10^{-4}	$10^{9.9}$	$10^{6.7}$	$10^{-3.2}$	$10^{18.3}$

Materials

Terpyridine ligand



+



Cu^{2+}



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Thermodynamic stability

M-L cross-linking stability

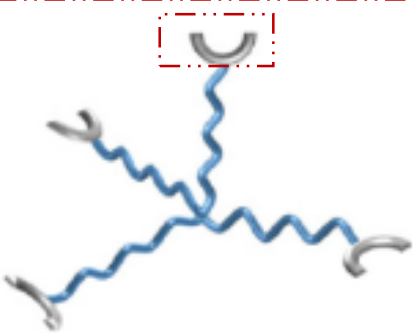
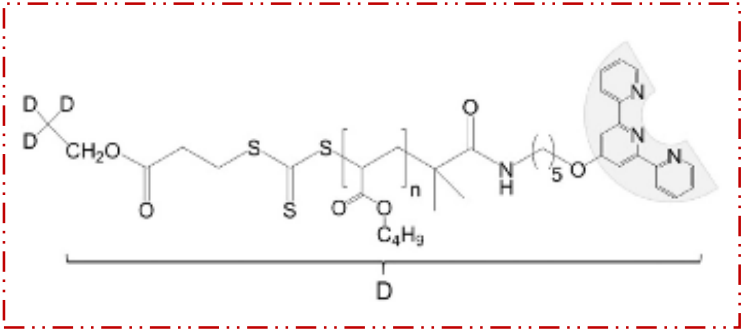
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Zn(II)	10 ⁶	10 ^{6.1}	10 ^{0.1}	10 ^{5.2}	-	-	10 ^{11.2}
Cu (II)	10 ¹²	10 ^{7.3}	-	10 ⁶	-	-	10 ¹⁸
Co (II)	10 ^{8.4}	10 ^{4.4}	10 ⁻⁴	10 ^{9.9}	10 ^{6.7}	10 ^{-3.2}	10 ^{18.3}

Materials

Terpyridine ligand



+



Co²⁺



M-L cross-linking stability

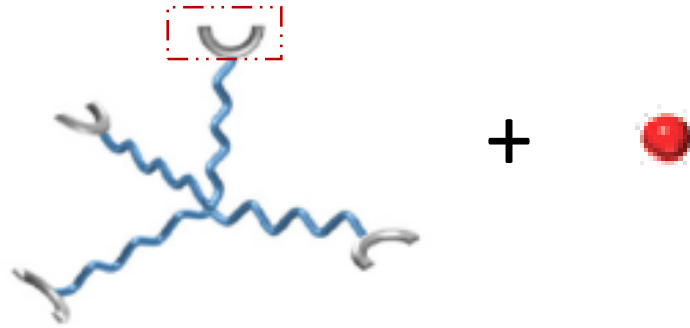
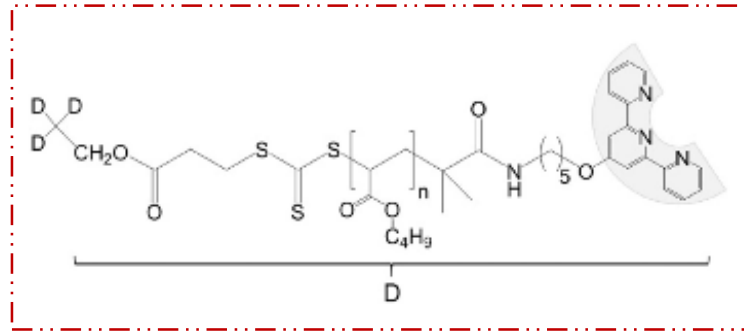
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M _n [kg/mol]	M _e [kg/mol]	Z=M _n /M _e	Dispersity	T _g [°C]
249	18	3.4 per arm	1.2	-52

	K ₁ [M ⁻¹]	k _{a,1} [M ⁻¹ s ⁻¹]	k _{d,1} [s ⁻¹]	K ₂ [M ⁻¹]	k _{a,2} [M ⁻¹ s ⁻¹]	k _{d,2} [s ⁻¹]	β=K ₁ ·K ₂
Zn(II)	10 ⁶	10 ^{6.1}	10 ^{0.1}	10 ^{5.2}	-	-	10 ^{11.2}
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Materials

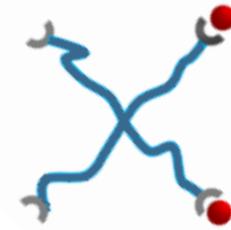
Terpyridine ligand



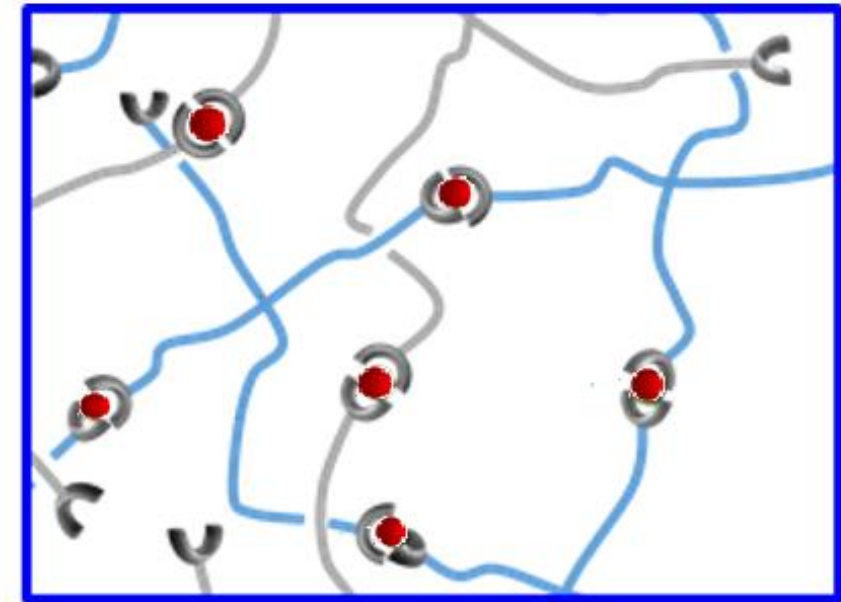
4 arm star poly(n-butyl acrylate)s

Single Network

0.5eq.



Stoichiometric amount

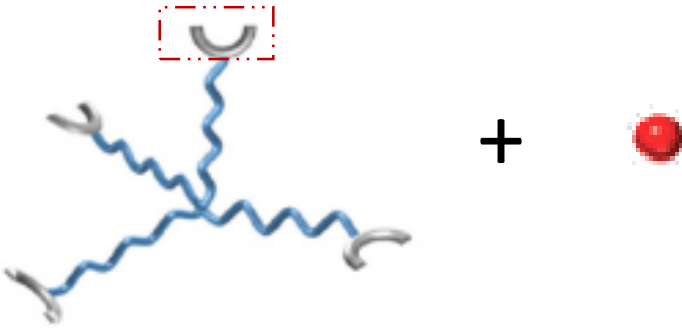
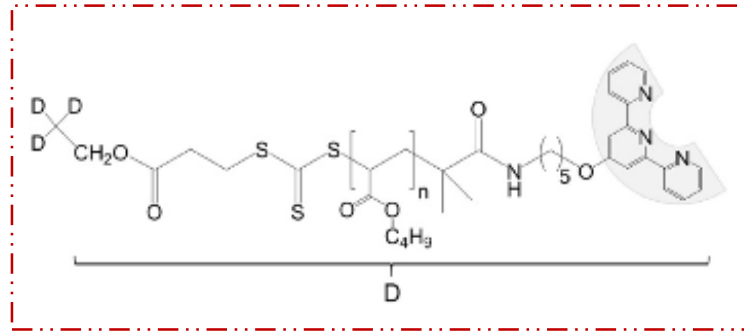


M-L cross-linking density

M_n [kg/mol]	M_e [kg/mol]	$Z=M_n/M_e$	Dispersity	T_g [°C]
249	18	3.4 per arm	1.2	-52

Materials

Terpyridine ligand

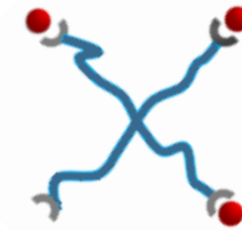


4 arm star poly(n-butyl acrylate)s

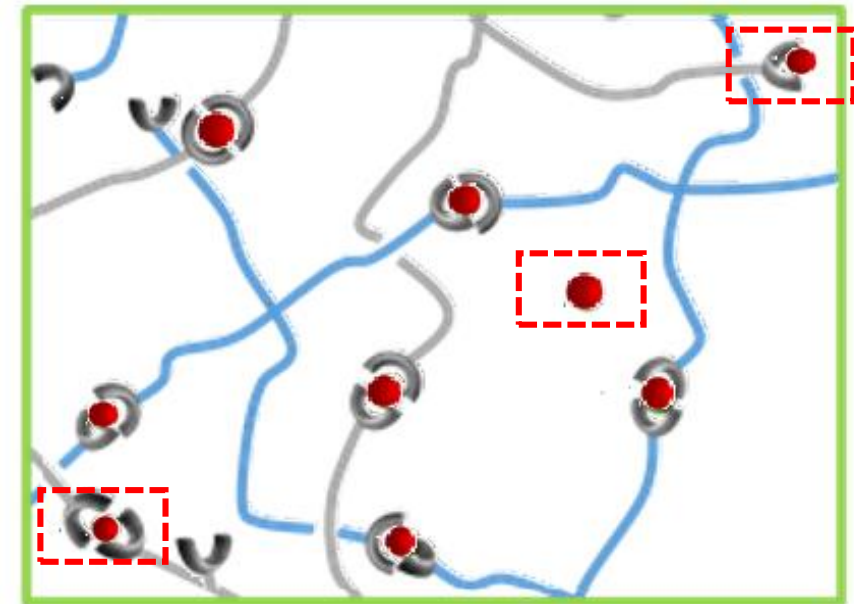
M_n [kg/mol]	M_e [kg/mol]	$Z=M_n/M_e$	Dispersity	T_g [°C]
249	18	3.4 per arm	1.2	-52

Single Network

0.75eq.



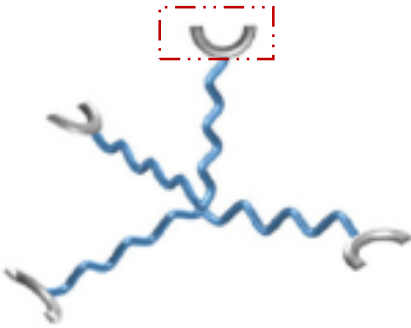
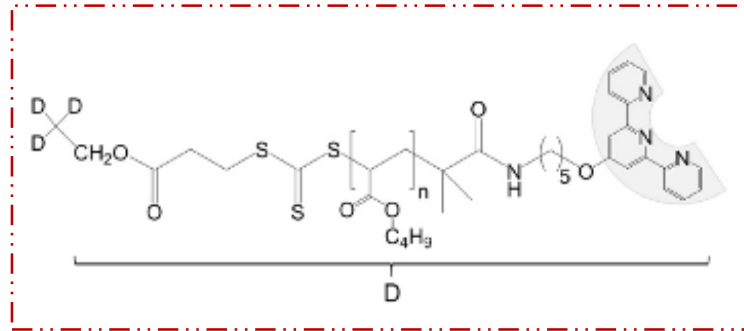
Excess



M-L cross-linking density

Materials

Terpyridine ligand



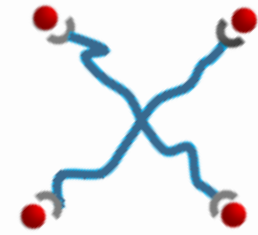
+



4 arm star poly(n-butyl acrylate)s

Single Network

1.00eq.



Excess

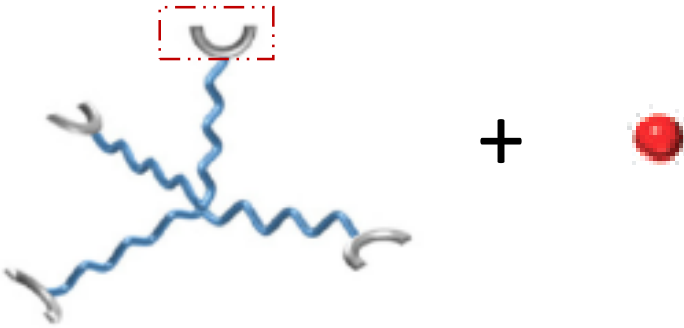
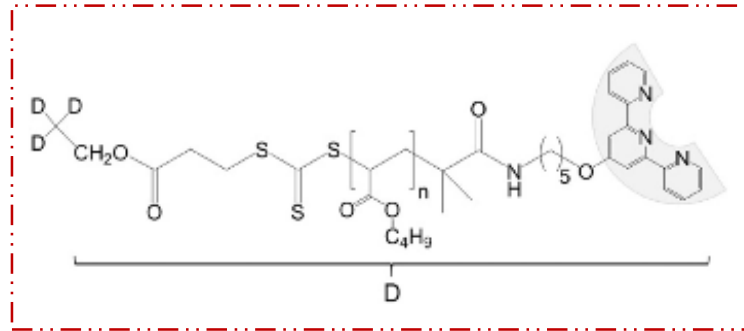


M-L cross-linking density

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Materials

Terpyridine ligand

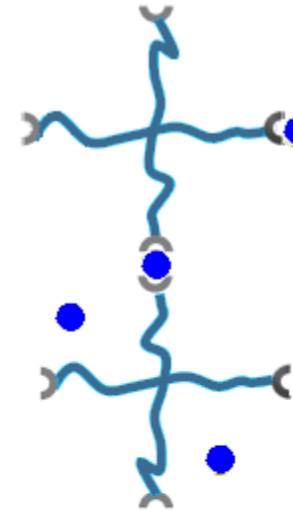


4 arm star poly(n-butyl acrylate)s

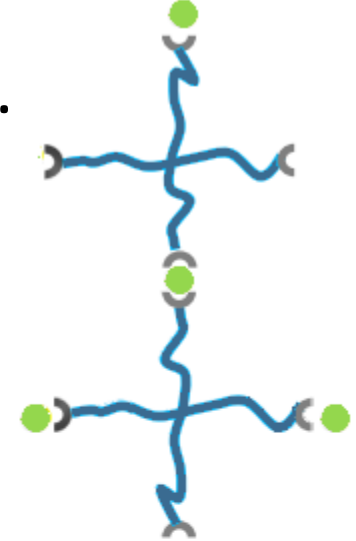
M_n [kg/mol]	M_e [kg/mol]	$Z=M_n/M_e$	Dispersity	T_g [°C]
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Blends

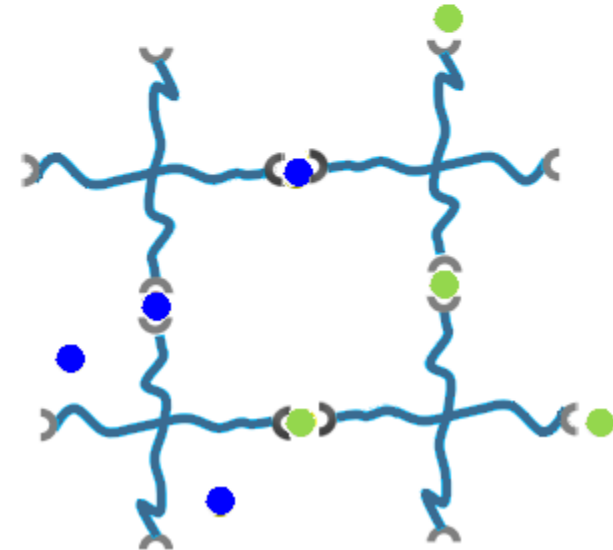
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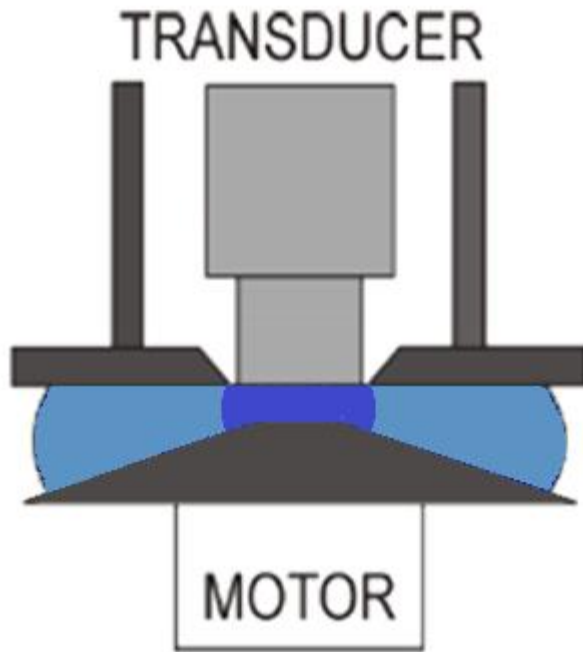
0.50eq.



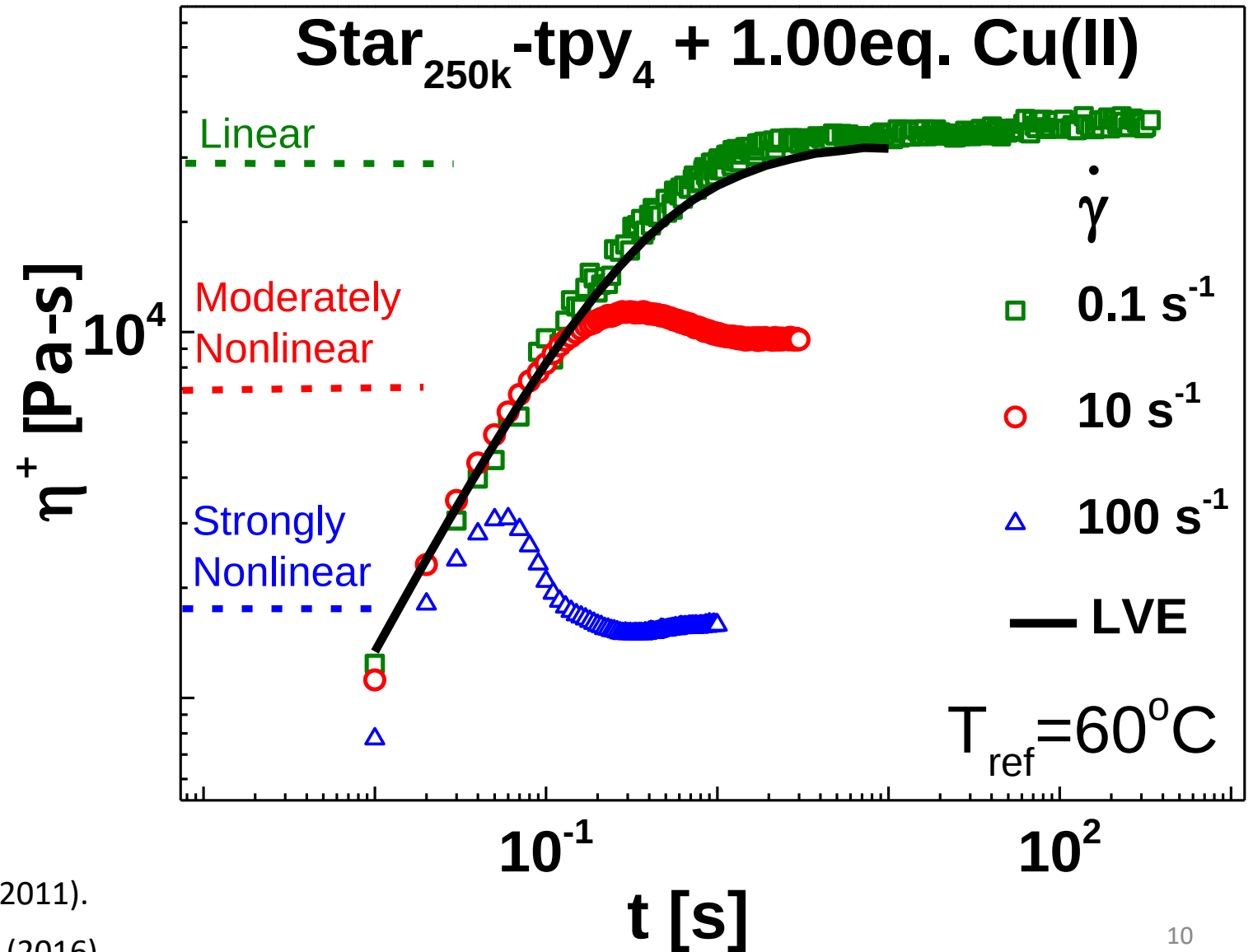
0.50eq.



Methods



Edge fracture mitigation delay



Snijkers, F. & Vlassopoulos, D., *JoR* **55**, 1167–1186 (2011).

Costanzo, S. *et al.*, *Macromolecules* **49**, 3925–3935 (2016).

Methods

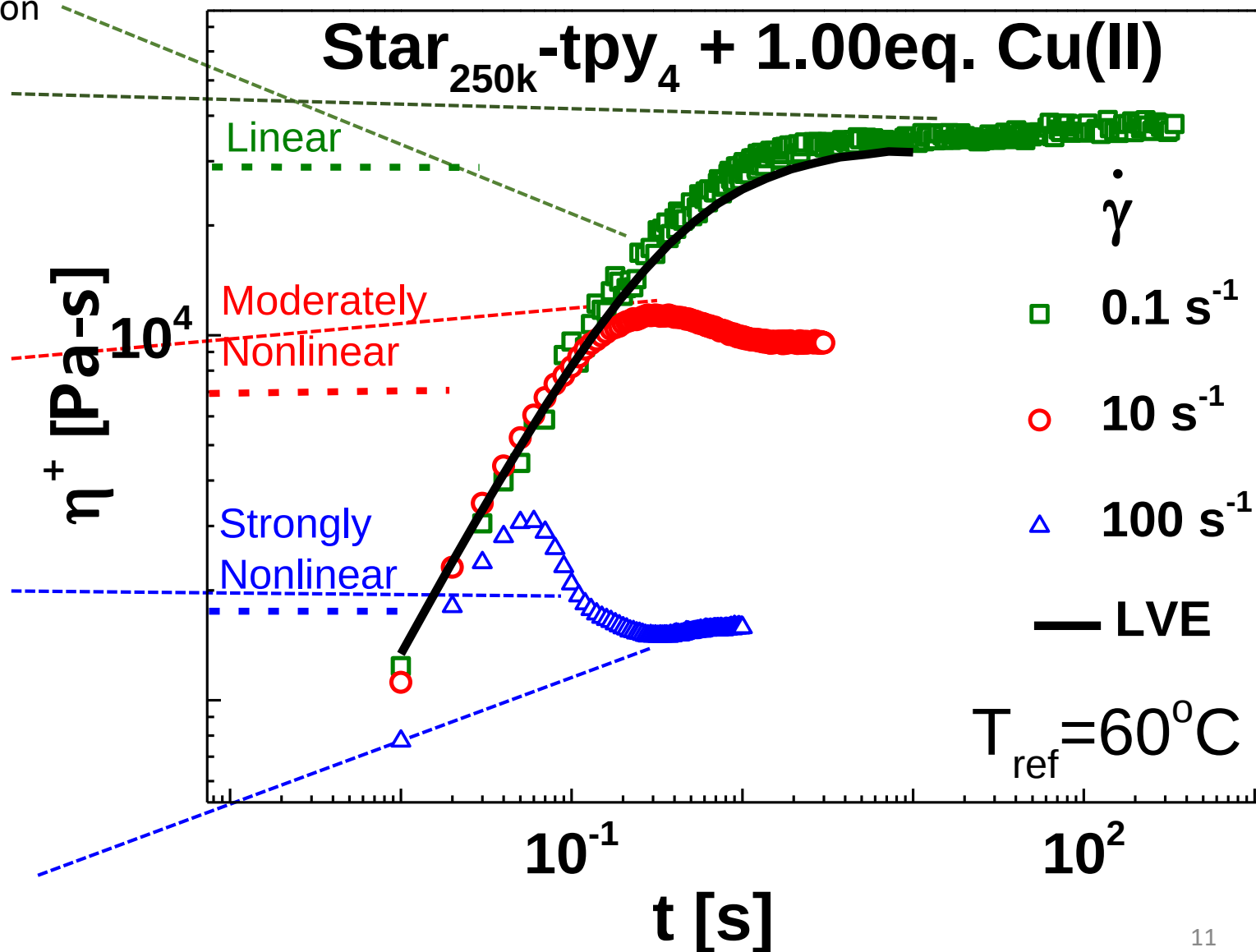
Linear response: elastic deformation region

Structure has time to rearrange and the deformation is dissipative

Overshoot: collective orientation and stretching/ dynamic bond breakage

Flow-induced disentanglement
Bond rearrangement

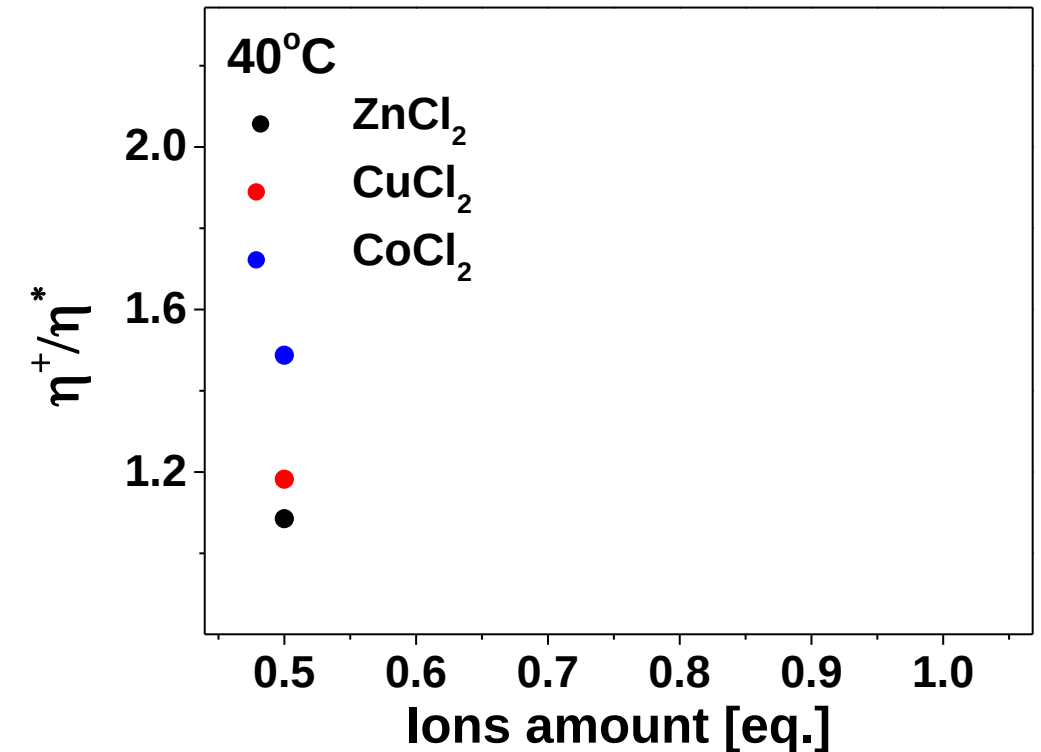
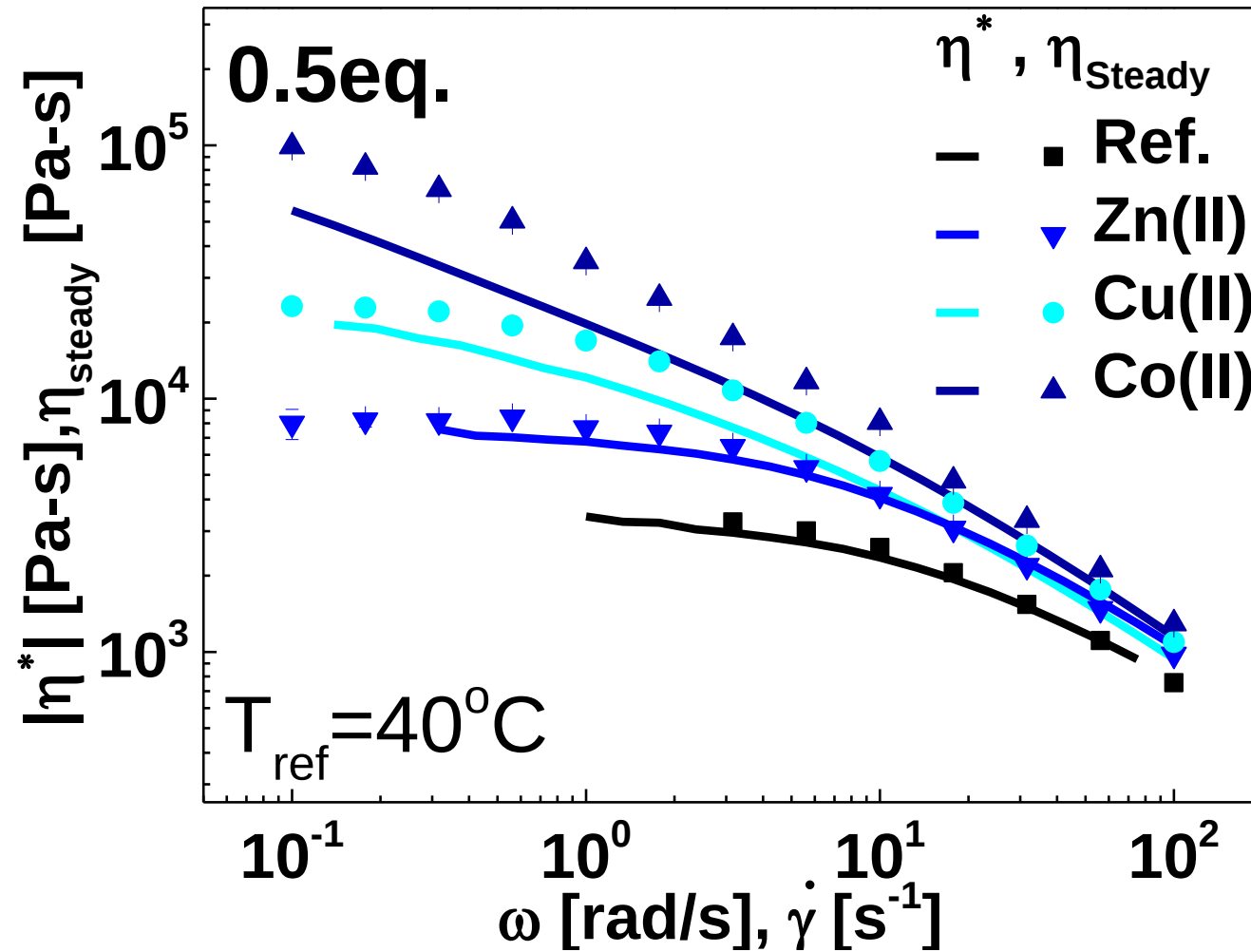
Steady state region: macroscopic flow/
dissipative deformation



Cox-Merz Rule Fails

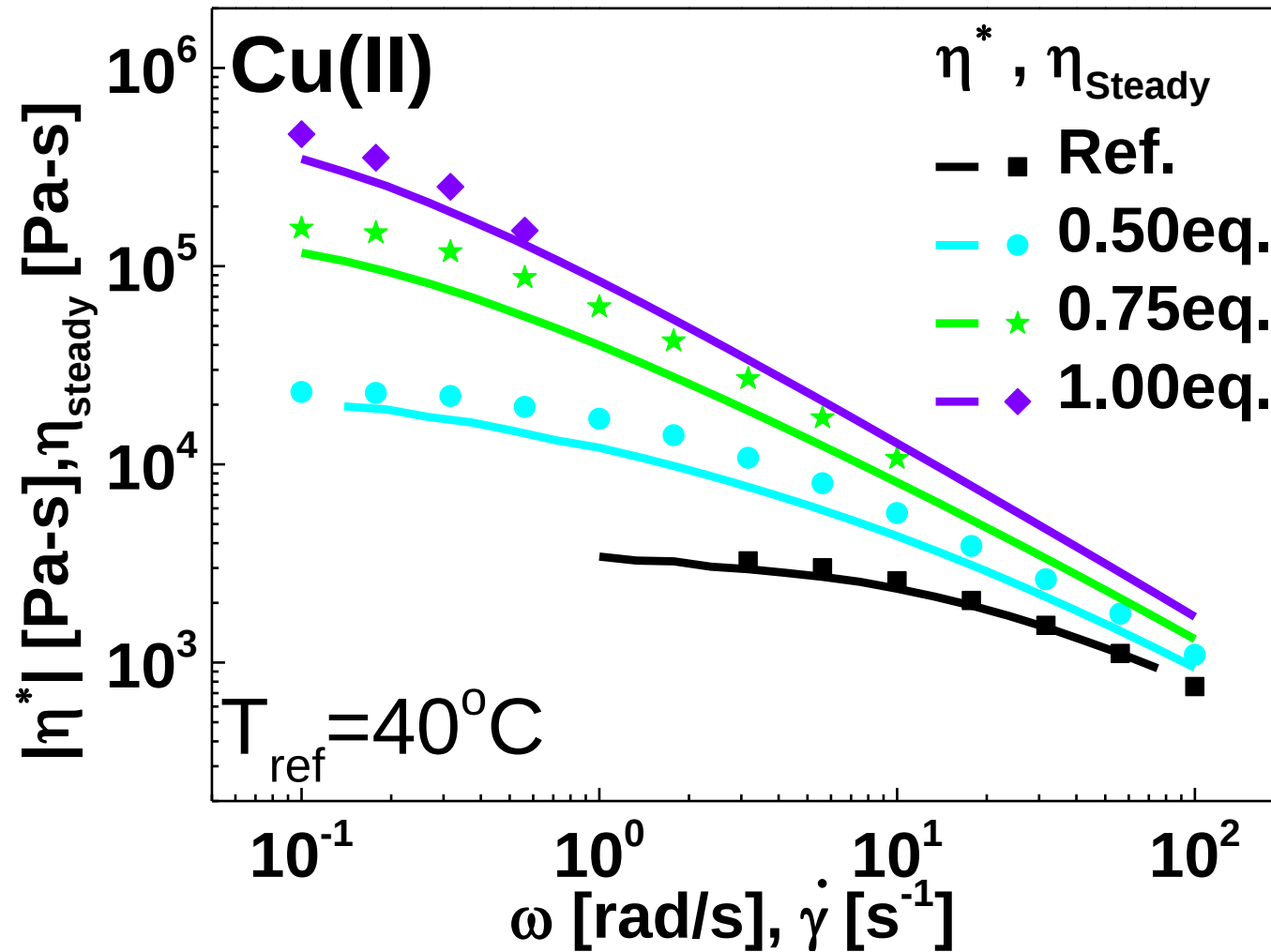
Ion Strength
Ion Content
Temperature

$$\dot{\gamma} = 10s^{-1}, \omega = 10rad/s$$

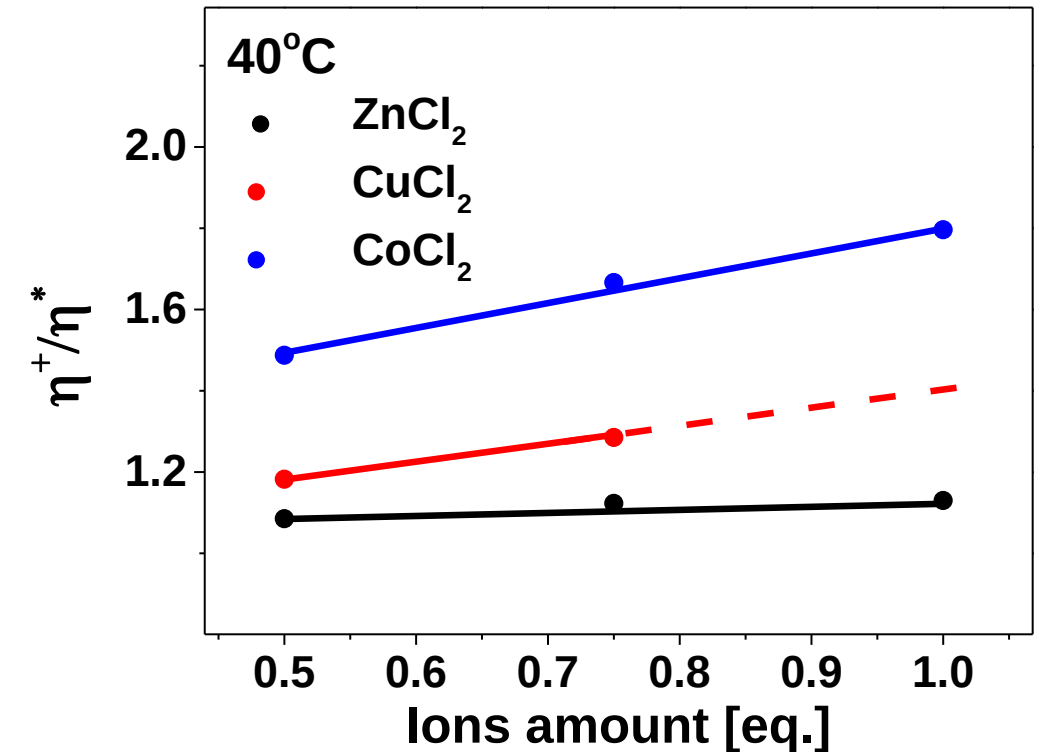


Cox-Merz Rule Fails

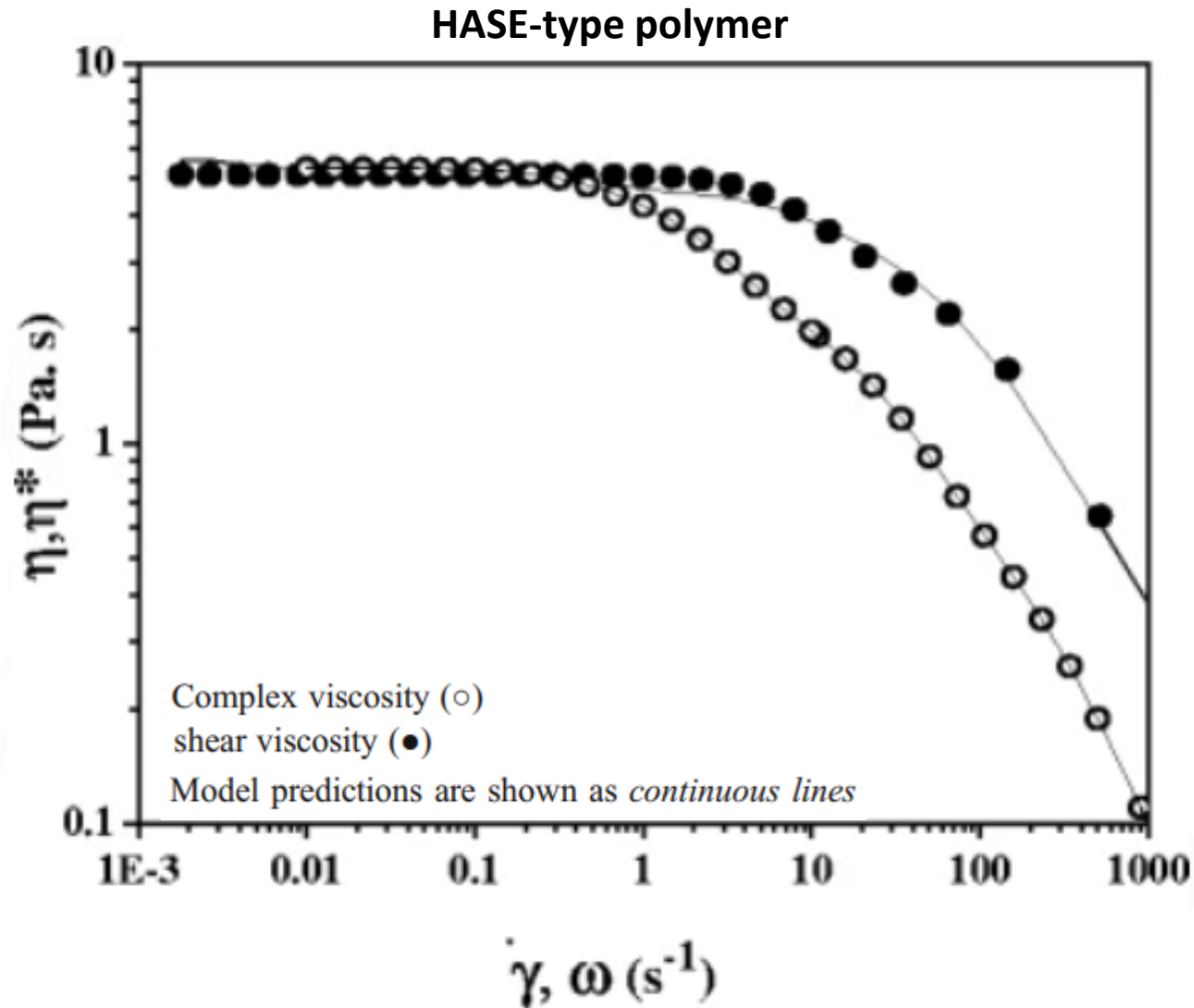
Ion Strength
Ion Content
Temperature



$$\dot{\gamma} = 10\text{s}^{-1}, \omega = 10\text{rad/s}$$



Origin of Cox-Merz Rule Failure



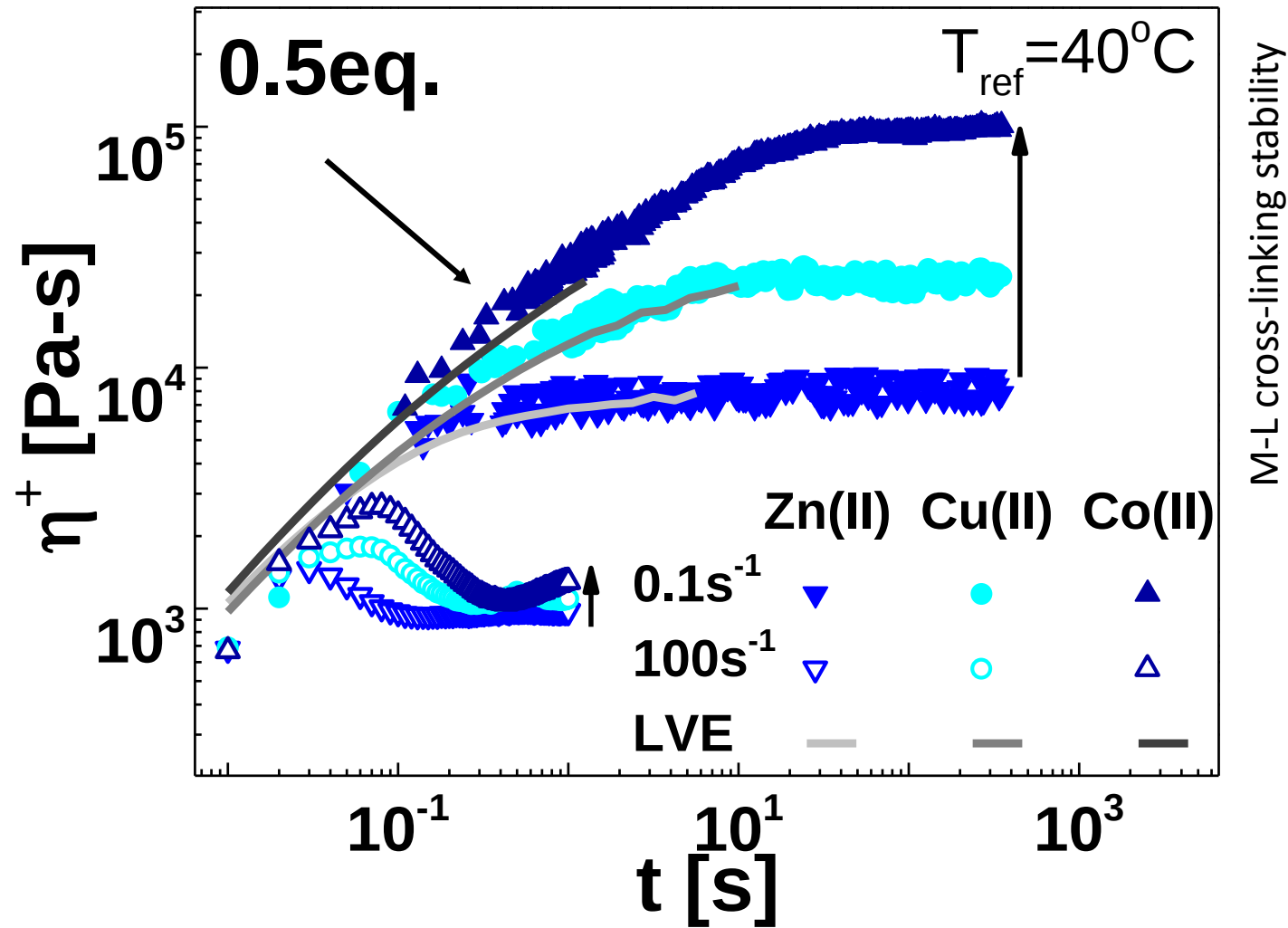
- Associative polymers do not obey Cox-Merz rule
- Viscosity ratio is typically higher than unity
- Multi-mode model with a kinetic balance that destruction (convective) of long-range associations prevails over formation (diffusive) of short-range associations.

Caram, Y., et al., *Rheol Acta* **46**, 45–57 (2006)

Pellens, L., et al., *JoR* **48**, 379–393 (2004) ¹⁴

Shear stress growth coefficient

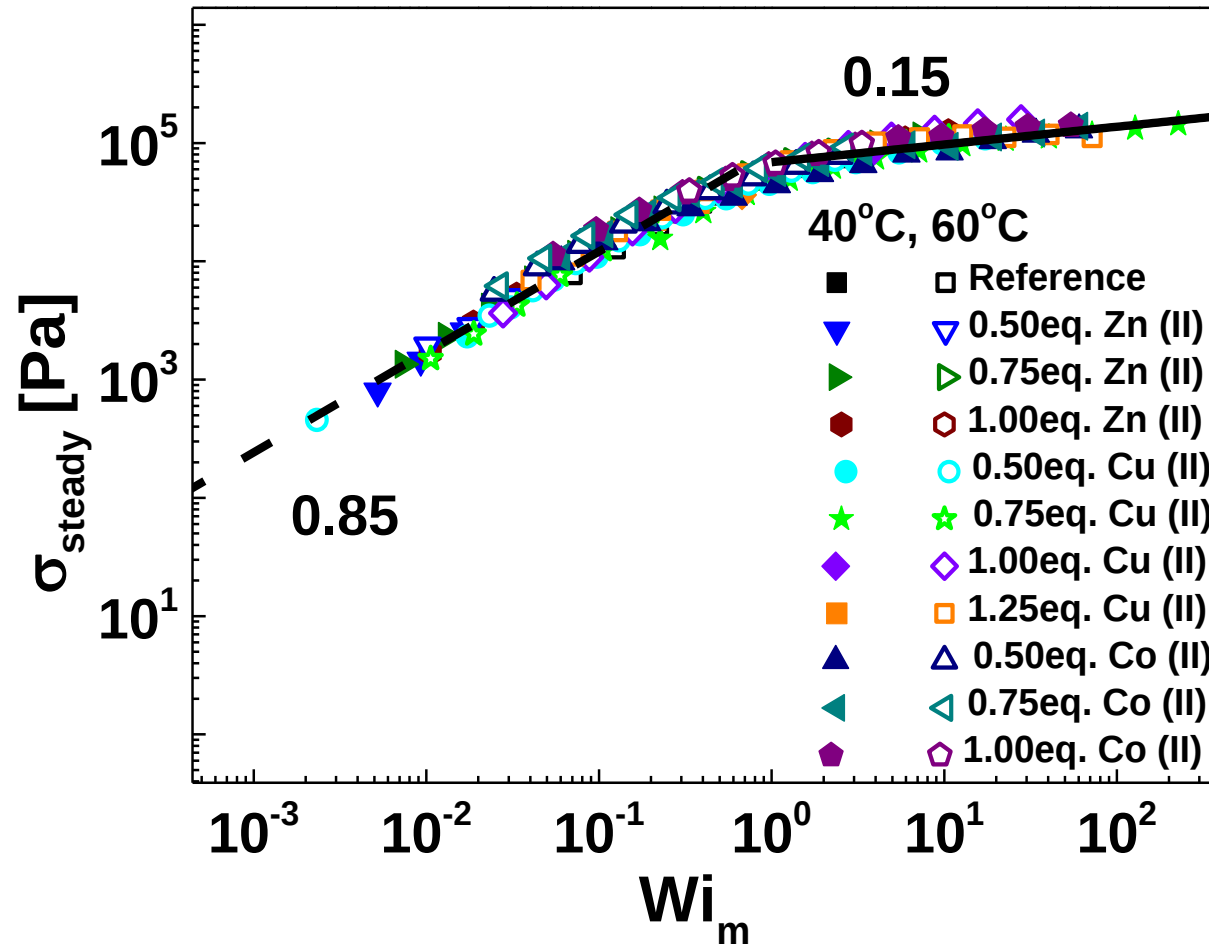
Ion Strength
Ion Content
Temperature



- Steady-state viscosity increase with M-L crosslinking stability
- Pronounced overshoot for stable ion
- Shear strain hardening region is related to slow disassociation kinetics that result strong resistance to deformation
- Supramolecular dynamics dominate the response lower deformation rates (or longer times)

Steady-state stresses

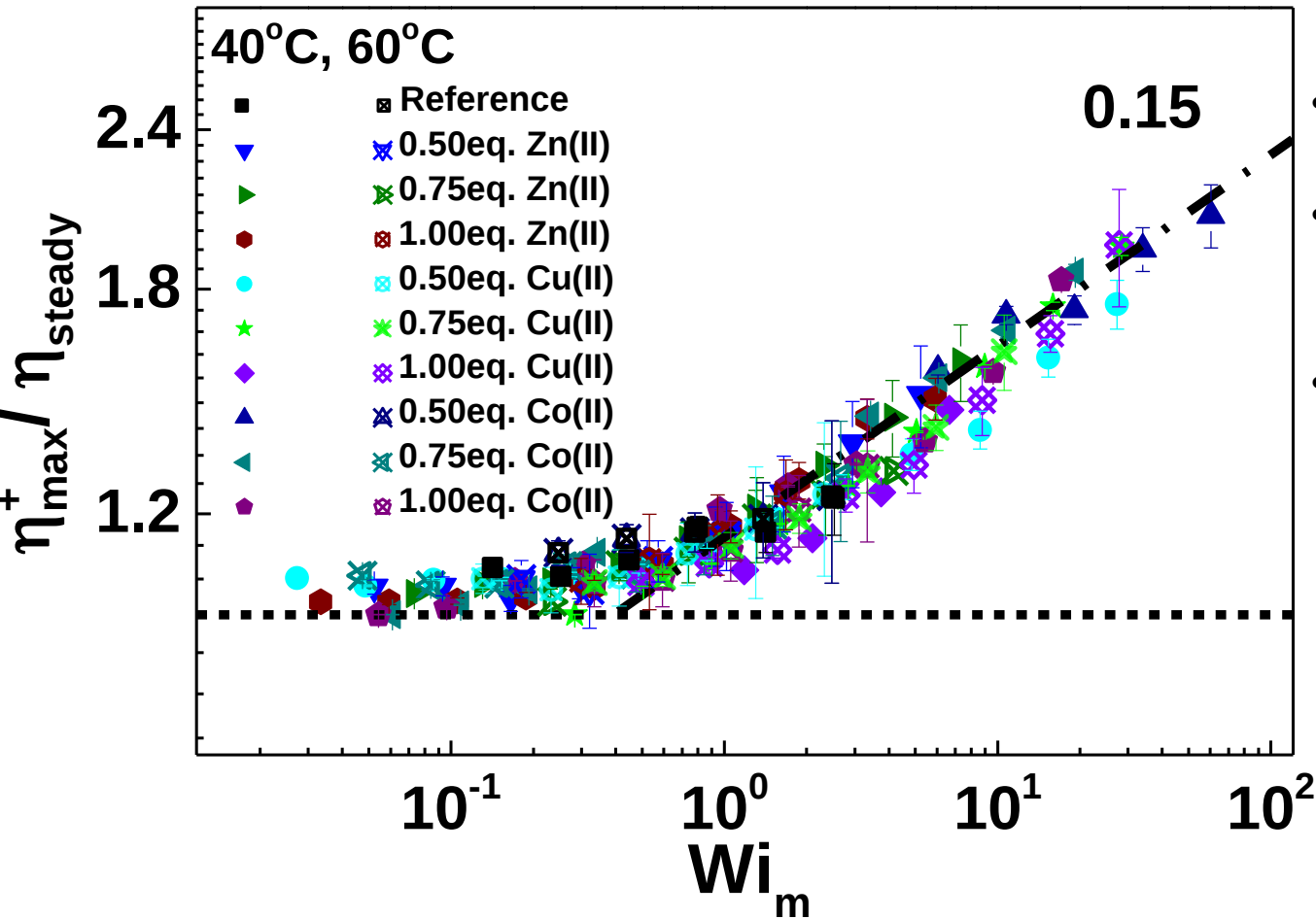
Ion Strength
Ion Content
Temperature



- Steady state stresses of all systems collapse into a universal trend
- Coupling of star-arm and sticker dynamics
- Universal 0.85 slope indicates dynamic bond predominance for $Wi < 1$
- becomes weaker indicating the control of the star-arm dynamics at $Wi > 1$

Fractional viscosity overshoot

Ion Strength
Ion Content
Temperature

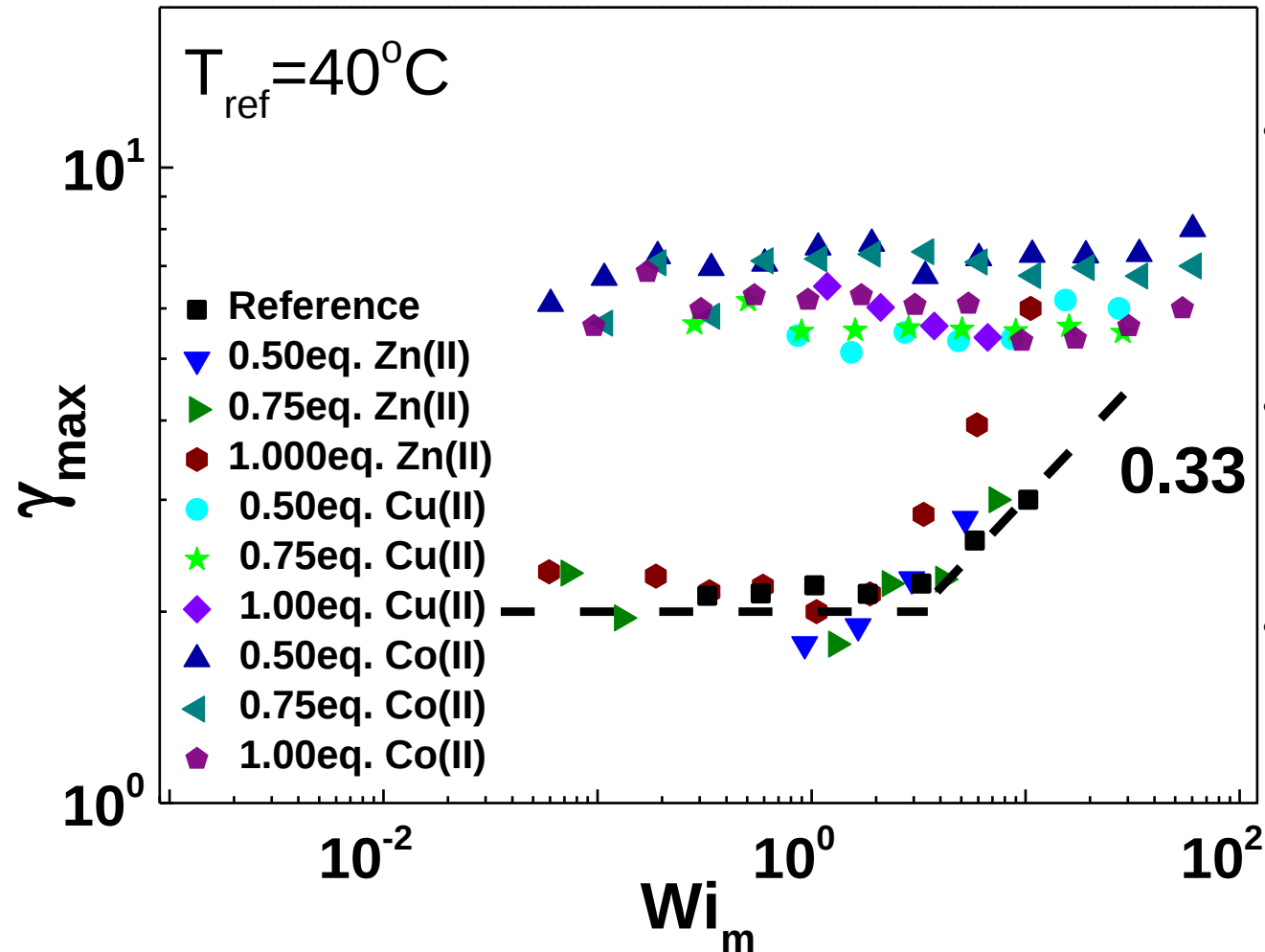


- Data of all systems collapse in a universal trend
- Fractional overshoot is a measure of mechanical coherence.
- The associated networks exhibit an improved coherence compared to the reference sample without metal ions, reaching larger yield stress and having enhanced topological/bond rearrangement until steady state

Polgar, L. M. *et al.*, *Macromolecules* **48**, 6662–6671 (2015).

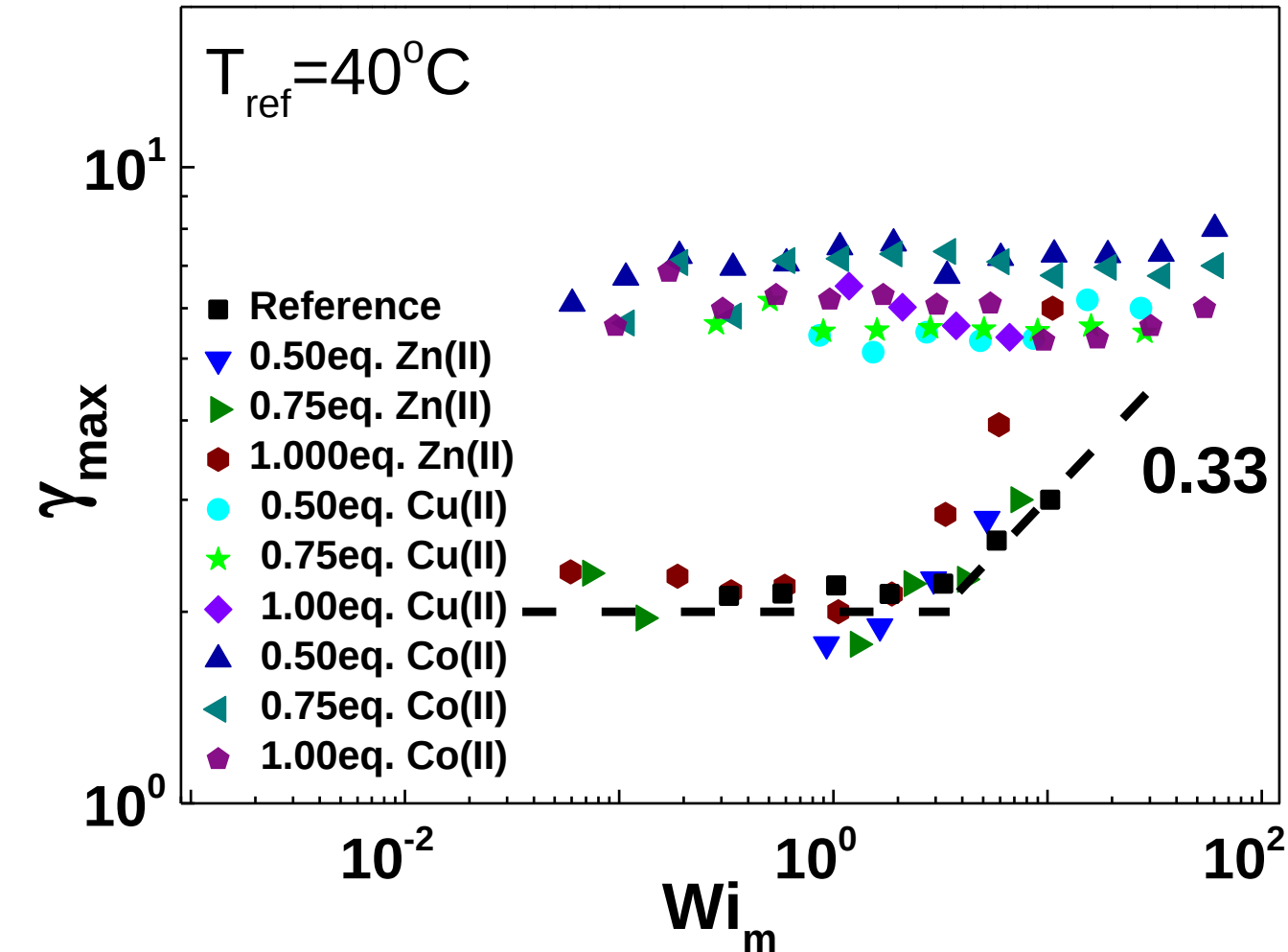
Costanzo, S. *et al.*, *Macromolecules* **49**, 3925–3935 (2016).

Maximum (yield) strain

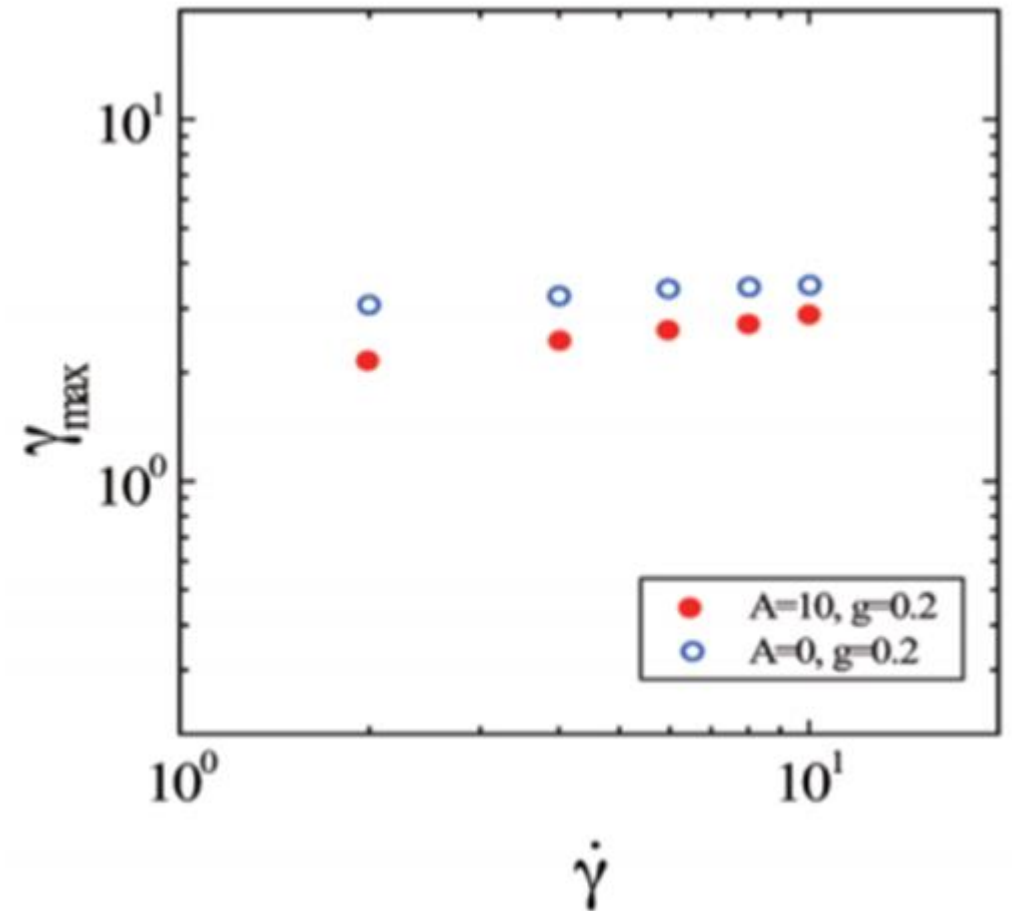


- Weakly associated networks are dominated by polymeric response. First orientation and then stretching at $Wi > 1$ with the 0.33 slope of entangled polymers
- Yielding in strongly associated networks is independent to Wi , the network strands seem fully deformed- strain saturation.
- Accumulation of strain indicates that the lifetime of the dynamic bonds remains unchanged during deformation

Maximum (yield) strain



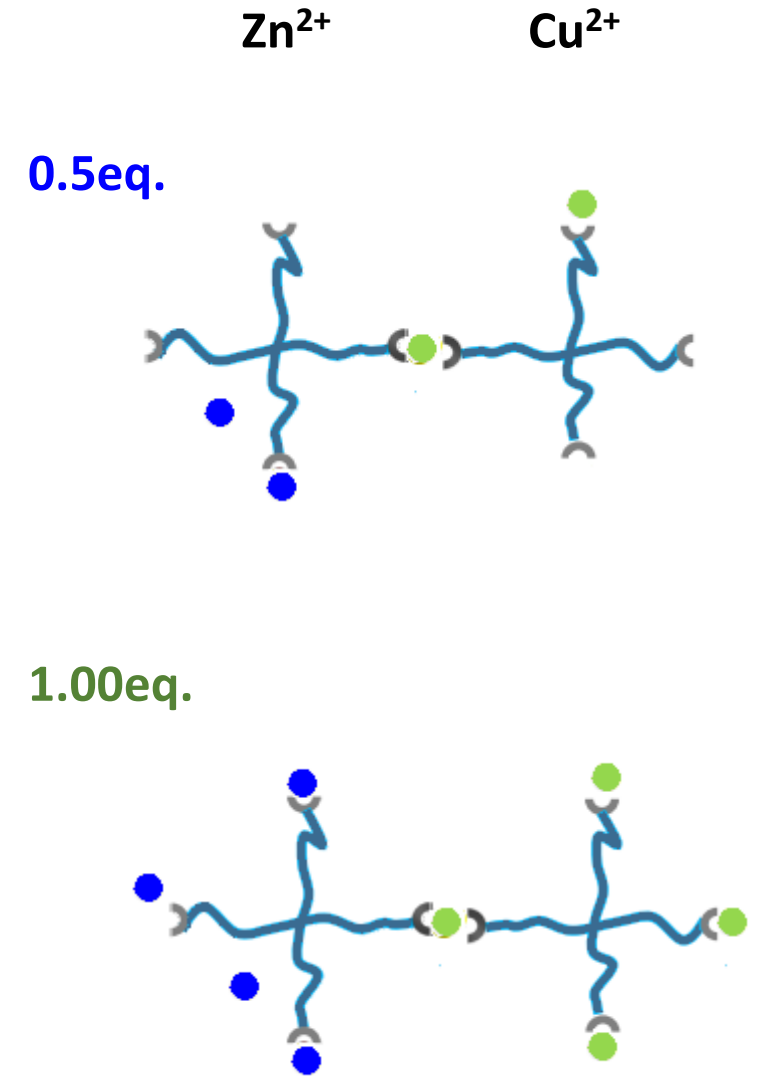
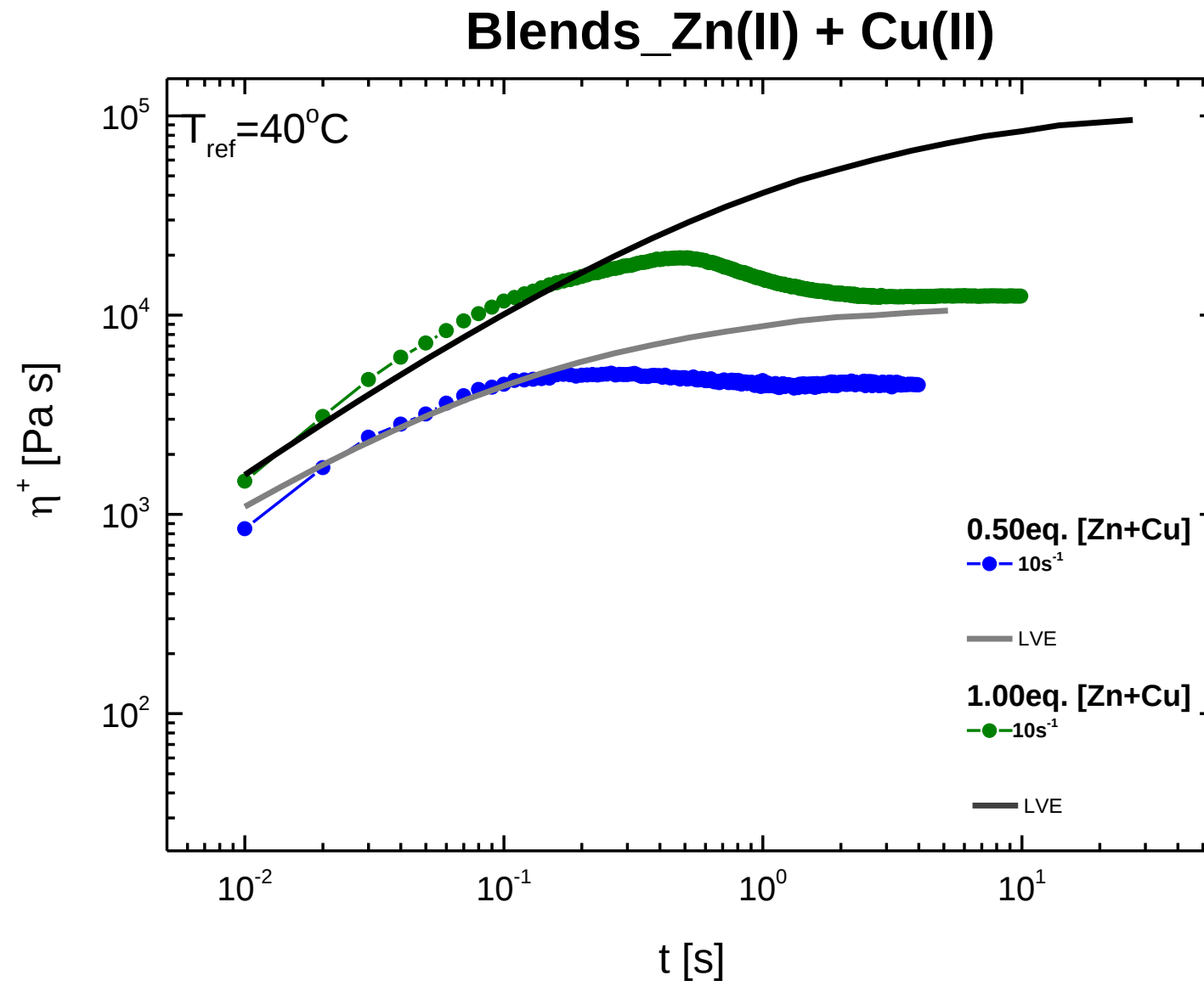
Fixed coupling constant, g



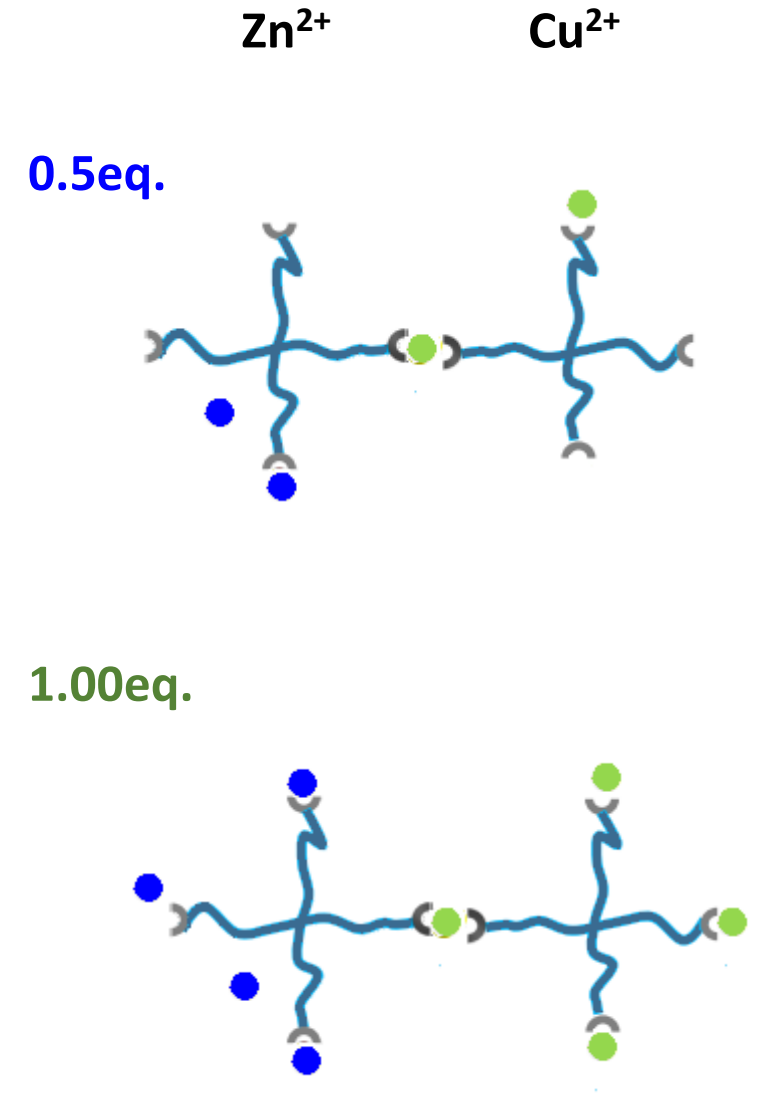
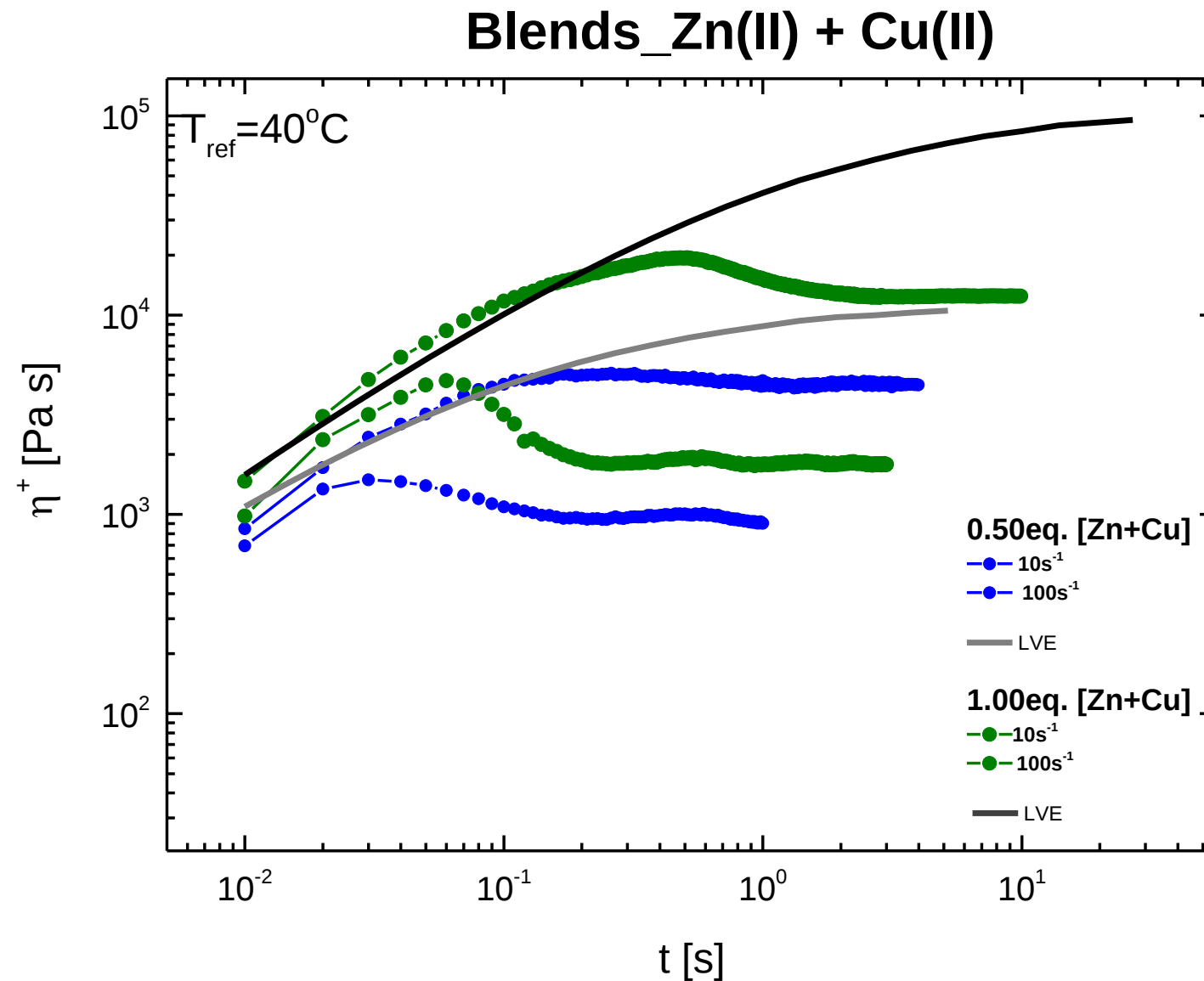
Koga, T., et al., *Langmuir* **25**, 8626–8638 (2009)

Zhang, Q. et al., *Macromolecules* **52**, 660–668 (2019)

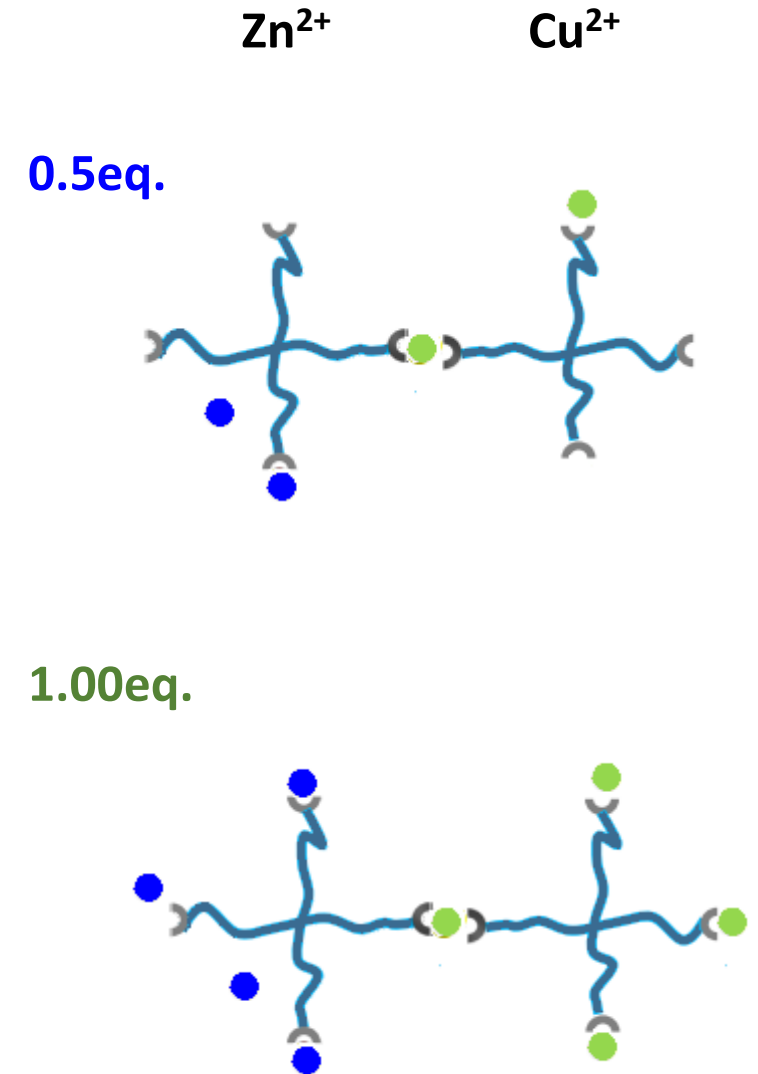
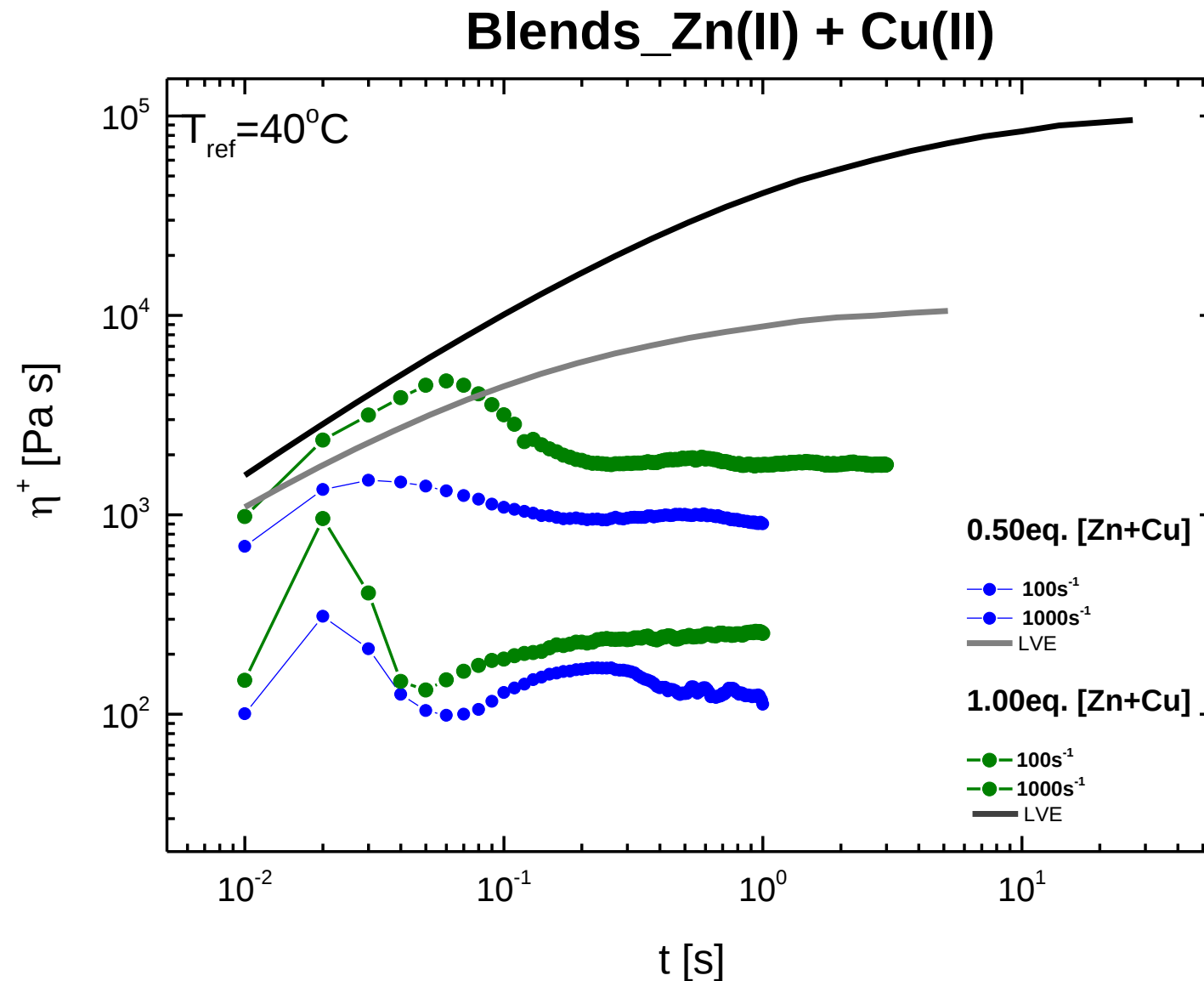
Double sticker dynamic networks



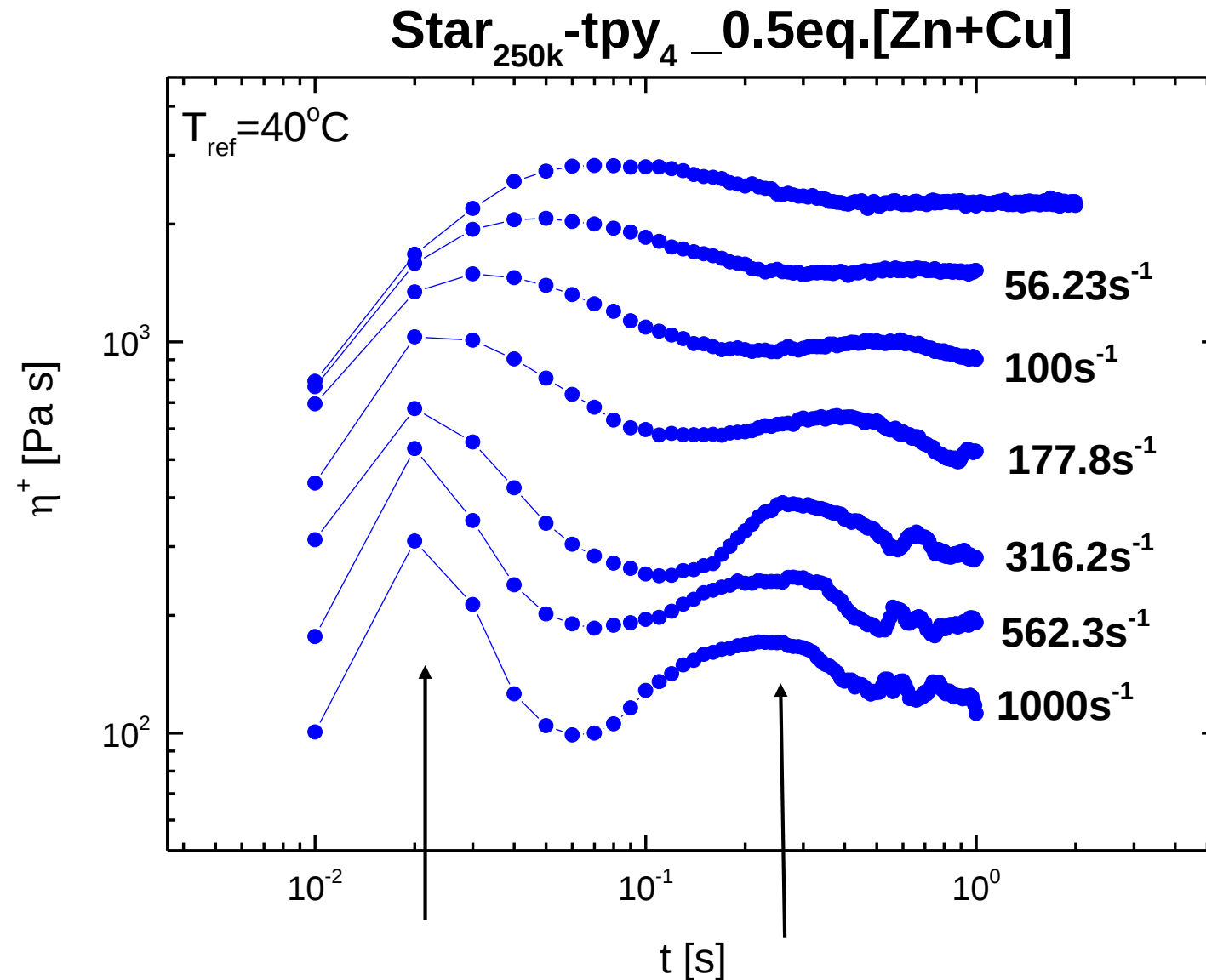
Double sticker dynamic networks



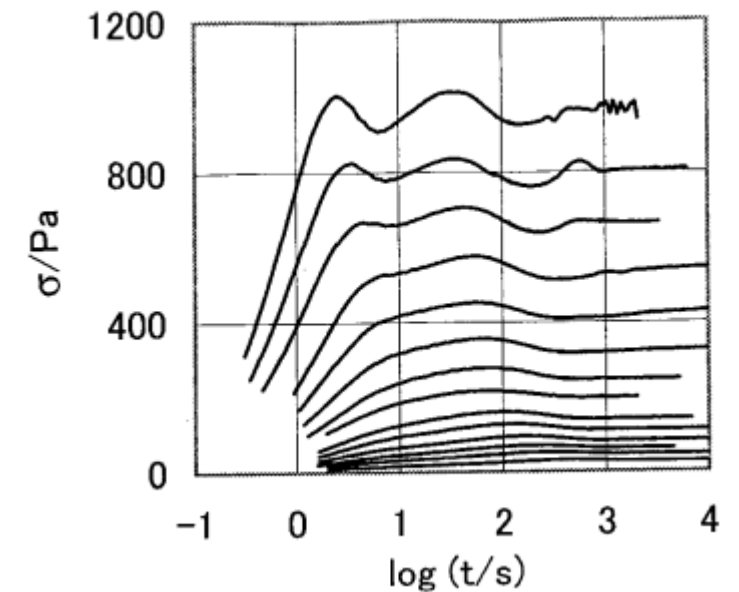
Double sticker dynamic networks



Double peak in transient response



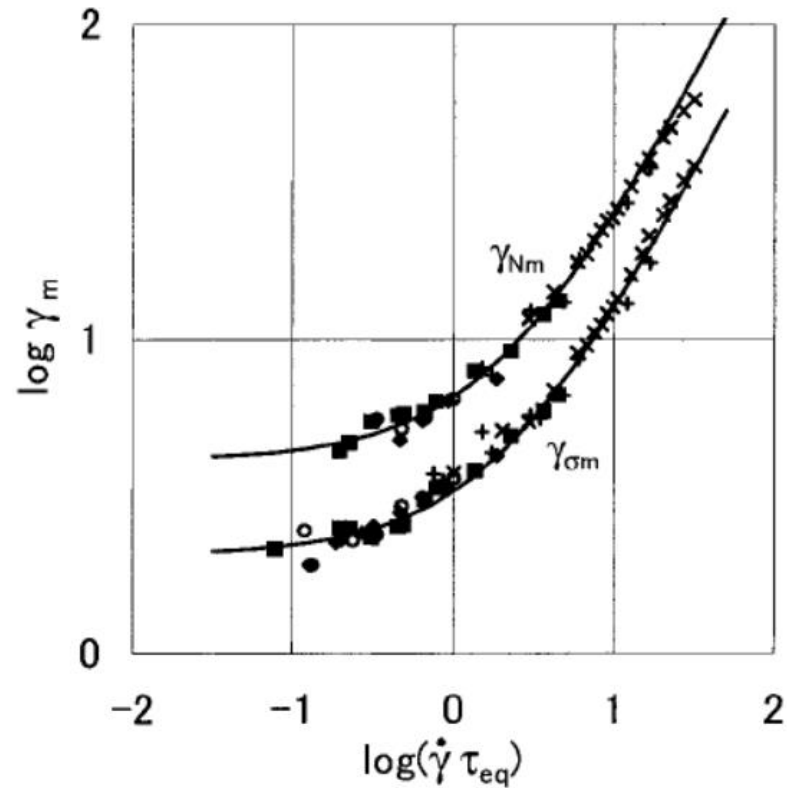
PS solutions with bimodal
molecular weight
distributions



Osaki, K., et al., *Journal of Polymer Science: Polymer Physics* **38**, 2043–2050 (2000).

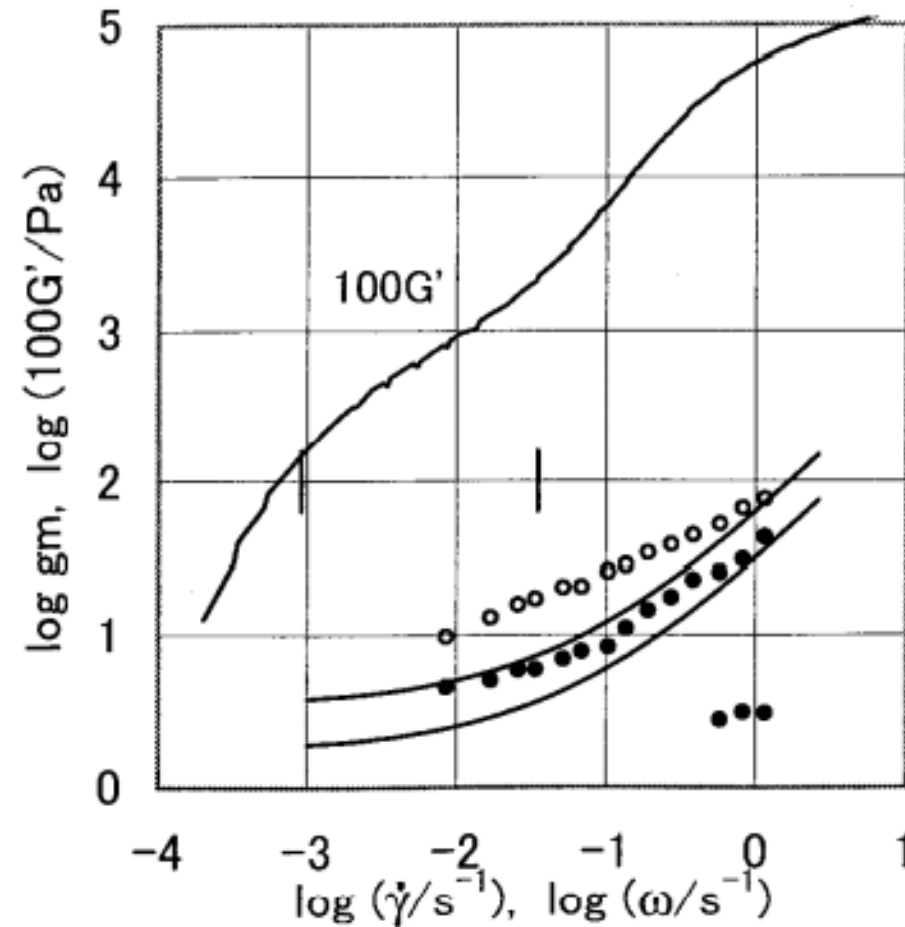
Snijkers, F. et al., *ACS Macro Lett.* **2**, 601–604 (2013).

Evidence if double peak: interplay of relaxation times of blend components



$$\gamma_{max} = 0.87[1.59 + (\dot{\gamma} \tau_R)^{2/3}]^{3/2}$$

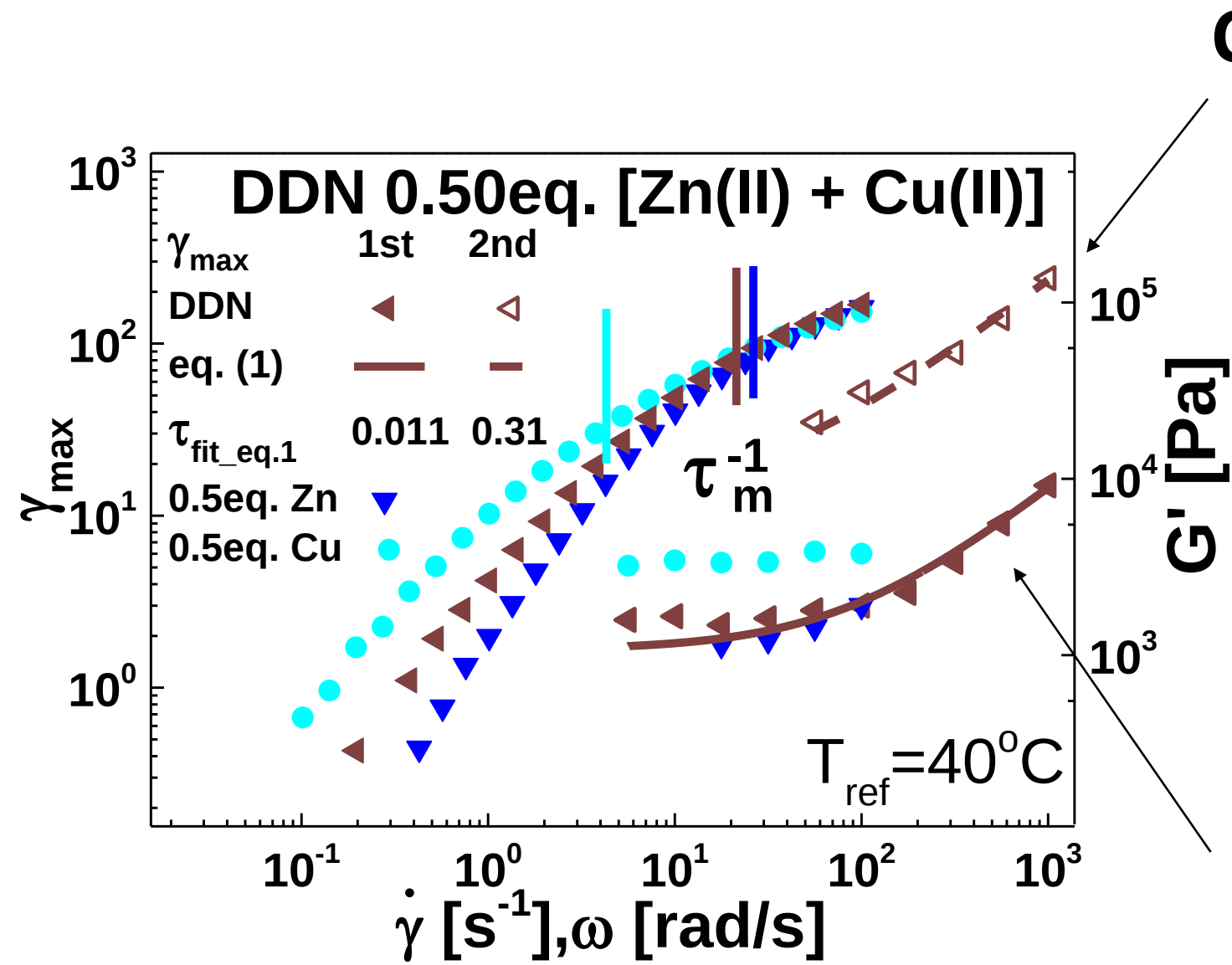
Osaki, K., et al., *Journal of Polymer Science: Polymer Physics* **38**, 1917–1925 (2000).



Osaki, K., et al., *Journal of Polymer Science: Polymer Physics* **38**, 2043–2050 (2000).

Snijkers, F. et al., *ACS Macro Lett.* **2**, 601–604 (2013).

Peak allocation



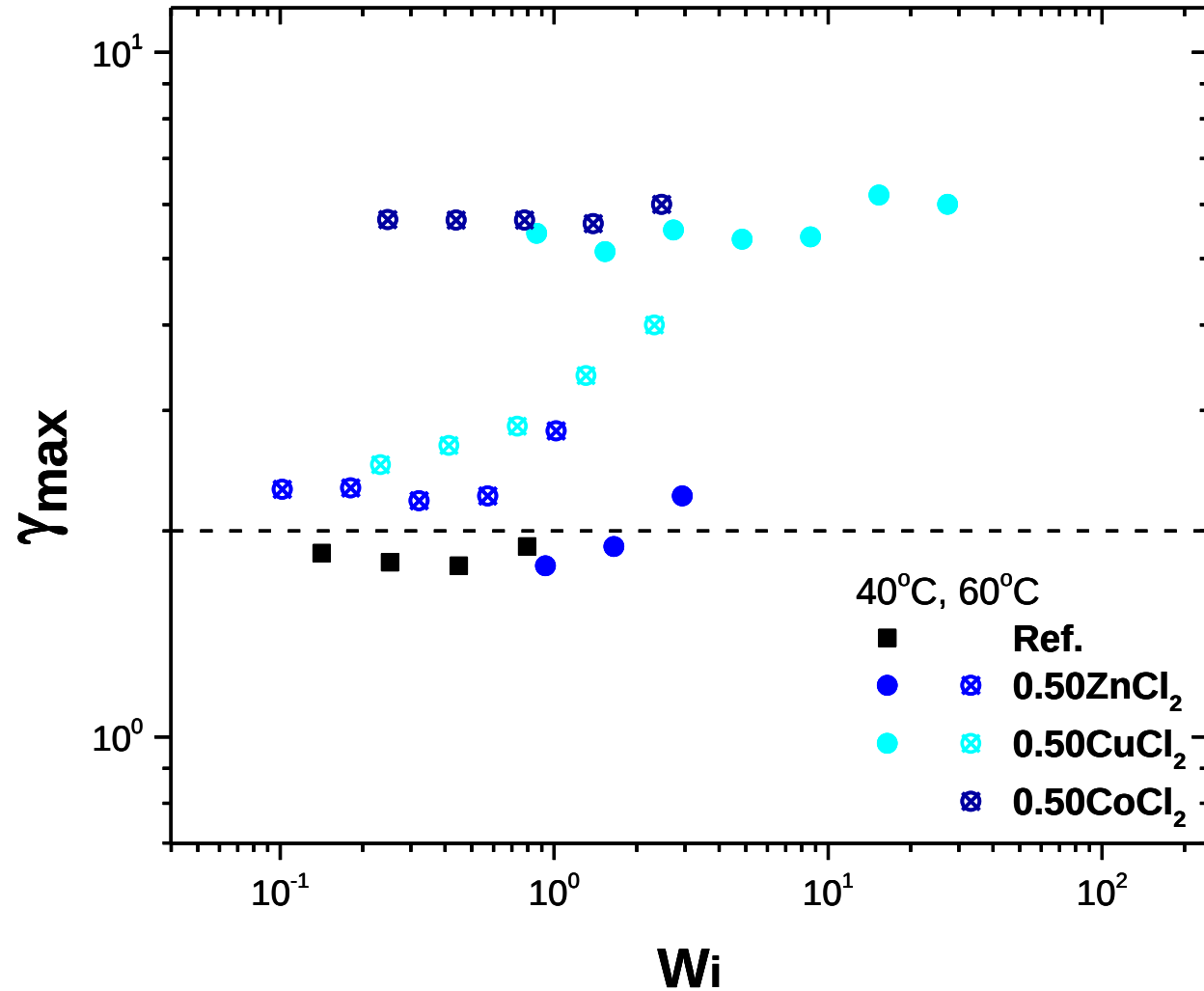
Cu²⁺

$$\gamma_{max} = A[1 + (\dot{\gamma}\tau)]^B$$

40°C	τ_{cross} [sec] Experimental	τ_{cross} [sec] TMA
0.5Zn	0.052	0.047
0.5Cu	0.27	0.31
0.5[Cu+Zn]	0.068	0.071

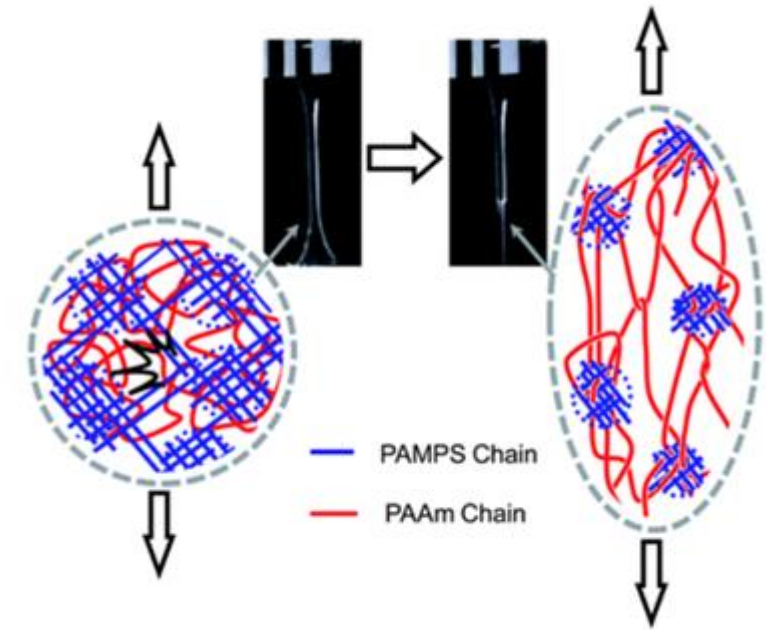
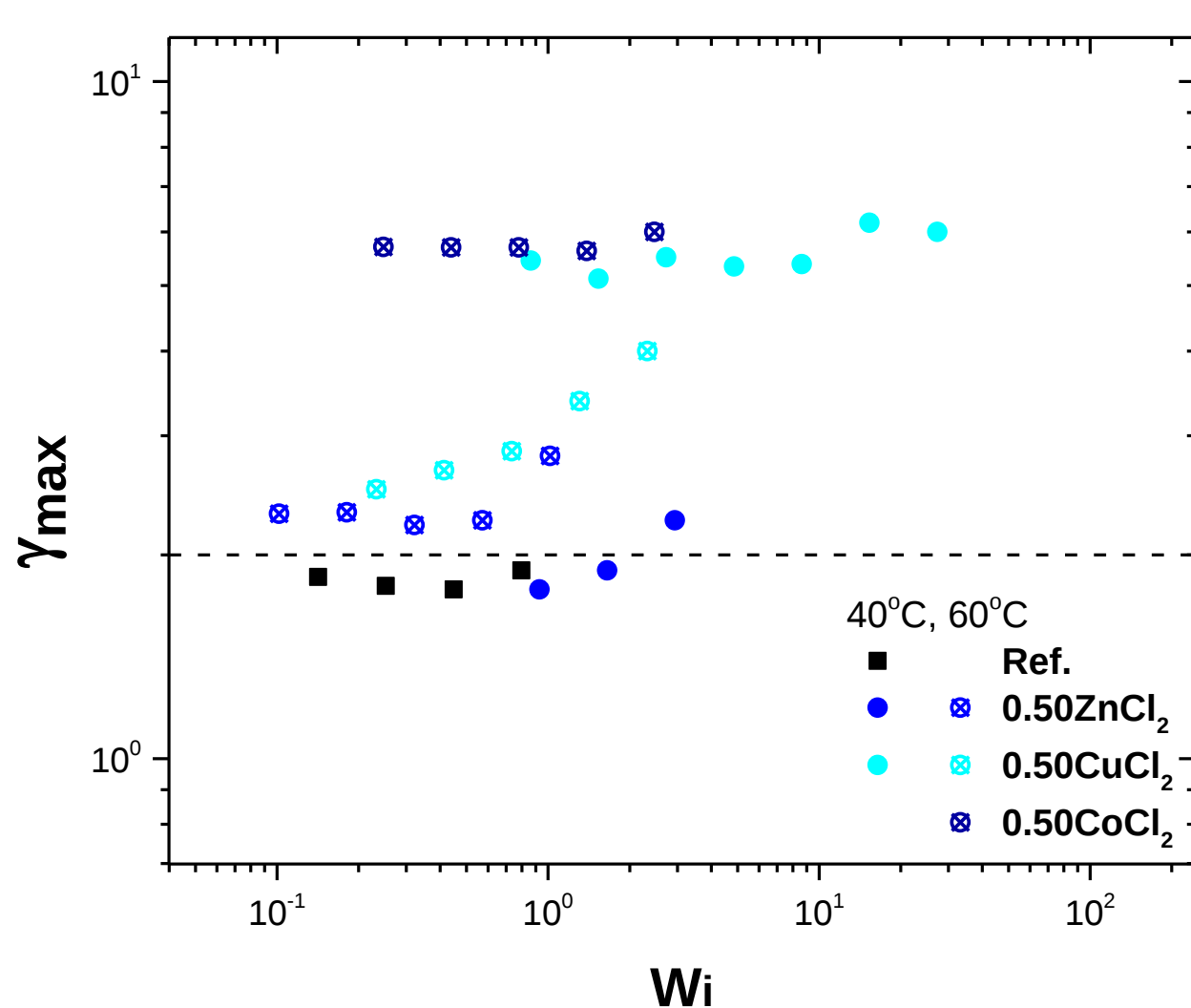
Zn²⁺

Proposed criterion for double overshoot



- Double peak is attributed to the coupled response of the two single network components
- The proposed criterion for double dynamics: one relatively weak single network with disassociation rate that depends shear rate (sacrificial bond) while the disassociation rate of the other is independent of the applied rate (deformability & coherence)

Proposed criterion for double overshoot



Gong, J. P. *Soft Matter* **6**, 2583–2590 (2010)

Sun, T. L. *et al. Nat Mater* **12**, 932–937 (2013)

Filippidi, E. *et al., Science* **358**, 502–505 (2017)

Li, Y. *et al., Chemistry of Materials* (2019)

Li, S. *et al., Advanced Functional Materials* **30**, 2000130 (2020).

Conclusions

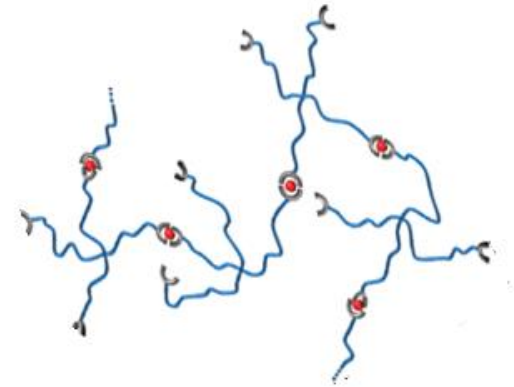
- Systematic nonlinear rheological study unravels the role of molecular parameters on properties
- Enhanced mechanical properties, coherence and deformability
- Role of associations strength is important in ultimate tunability- from polymer to bond dominated response

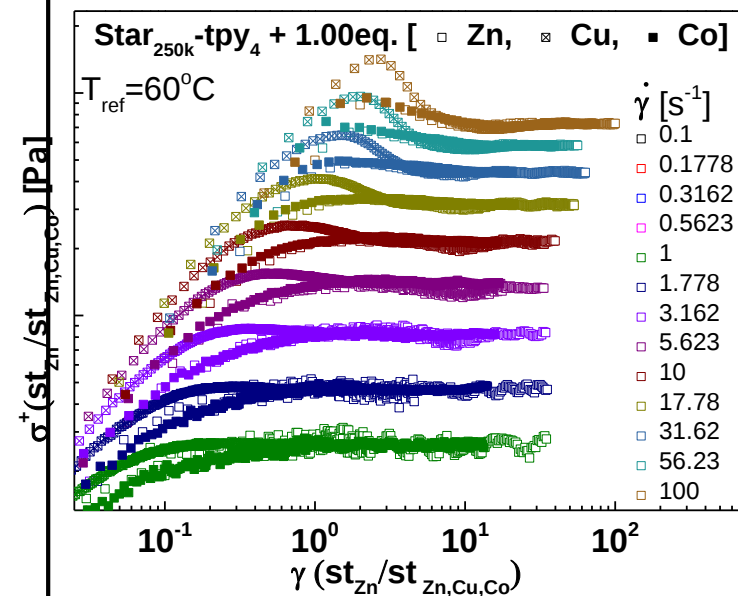
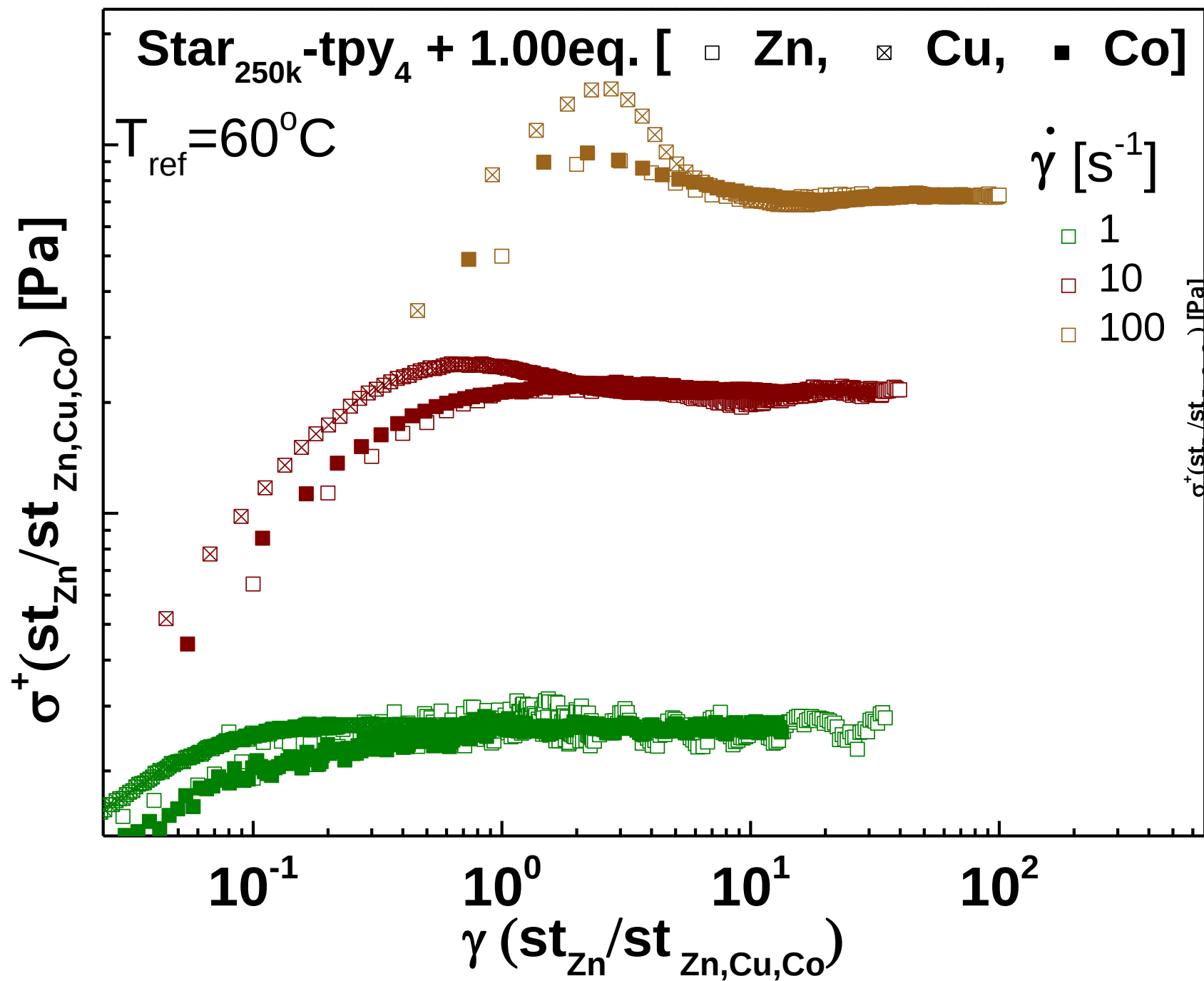
Perspectives

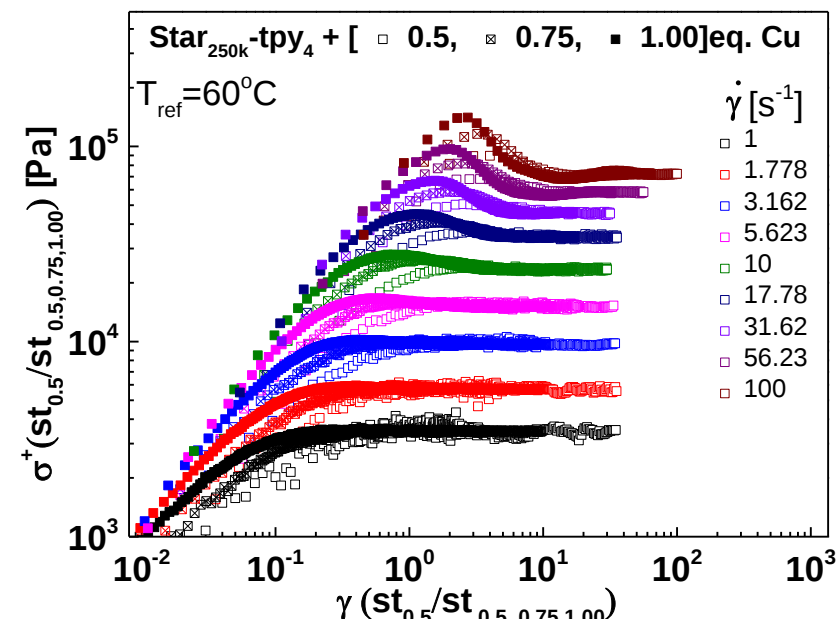
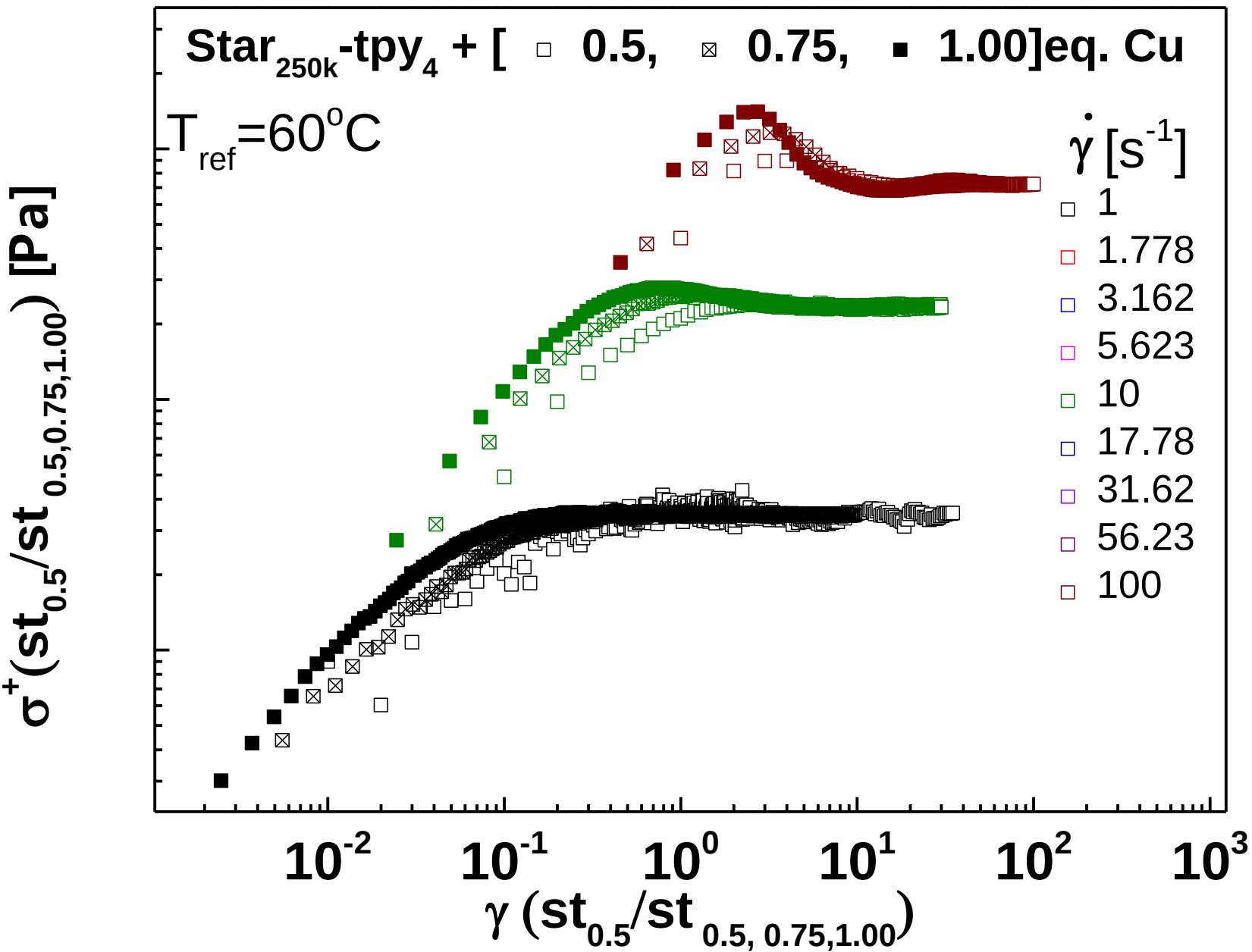
- Further double peak study, including extensional deformation
- Understanding the relaxation mechanism of transient networks
- Rationalization of experimental observations using TMA

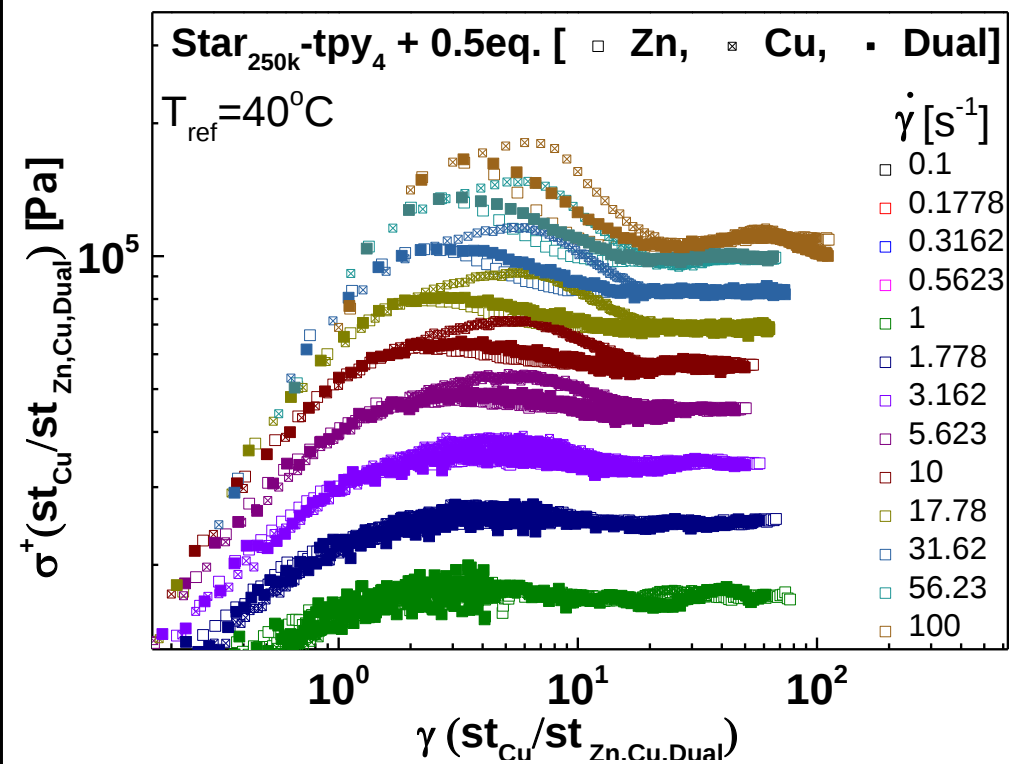
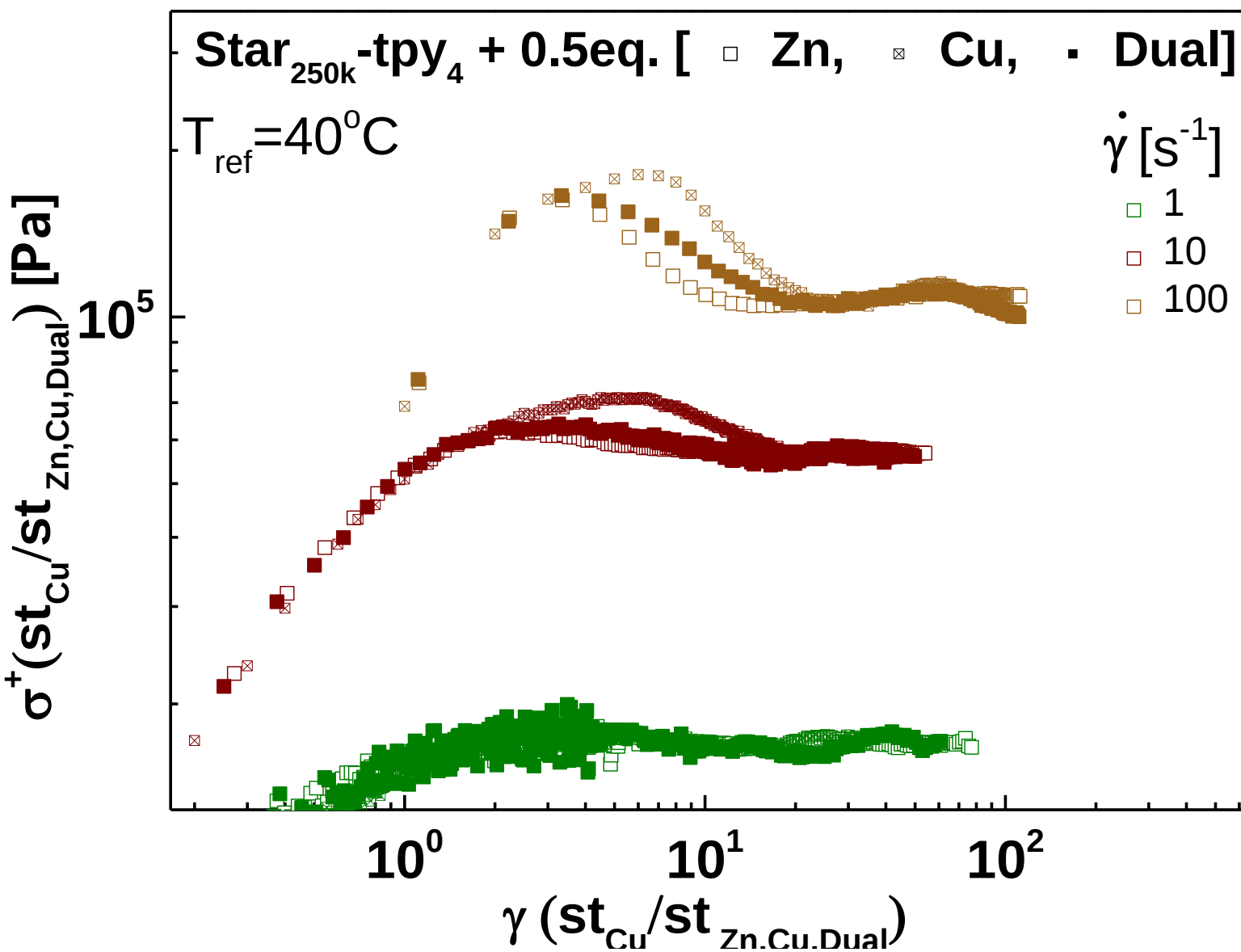
Thank you for your attention!

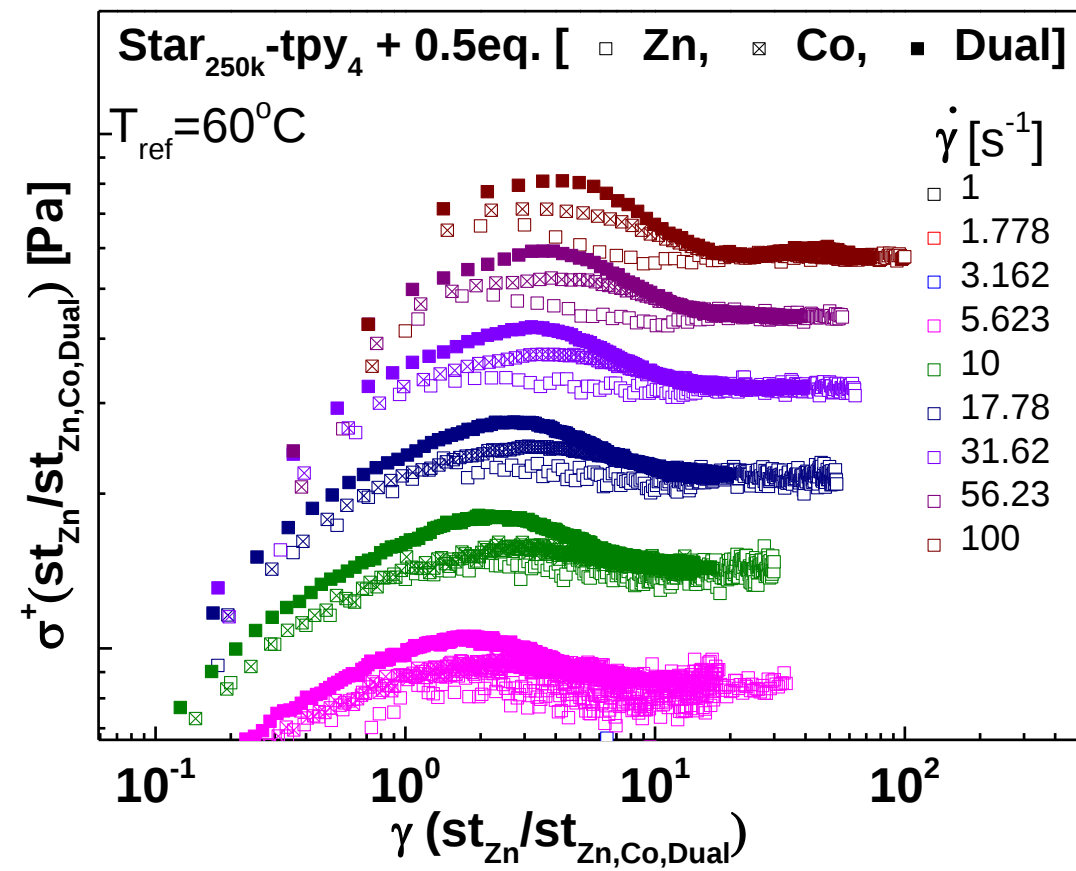
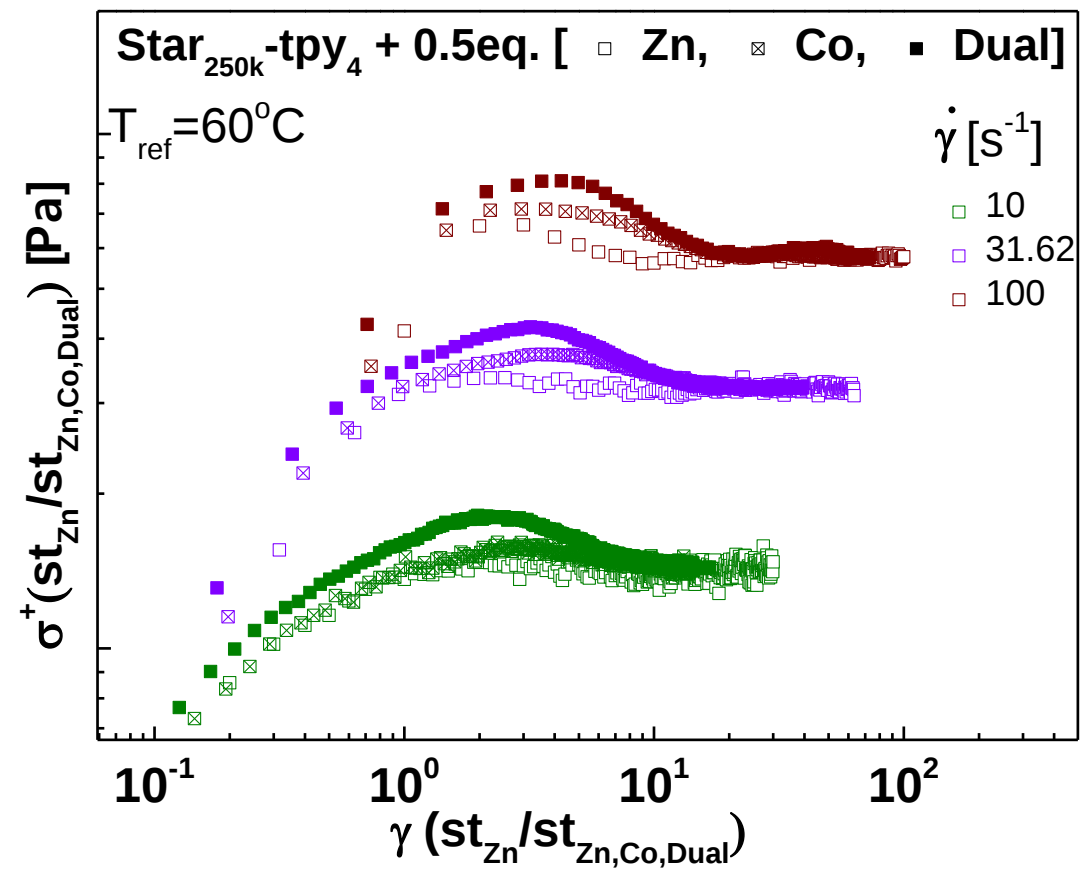
Questions?





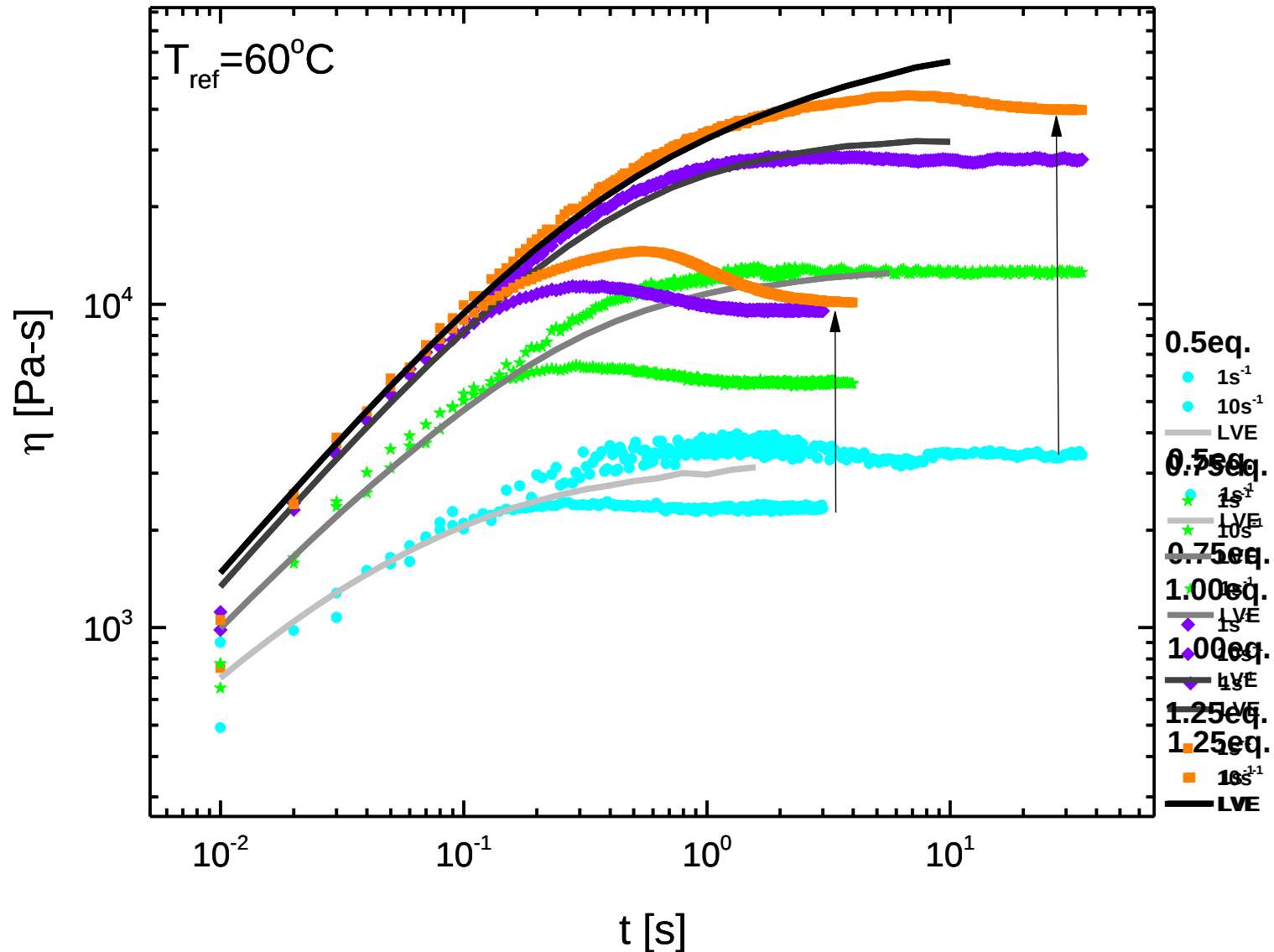






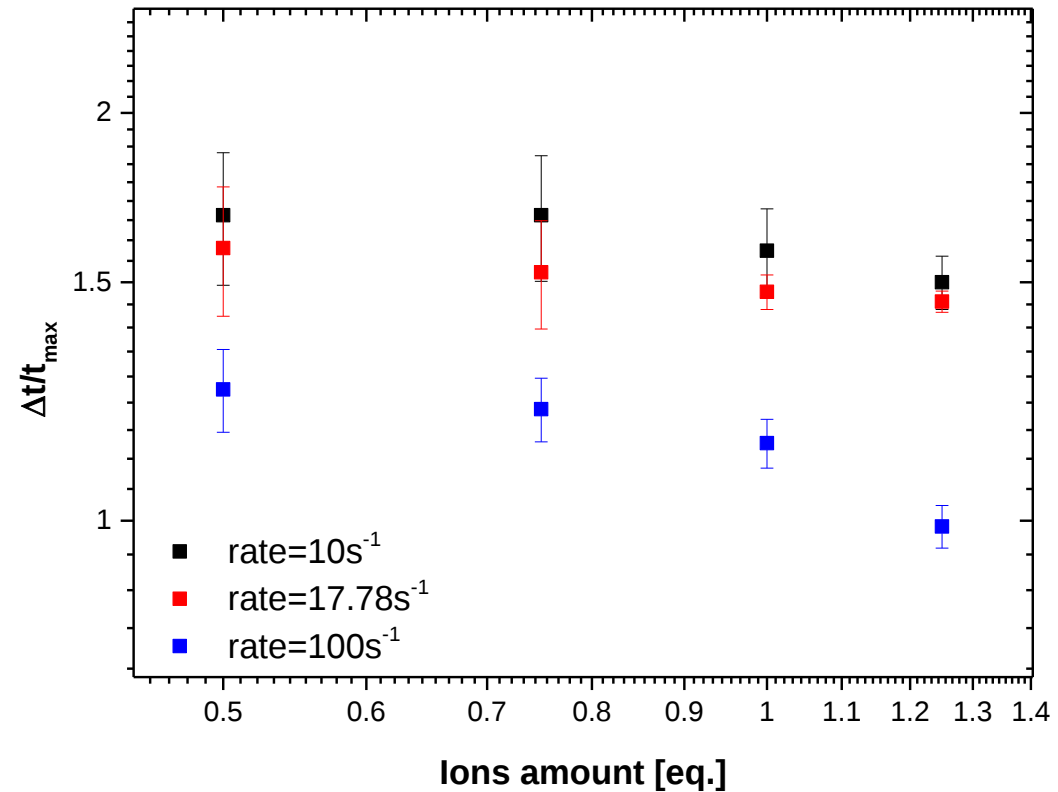
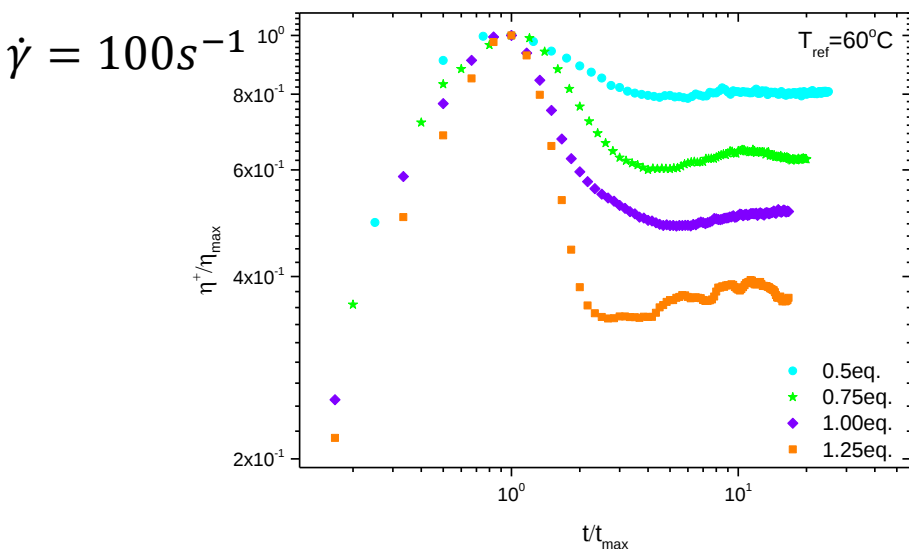
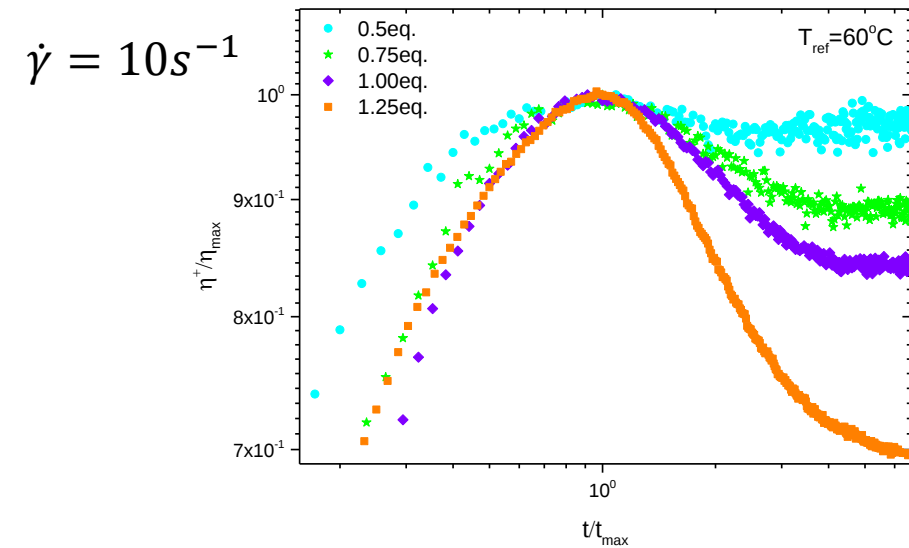
Single Networks_Ion Content

Cu



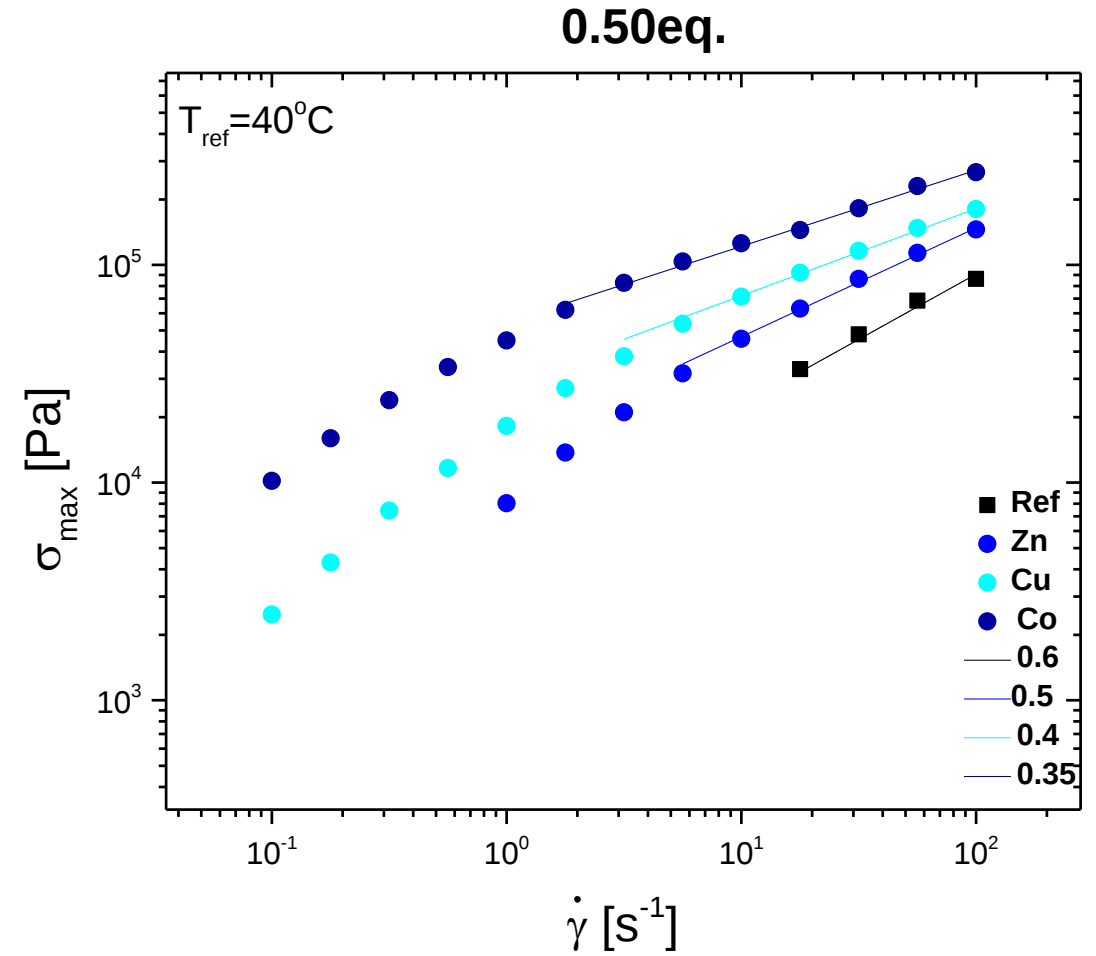
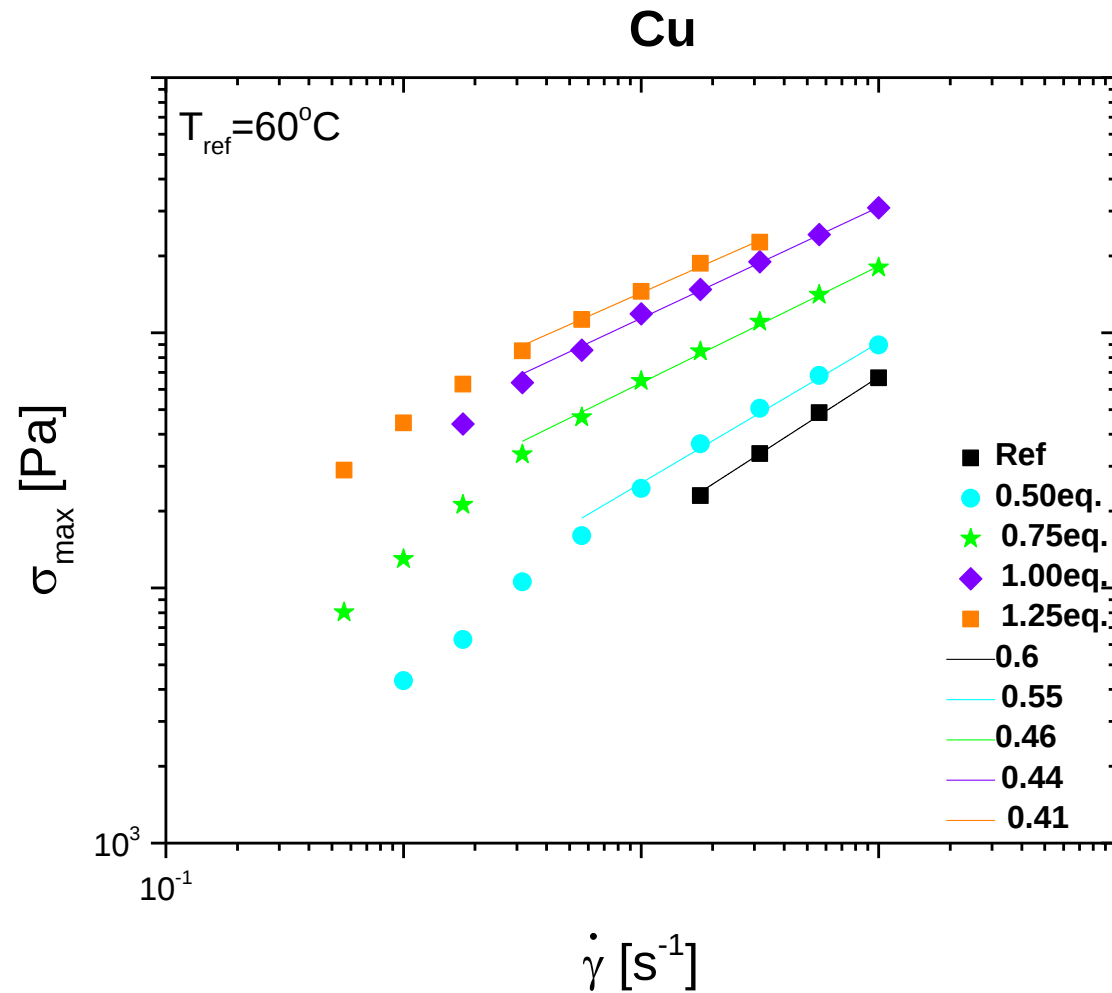
Half Height Width

Ion Nature
Ion Content
Temperature

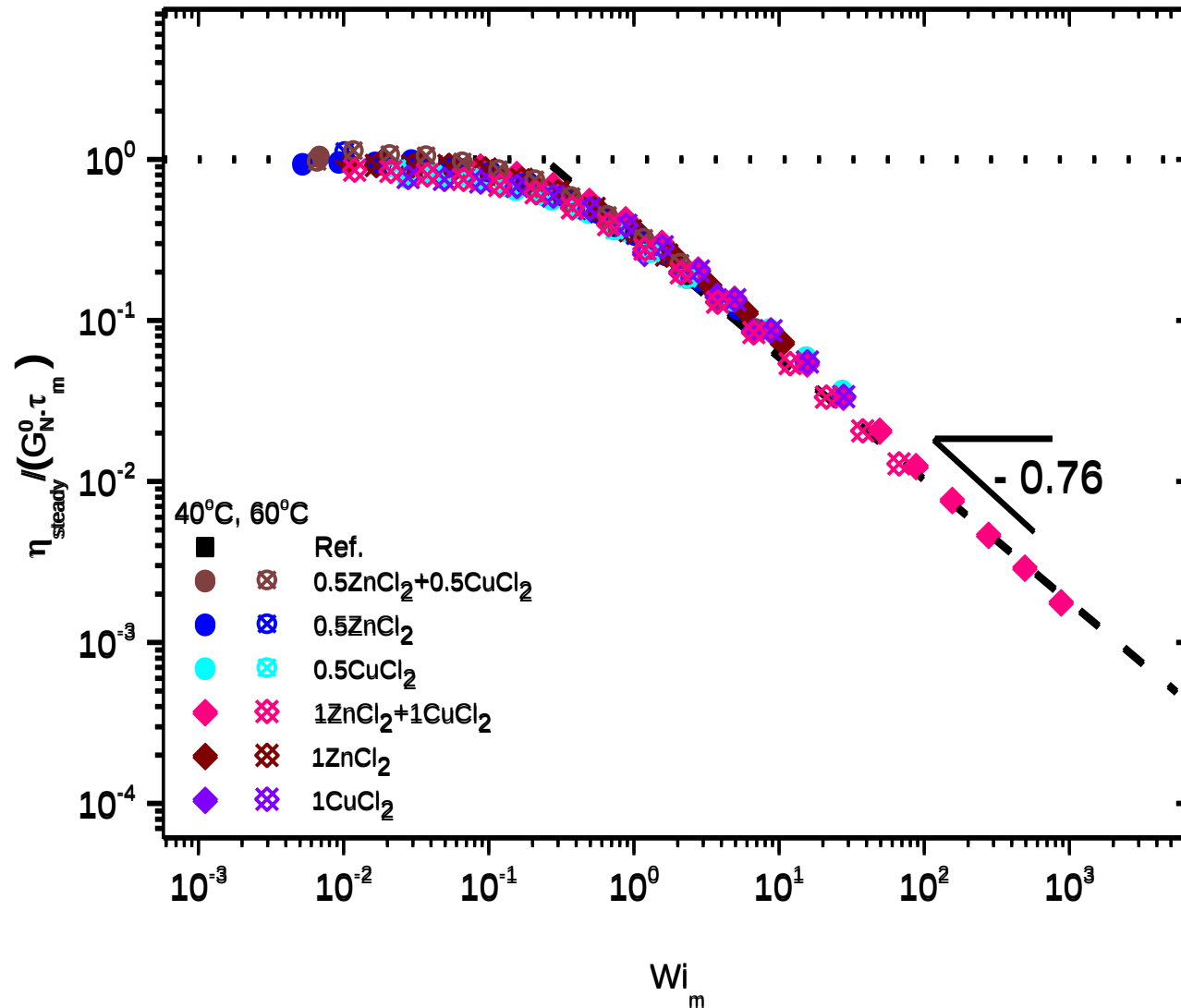


Half height width of peak transient viscosity decrease as the ions content increase i.e. elastic recoil to the initial structure is enhanced in abundant of metal ions

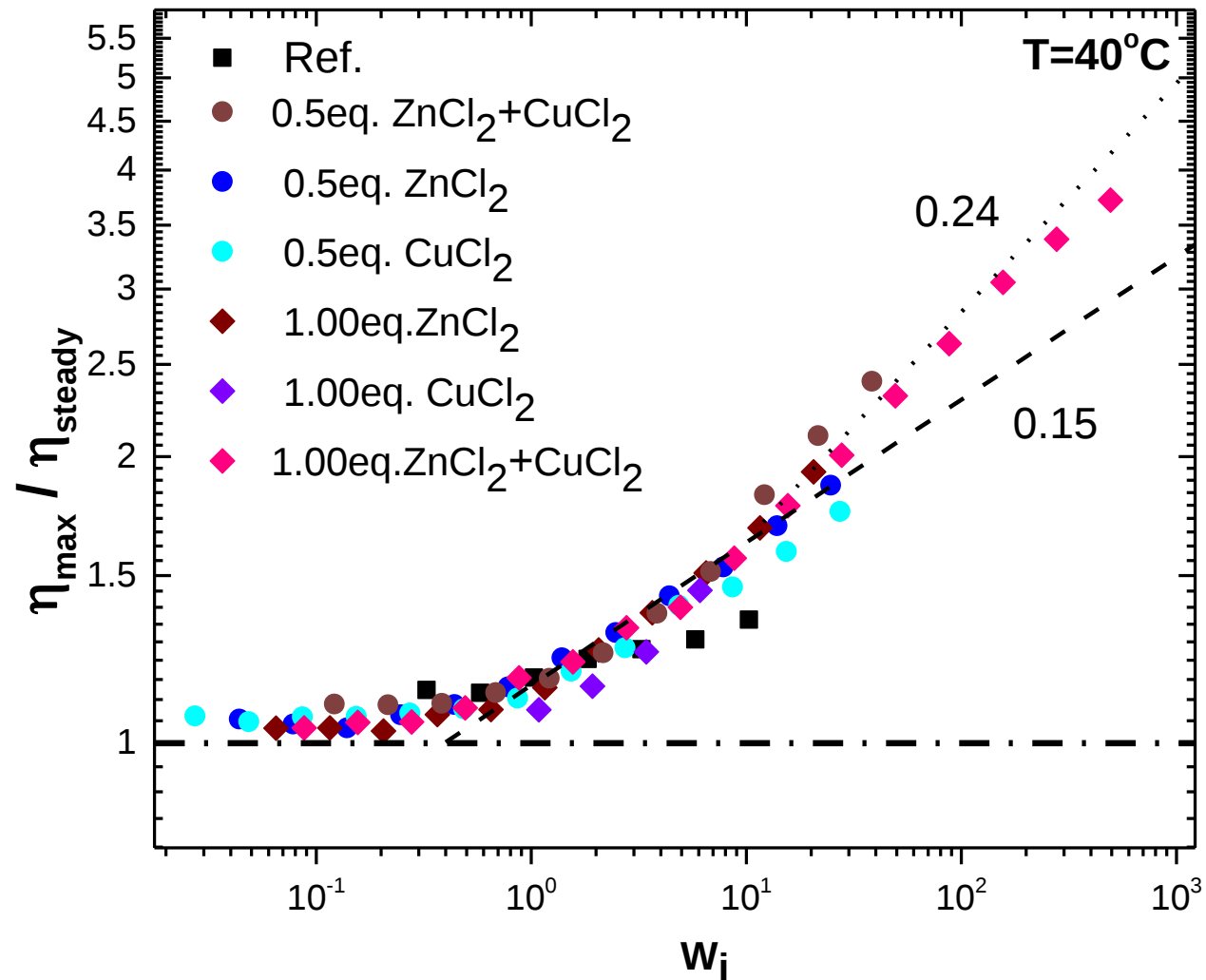
Yield Stress



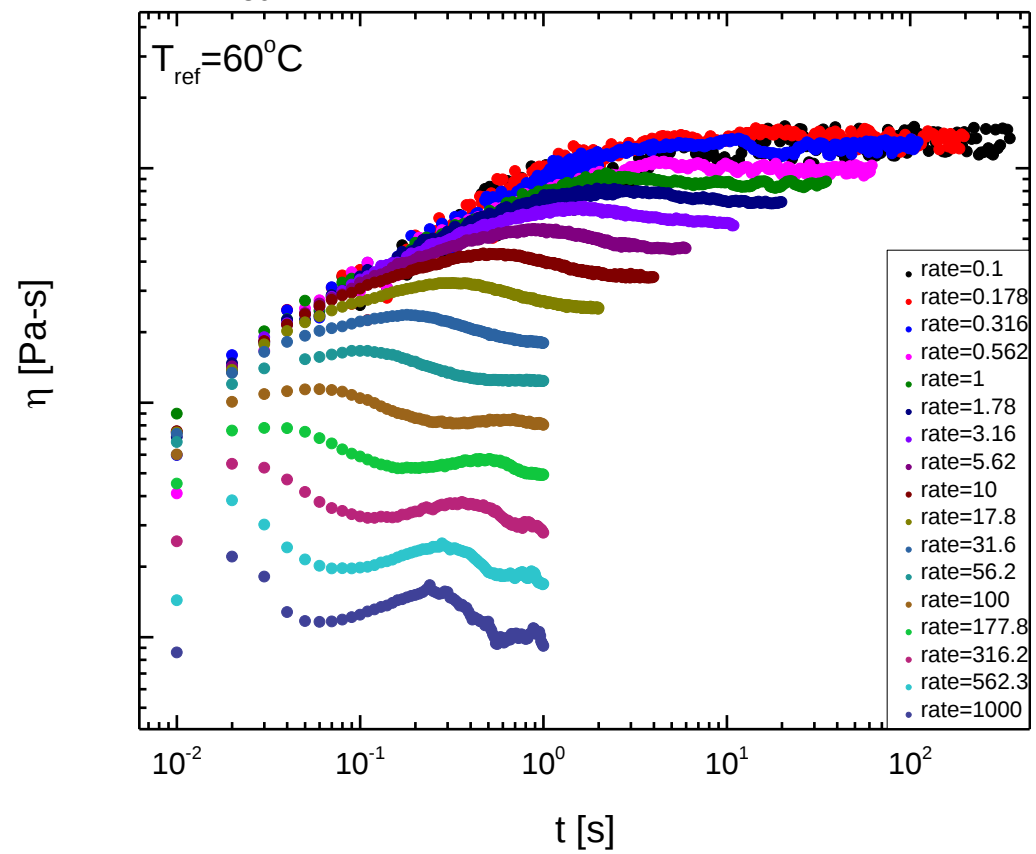
Shear Thinning_Single & Double Sticker Lifetime

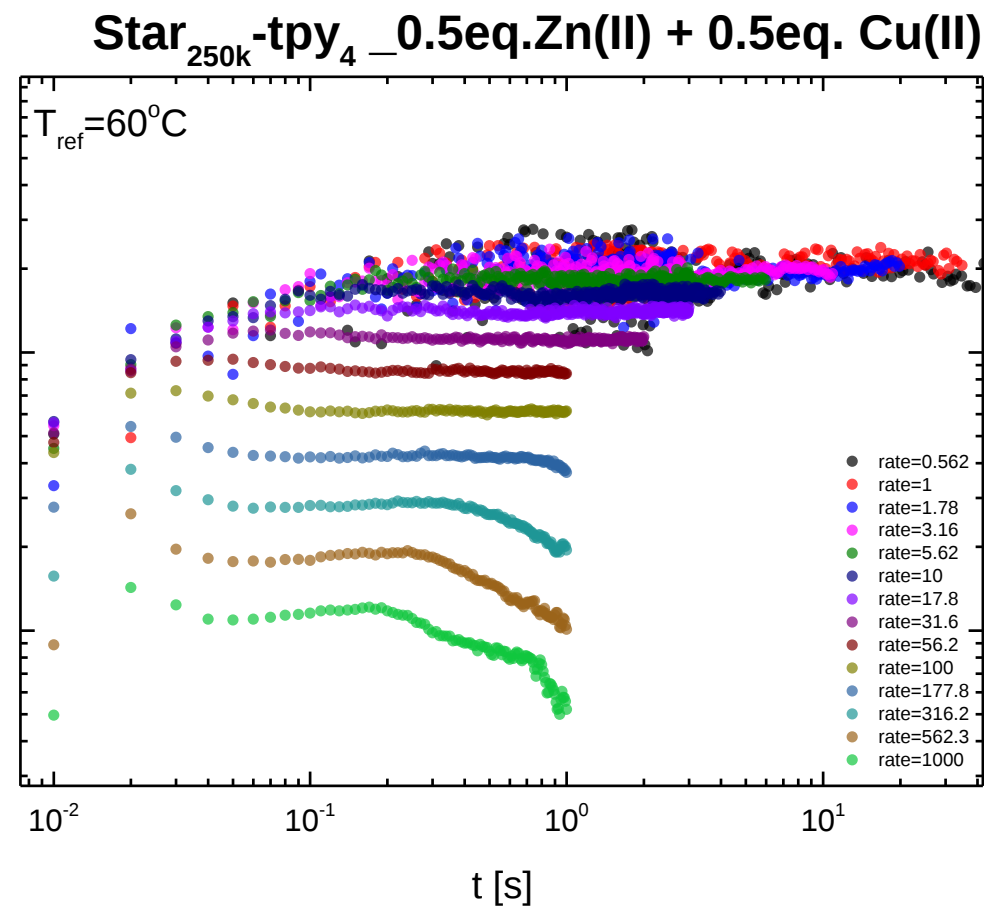
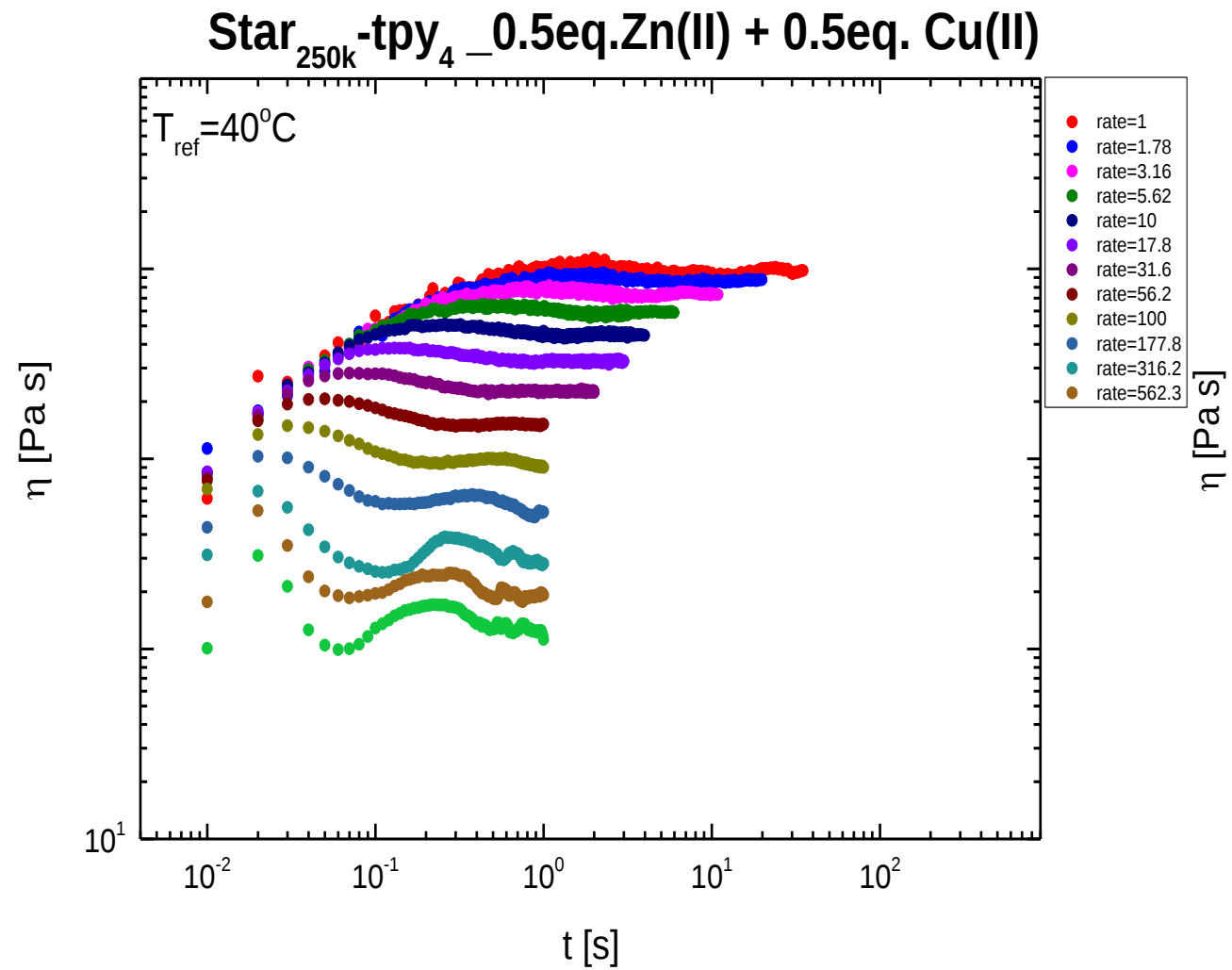


Fractional viscosity overshoot_Single & Double Sticker Lifetime

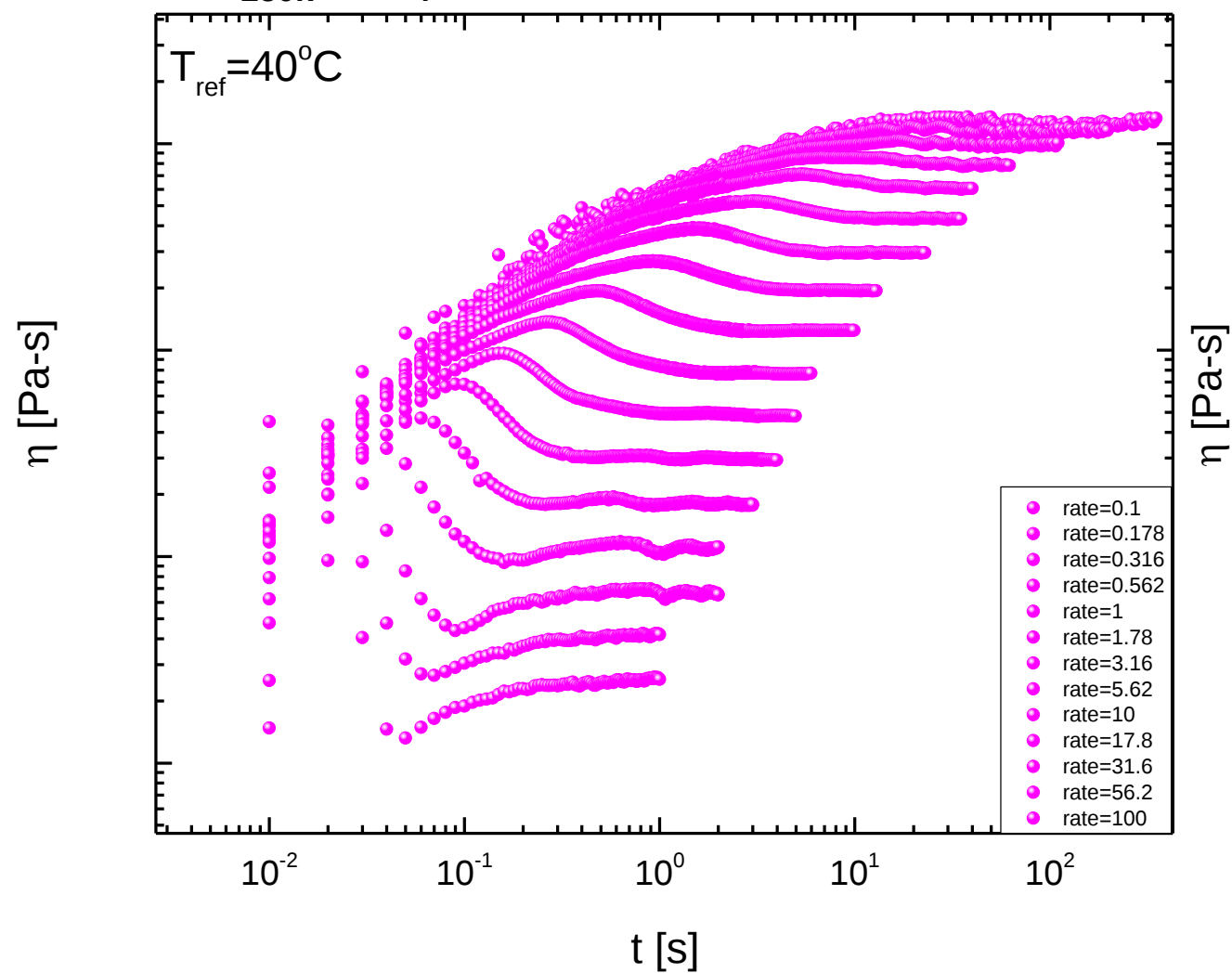


Star_{250k}-tpy₄ + 0.5eq. Zn(II) + 0.5eq. Co(II)





Star_{250k}-tpy₄ + 1.00eq. Zn(II)+1.00eq. Cu(II)



Star_{250k}-tpy₄ + 1.00eq. Zn(II)+1.00eq. Cu(II)

